

Electronic Supplementary Information (ESI)

Super-Hydrophilic Leaflike Sn_4P_3 on the Porous Seamless Graphene–Carbon Nanotube Heterostructure as an Efficient Electrocatalyst for Solar-Driven Overall Water Splitting

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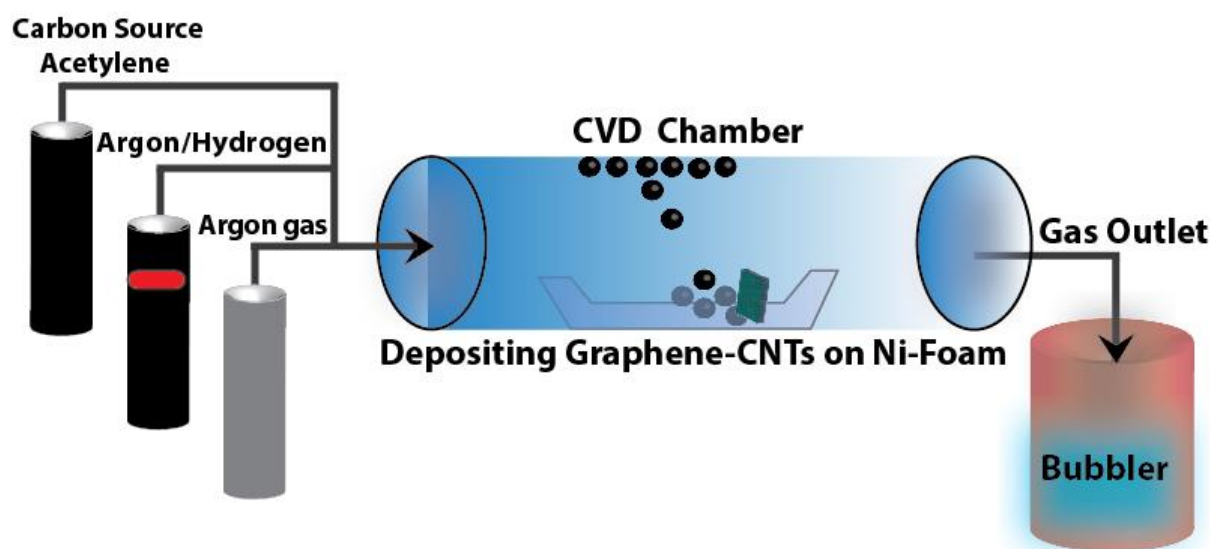


Figure S1. Schematic illustration of the synthesis of Graphene-CNTs on 3D porous Ni-foam by CVD method.

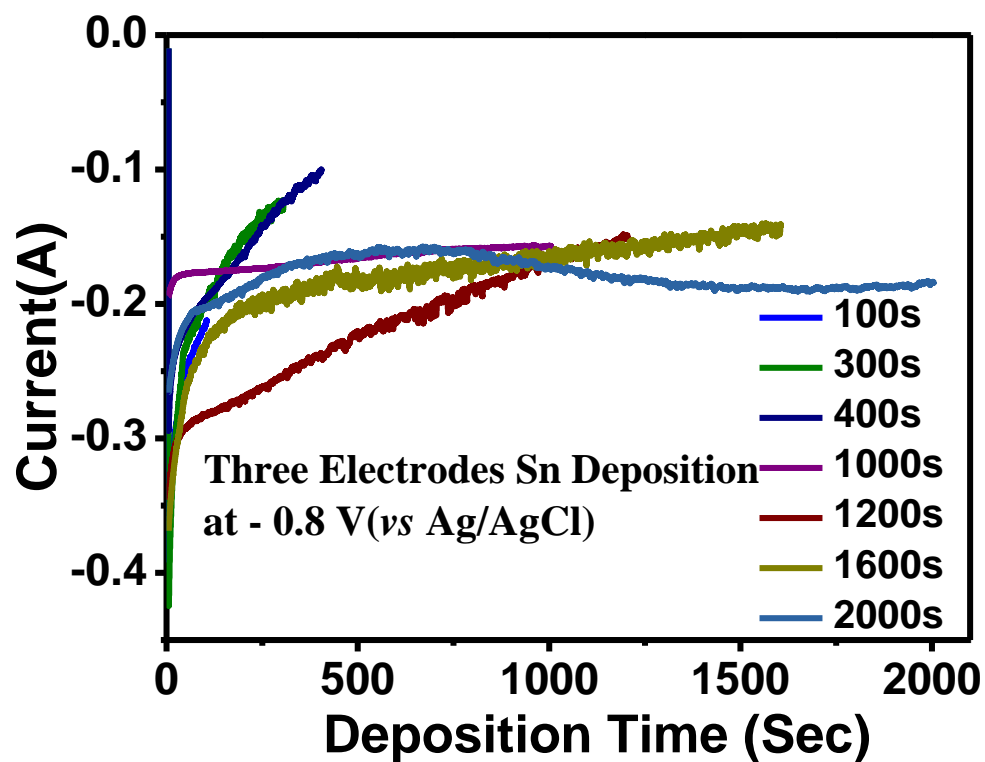


Figure S2. Chronoamperometric deposition of Sn on Ni-Gr-CNTs heterostructure in the three-electrode system at an applied voltage of -0.8 V (vs Ag/AgCl) with varying deposition time of 100s to 2000s.

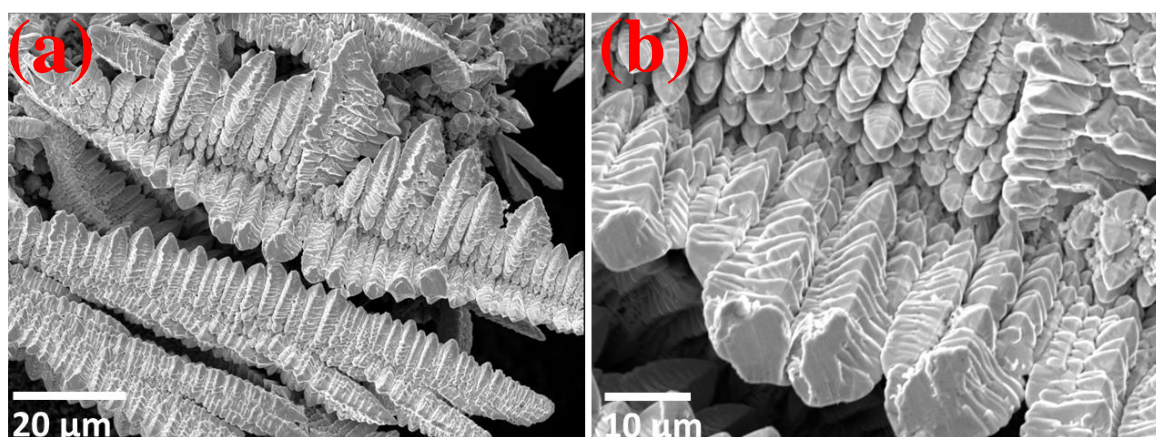


Figure S3. SEM image of Sn_4P_3 on top of Ni-Gr-CNTs matrix where cactus-like structure clearly been observed.

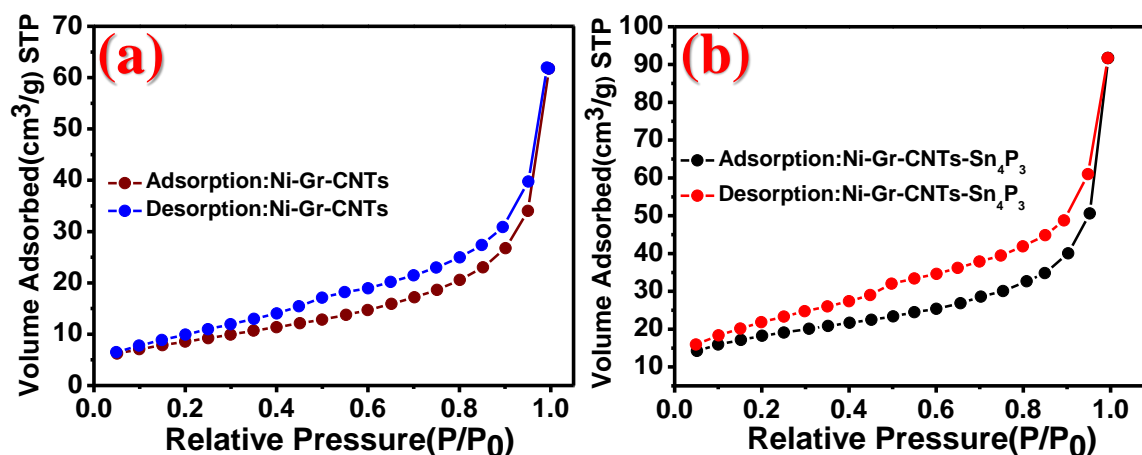


Figure S4. Nitrogen adsorption-desorption isotherm of (a) Ni-Graphene-CNTs and (b) Ni-Graphene-CNTs-Sn₄P₃ sample which indicates significant improvement of the surface area after the decoration of leaf-like Sn₄P₃ on Ni-Gr-CNTs heterostructure.

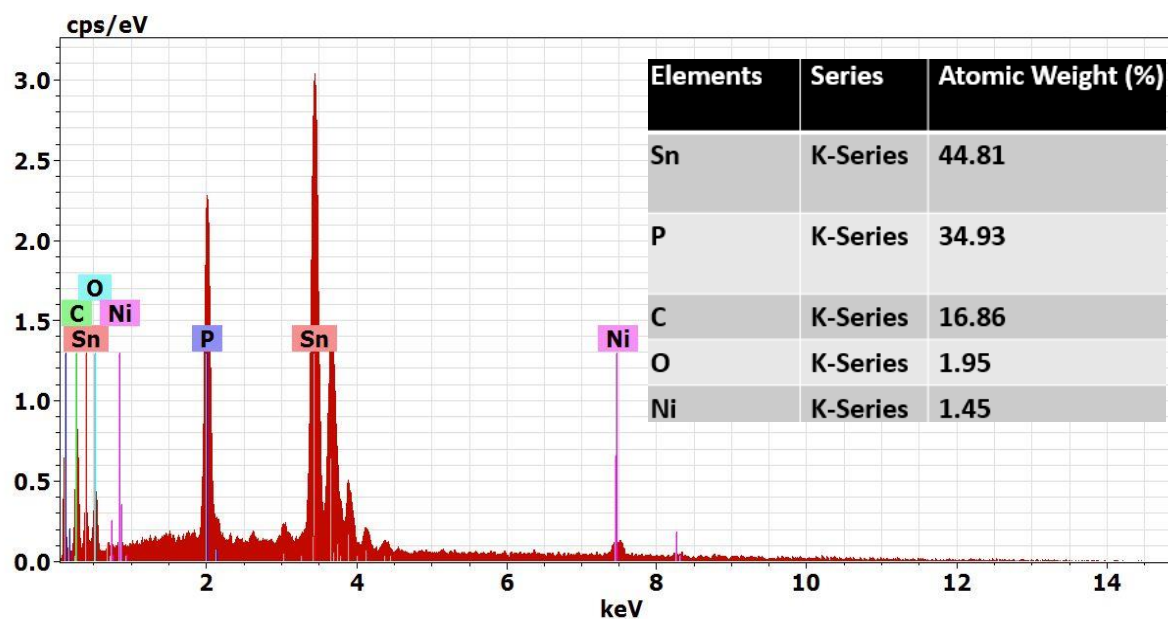


Figure S5. EDX spectrum and elemental percentages (inset) of Ni-Gr-CNTs-Sn₄P₃. The atomic percentage ratio of Sn and P indicates the formation of Sn₄P₃ in the as-grown matrix. A very small percentage of the oxygen may be attributed to the surface oxidation of the as-deposited metal phosphides.

XPS Analysis:

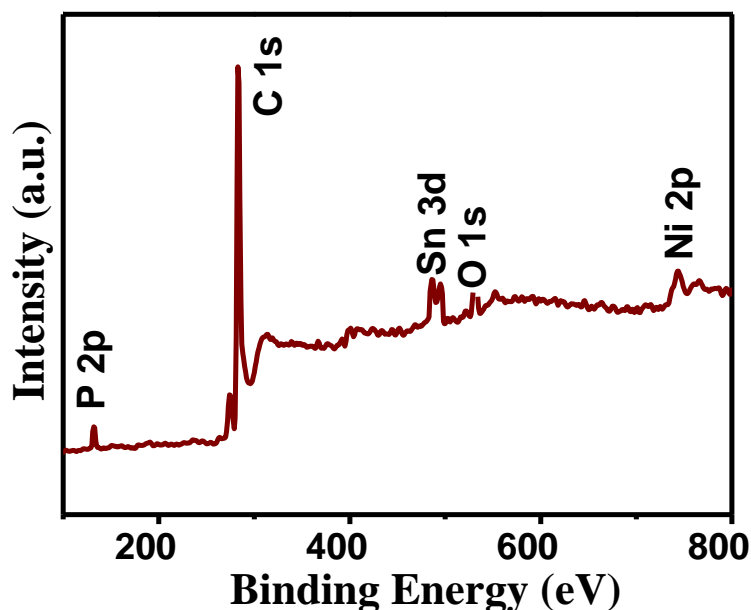


Figure S6. XPS survey spectrum of Ni-Graphene-CNTs-Sn₄P₃ sample refers to the presence of Ni, P, Sn, O and C elements.

In the high-resolution spectra of Ni 2p (Figure S7a) the peak centred at 852.21 and 855.76 eV corresponds to Ni 2p_{3/2} of metallic nickel (Ni⁰) and Ni in the form of oxide (Ni-O).^{1,2} The higher binding energy Ni 2p_{1/2} peak centred at 873.15 eV corresponds to nickel hydroxide (Ni(OH₂)) formation.² These peaks are accompanied by two other satellite peaks of Ni 2p_{1/2} at 861.36 eV and 864.1 eV indicating the presence of Ni in the form of oxide/hydroxide.² In C1s spectra (Figure S7b) the formation of C-C, C-O-C, and O-C=O bonds have been verified by the peaks centred at 284.89, 286.65, 289.10 eV, respectively.^{3,4} The presence of C-O-C, and O-C=O bonds may be related to the induced defect state of the carbonaceous matrix during the growth process (as discussed in the Raman analysis).³ Figure S7c shows the core-level spectra of Sn 3d where the two peaks at 487.28 eV and 495.40 eV can be assigned to Sn 3d_{5/2} and Sn 3d_{3/2} of Sn₄P₃, respectively.⁵⁻⁷ These two peaks have been shifted by approximately 0.7 eV as compared to the reported Sn₄P₃, which indicates that the Sn centre is rich in electron density due to backflow of electron density from carbon matrix to Sn centre in the heterostructure.⁸⁻¹⁰ Further, in high-resolution spectra of P 2p (Figure S7d), peak at 128.51 eV represents the P 2p_{3/2} bonding. Another peak at around 133.40 eV is ascribed to the surface oxidation of Sn₄P₃ and may be attributed to the possible P-O-C/P-O bond formation.^{6,11} The formation of such phosphate bonds not only helps in anchoring the electrolyte on the catalyst surface as well as promotes the proton-coupled electron transfer process (PCET) for OER activity.^{4,12,13} Conclusively, the XPS analysis reveals a strong

electronic interaction between Sn₄P₃ and carbonaceous matrix *via* synergetic effect for boosting the electrochemical performance of the heterostructure.³

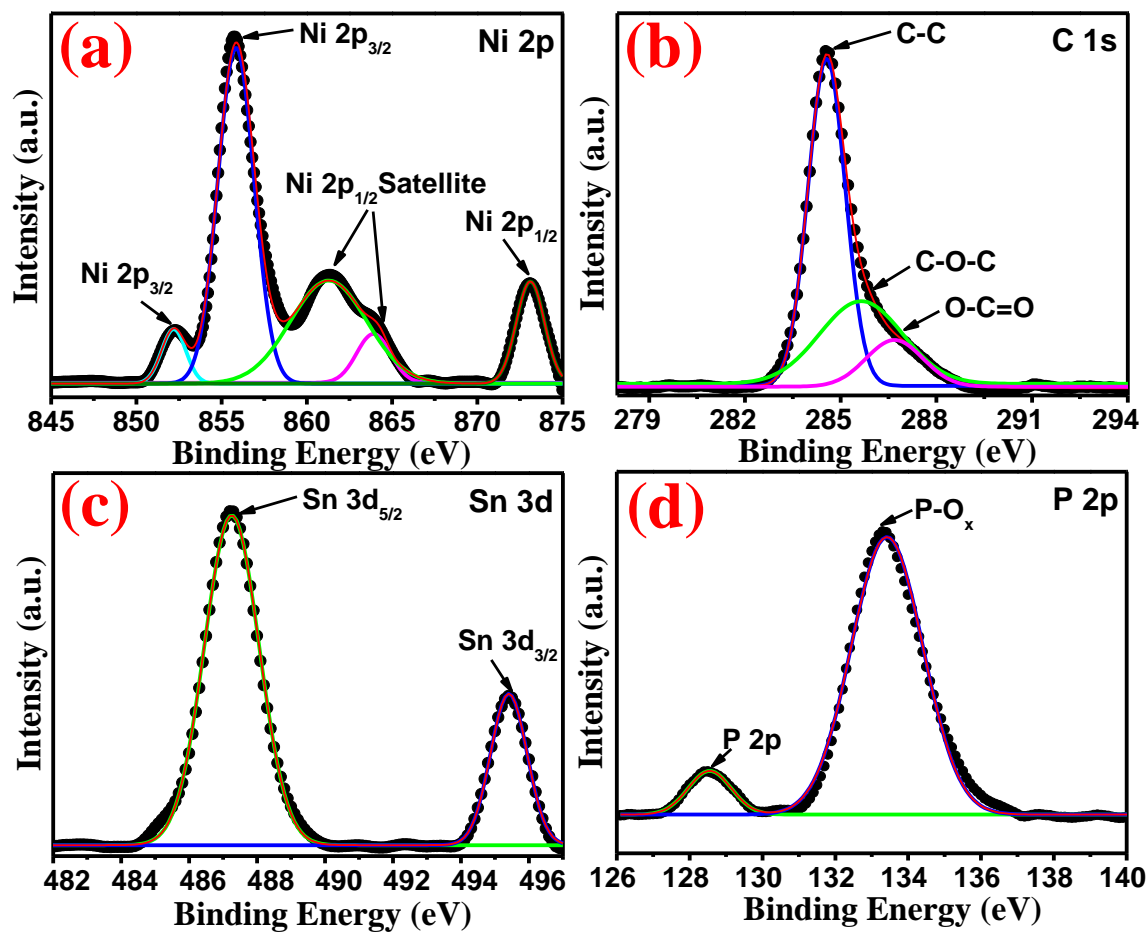


Figure S7. High-resolution XPS spectra of (a) Ni 2p, (b) C 1s, (c) Sn 3d (d) P 2p in Ni-Gr-CNTs-Sn₄P₃ heterostructure.

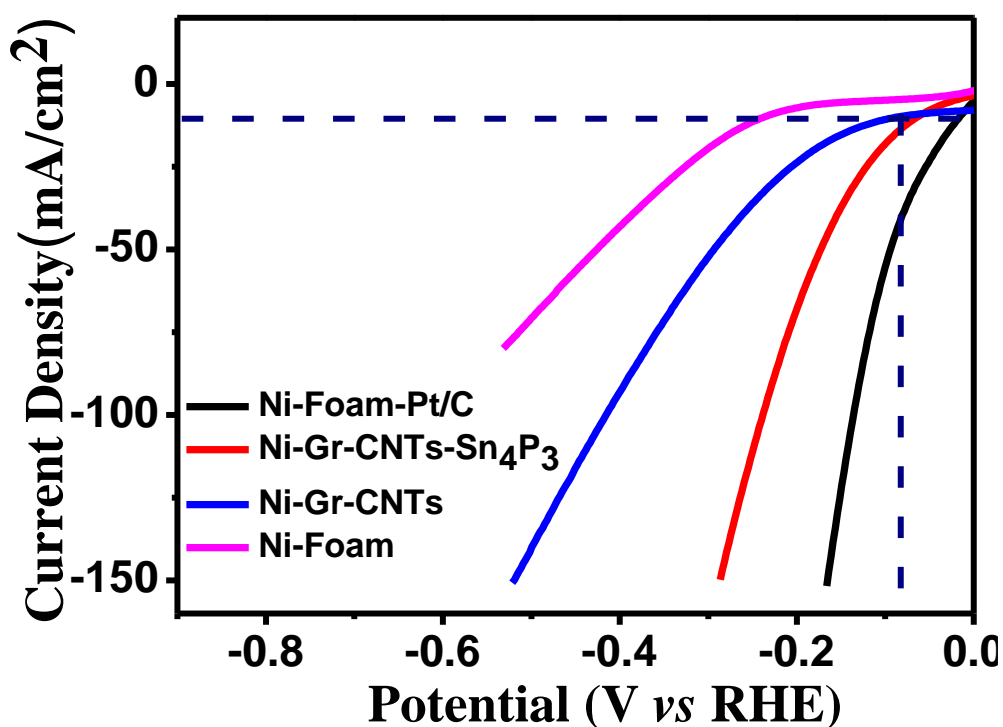


Figure S8. LSV polarization curves of as-designed Ni-Pt/C, Ni-Gr-CNTs-Sn₄P₃, Ni-Gr-CNTs, and Ni-foam electrodes in 0.5M H₂SO₄ electrolyte solution (zoomed-in portion of Figure 4a till 150 mA/cm² current density), which indicates the significant reduction of overpotential after the decoration of Sn₄P₃ on carbon matrix.

Table S1. Comparison of the overpotential values (for HER) at different current densities for all the as-grown samples in 0.5M H₂SO₄ solution.

Samples	Over Potential (mV) @10 mA/cm ²	Over Potential (mV) @50 mA/cm ²	Over Potential (mV) @100 mA/cm ²
Ni-Pt/C	15	94	136
Ni-Gr-CNTs-Sn₄P₃	62	172	239
Ni-Gr-CNTs	96	293	415
Ni-foam	237	425	-

Table S2. The catalytic HER performance of the as-designed Ni-Gr-CNTs-Sn₄P₃ electrode in comparison with the recent development of monometallic phosphides.

S.No.	Catalysts	Medium	Current Density (mA/cm ²)	Overpotential (mV)	Tafel Slope (mV/dec)	Ref
1	CoP/CFP	0.5 M H ₂ SO ₄	10 20 100	128.1 144.4 190.8	49.7	14
2	Co ₂ P@C/CC	0.5 M H ₂ SO ₄	10 100	103 179	40.8	15
3	Ni ₂ P NPs/CC	0.5 M H ₂ SO ₄ 1 M KOH	10 10	69 73	55 73	16
4	FeP/VAGNs/CC	0.5 M H ₂ SO ₄	10	53	42	17
5	CoP/CC	0.5 M H ₂ SO ₄	10 20 100	67 100 204	51	18
6	Cu ₃ P NW/CF	0.5 M H ₂ SO ₄	1 10 100	79 143 276	67	19
7	WP ₂ NS/CC	1.0 M PBS 1.0 M KOH	10 10	261 167	98 81	20
8	Cu ₃ P/NF	1 M KOH	10	115	62	21
9	FeP NAs/CC	0.5 M H ₂ SO ₄	1 10	113 202	45	22
10	Ni ₂ P-G/NF	0.5 M H ₂ SO ₄	5	50	51	23
11	CoP/Ti	0.5 M H ₂ SO ₄	10 100	90 146	43	24
12	MoP ₂ NPs/Mo	0.5 M H ₂ SO ₄ 1.0 M PBS 1 M KOH	10	143 211 194	57 81 80	25
13	Ni ₂ P-G@NF	0.5 M H ₂ SO ₄ 1 M KPi 1 M KOH	10 2 10	105 40 50	38 40 30	26
14	CoP ₃ /NAs	0.5 M H ₂ SO ₄	10 100	65 137	46	27
15	Ni ₂ P/rGO/NF	0.5 M H ₂ SO ₄	10	115	100	28
16	Ni ₁₂ P ₅ NPs	0.5 M H ₂ SO ₄	20	143	63	29
17	Ni ₂ P NPs	0.5 M H ₂ SO ₄	20	130	46	30

18	CoP/Ti foil	0.5 M H ₂ SO ₄	20	85	50	³¹
19	MoP ₂ NS/CC	0.5 M H ₂ SO ₄	10	58	63.6	³²
20	CoP/rGO	0.5 M H ₂ SO ₄	10	105	50	³³
21	CoP/CNT	0.5 M H ₂ SO ₄	2 10	70 122	54	³⁴
22	Mo ₃ P	0.5 M H ₂ SO ₄			92	³⁵
23	Cuf@Ni ₅ P ₄	0.5 M H ₂ SO ₄	10	90	49	³⁶
24	NiP ₂ /CNFs	0.5 M H ₂ SO ₄	10	71	74	³⁷
25	MOF Derived porous Ni ₂ P NS	1 M KOH	10	168	63	³⁸
26	MOF Derived Ni ₂ P polyhedron	0.5 M H ₂ SO ₄	10	310	73	³⁹
27	Ni ₂ P/Ni/NF	1 M KOH	10 20	98 120	72	⁴⁰
28	Ni ₅ P ₄ on Nickel foil	0.5 M H ₂ SO ₄	10	140	40	⁴¹
29	FeP NA/Ti	0.5 M H ₂ SO ₄	10 20 100	55 72 127	38	⁴²
30	Co-PP/Au	0.5 M H ₂ SO ₄	10	183	59	⁴³
31	Ni ₂ P@NSG	0.5 M H ₂ SO ₄	10	110	43	⁴⁴
32	Ni-Gr-CNTs- Sn₄P₃	0.5 M H₂SO₄	10	62	56	This Work

CFP- carbon fiber paper, CC-carbon cloth, NPs-nanoparticles, VAGNs- vertical aligned graphene nanosheets, NWs-nanowire arrays, CF-carbon fiber, NS-nanosheet arrays, NF- nickel foam, NAs- nanorod arrays, G/NF- 3D graphene-nickel foam, Cuf-copper foam

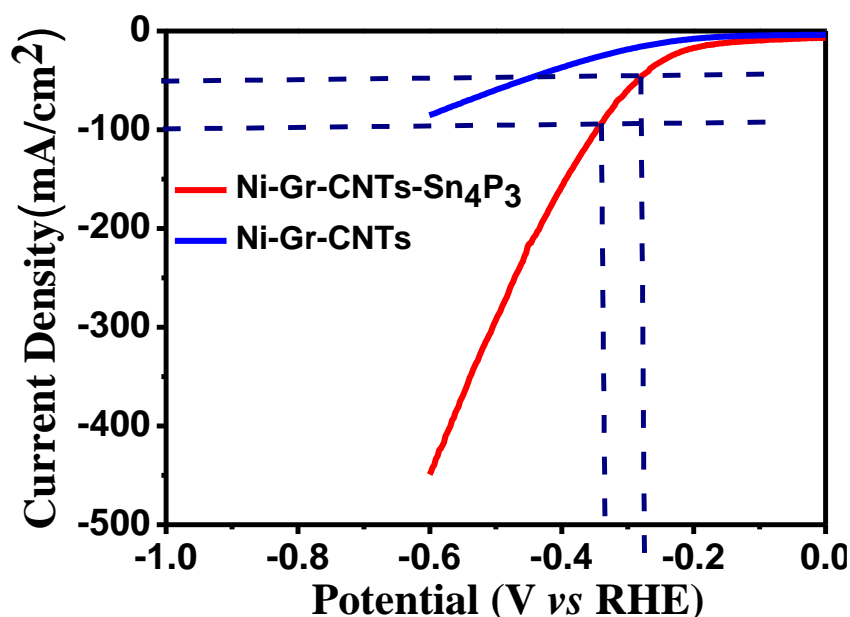


Figure S9. LSV polarization curves corresponding to Ni-Gr-CNTs-Sn₄P₃ and Ni-Gr-CNTs electrodes in 1M KOH solution.

Tafel plot:

To calculate the value of the Tafel slope, the linear region of the Tafel plot is fitted using the Tafel formula: $\eta = b \log(J) + a$, where η refers to overpotential, J refers to current density, b refers to the Tafel slope and a is constant. $b = 2.3RT/\alpha F$ (R -gas constant, T - absolute temperature, α -symmetry coefficient, F -faraday constant). Faster charge transfer in the electrochemical process can be predicted by the lower value of the Tafel slope (b).³

Hydrogen evolution reaction mechanism:

There are two different reaction mechanisms involved in the Hydrogen evolution reaction under acid and alkaline medium proceed *via* three steps. One of the mechanisms is Volmer-Heyrovsky and another is Volmer-Tafel, in which the primary discharge step *i.e.*, Volmer is common in both the mechanism.³

Under acidic medium:

(1) $H_3O^+ + e^- + A \rightleftharpoons A-H_{ads} + H_2O$; Volmer reaction (discharge Step).

Here, electrons are transferred to the surface of the cathode and simultaneously hydronium ions (H_3O^+) are coming from the electrolyte to form $A-H_{ads}$ intermediate state (A denotes the active site of the catalyst). Here, $b \sim 120$ mV.

(2) $A-H_{ads} + H_3O^+ + e^- \rightleftharpoons H_2 + A + H_2O$; Heyrovsky reaction (desorption step). In this case, when the density of H_{ads} on the active catalyst surface is low then H_{ads} intermediates combine with the new electron and hydronium ions and produced H_2 molecule; $b \sim 40$ mV.

(3) $A-H_{ads} + H_{ads} - A \rightleftharpoons H_2 + 2A$; Tafel reaction (Recombination Step). Whenever the density of H_{ads} on the active catalyst surface is high then the two adjacent H_{ads} intermediates combine and generate H_2 molecule. $b \sim 30$ mV.

Under alkaline medium:

- (1) $H_2O + e^- + A \rightleftharpoons A-H_{ads} + OH^-$; Volmer-discharge Step
- (2) $H_2O + e^- + A-H_{ads} \rightleftharpoons A + H_2 + OH^-$; Heyrovsky-desorption step
- (3) $A-H_{ads} + H_{ads} - A \rightleftharpoons H_2 + 2A$; Tafel-recombination Step

The slowest step i.e., the rate-determining step is dependent on the affinity between intermediates and the active catalyst surface. The reaction will be dominated by the Volmer step if the bonding between the active catalyst site and hydrogen intermediate is weak. However, if the bonding strength between the active catalyst site and hydrogen intermediates is very strong then the Heyrovsky or Tafel would be the rate-limiting dominating step.^{3,45} To facilitate the adsorption and desorption process for the efficient production of H_2 molecule, it is necessary to get moderate bonding between the active catalyst site and hydrogen intermediate.⁴⁵

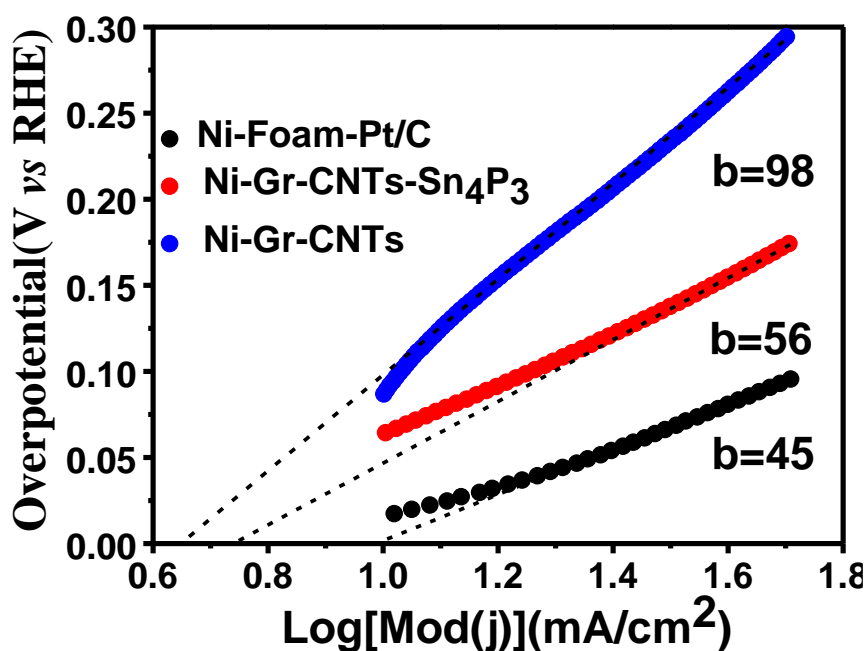


Figure S10. Calculated exchange current density from the extrapolation of Tafel slope at an overpotential of 0 V has been found to be 10, 5.6, and 4.4 mA/cm² for Ni-Pt/C, Ni-Gr-CNTs-Sn₄P₃ and Ni-Gr-CNTs electrodes, respectively. Higher the value of exchange current density leads to faster rate of HER kinetics.

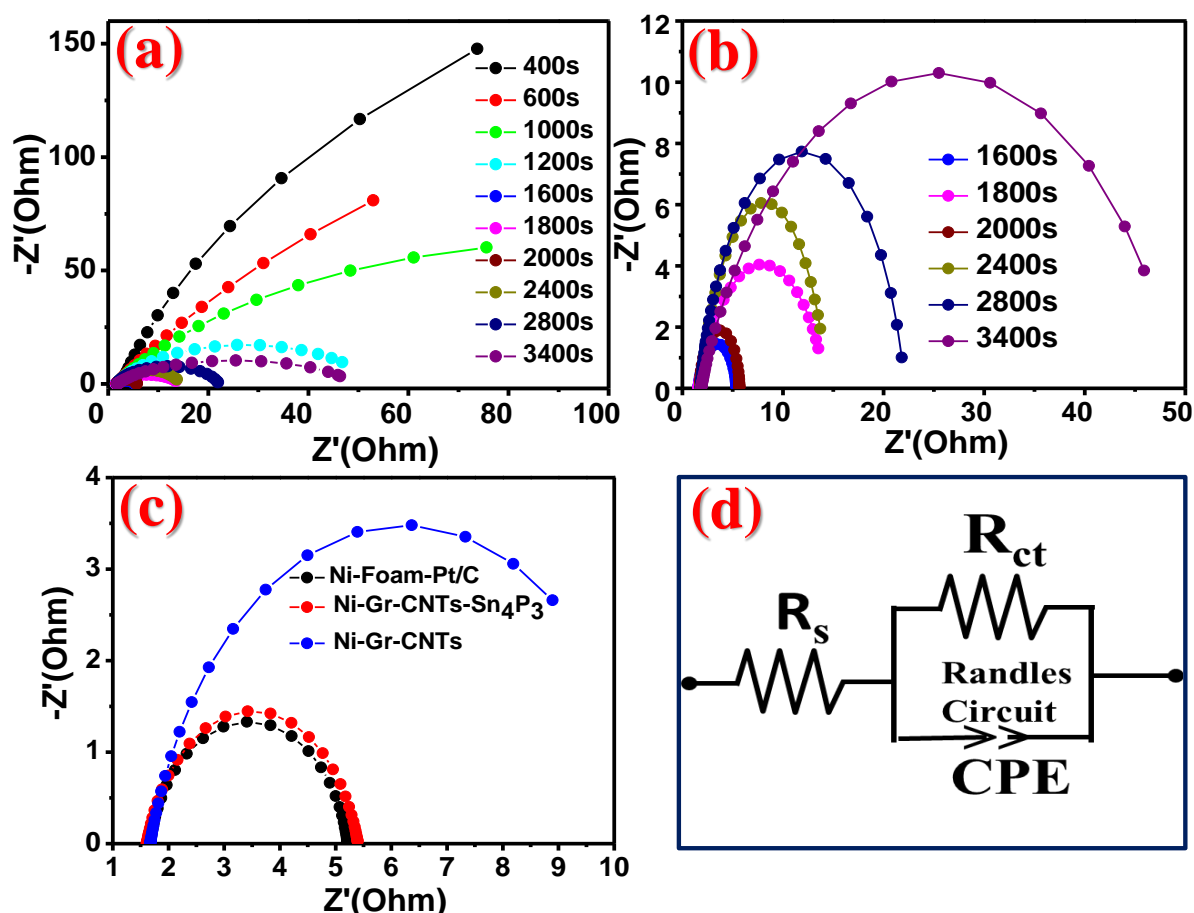


Figure S11. Electrochemical impedance spectra (EIS) analysis. Nyquist plots of (a) Ni-Gr-CNTs-Sn₄P₃ with different deposition times (400-3400s) of Sn (b) shows the zoomed-in portion for the deposition time of 1600, 1800, 2000, 2400, 2800 and 3400s. The lowest R_s and R_{ct} values have been found for the Sn-1600s deposition sample as compared to the others which lead to the best electrocatalytic activity; (c) Ni-Pt/C, Ni-Gr-CNTs-Sn₄P₃ and Ni-Gr-CNTs. (d) shows the equivalent circuit diagram to fit all the EIS data. The EIS is tested in a three-electrode system over a frequency range from 100 kHz to 0.01 Hz with an AC amplitude of 10 mV at a constant voltage of -300 mV (*vs* RHE) in 0.5M H₂SO₄ solution. The R_s and R_{ct} values of the Ni-Gr-CNTs-Sn₄P₃ and Ni-Pt/C are very close to each other indicating a high charge transfer process.

Table S3. Comparison of series (R_s) and charge transfer resistance (R_{ct}) of Ni-Gr-CNTs-Sn₄P₃ with the different deposition times of Sn where the Sn-1600s shows the lowest resistance values.

Ni-Gr-CNTs-Sn ₄ P ₃ with different Sn deposition time	Series Resistance (R_s) Ohm	Charge transfer Resistance (R_{ct}) Ohm
400 s	2.16	480
600 s	1.77	356
1000 s	1.67	196
1200 s	1.61	12.7
1600 s	1.55	3.79
1800 s	1.62	4.92

2000 s	1.78	3.91
2400 s	1.80	9.3
2800 s	1.82	16.3
3400 s	1.87	33.2

Table S4. Comparison of series (R_s) and charge transfer resistance (R_{ct}) of the as-prepared samples in 0.5M H_2SO_4 solution.

Samples	Series Resistance (R_s) Ohm	Charge transfer Resistance (R_{ct}) Ohm
Ni-Foam-Pt/C	1.61	3.51
Ni-Gr-CNTs-Sn ₄ P ₃	1.55	3.79
Ni-Gr-CNTs	1.67	11.12

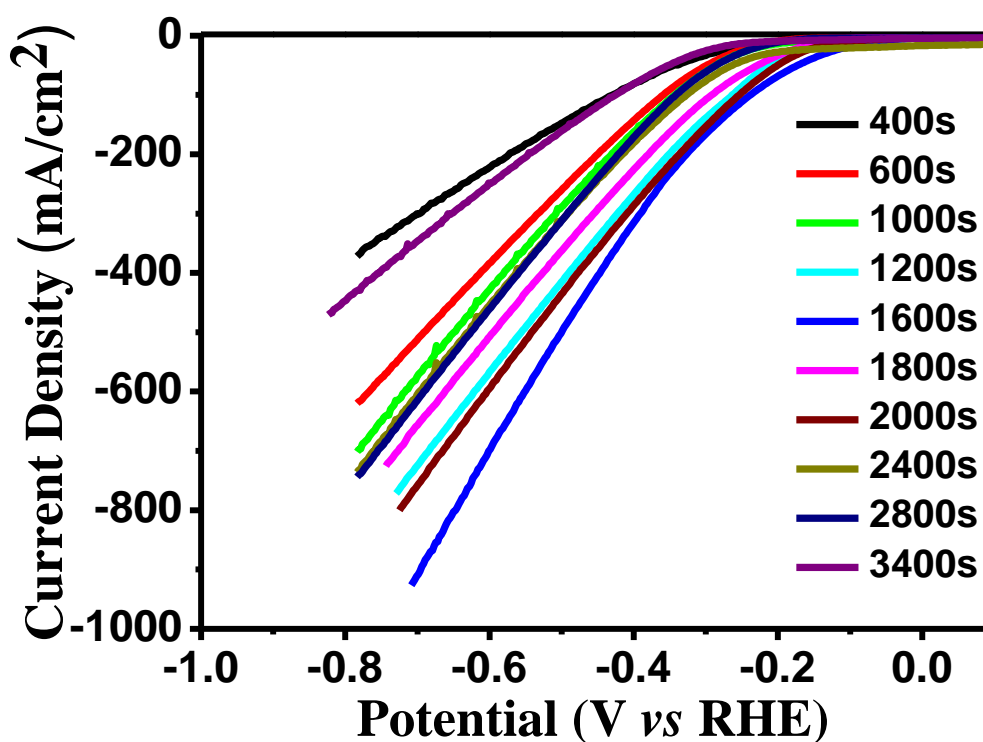


Figure S12. HER LSV polarization curves of Ni-Gr-CNTs-Sn₄P₃ at different deposition times of Sn in 0.5M H_2SO_4 . The best performance has been achieved from the sample where the deposition of Sn is 1600s.

Table S5. Summary of overpotential at a different current density of HER (0.5M H₂SO₄) for Ni-Gr-CNTs-Sn₄P₃ with the different deposition time of Sn.

Sn deposition time	Over Potential (mV) @10 mA/cm ²	Over Potential (mV) @50 mA/cm ²	Over Potential (mV) @100 mA/cm ²
400 s	176	342	428
600 s	203	298	359
1000 s	169	286	343
1200 s	90	213	263
1600 s	62	172	239
1800 s	137	227	291
2000 s	103	201	251
2400 s	164	262	327
2800 s	193	285	340
3400 s	209	350	424

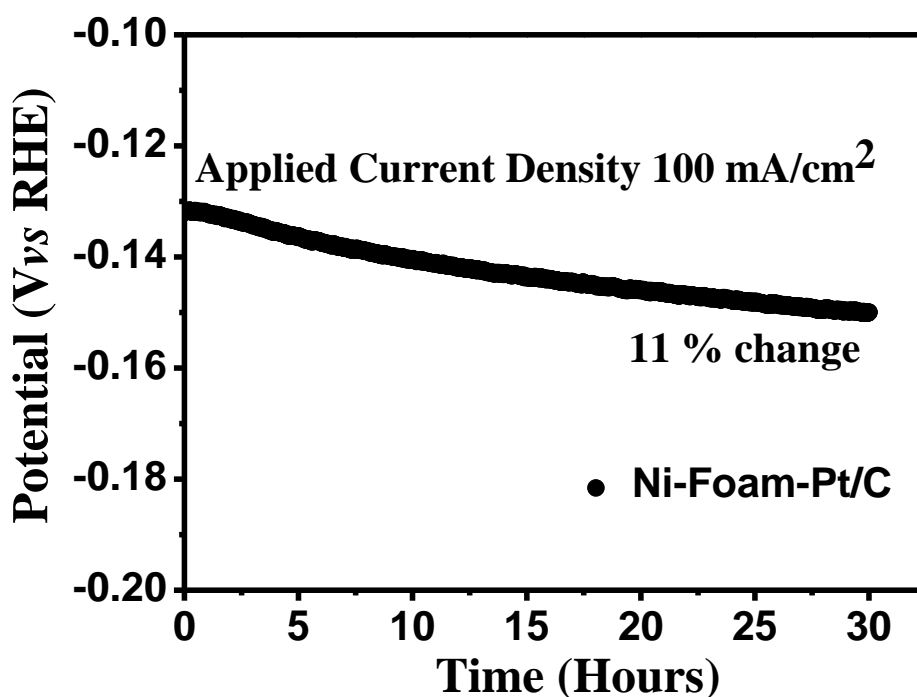


Figure S13. Chronopotentiometric analysis of the stability test at least for 30 hours with ~11% change of potential for Ni-foam-Pt/C at constant applied current density of 100 mA/cm².

Electrochemically Active Surface Area (ECSA) and Roughness factor (RF) calculation

To estimate ECSA, the double-layer capacitance (C_{dl}) is calculated by carrying out cyclic voltammetry (CV) curves in the non-faradaic region at a different scan rate of 10 to 100 mV/s in 0.5M H_2SO_4 and 1M KOH solution. The capacitive current density (ΔJ) can be determined by the product of the scan rate (v) and the double-layer capacitance⁴⁶

$$\Delta J = v C_{dl}$$

The scan rates are fitted with the capacitive current density differences ($\Delta J = |j_a - j_c|$) and from the slope, the C_{dl} value has been estimated. The C_{dl} values of Ni-Gr-CNTs-Sn₄P₃ is found to be 21 mF/cm² and 19.5 mF/cm² for HER and OER, respectively.

The ECSA can be calculated by dividing C_{dl} with the specific capacitance (C_{sp}) of the surface of the electrode or the capacitance of an atomically smooth planar surface of the material per unit area under identical electrolyte condition⁴⁶

$$ECSA = C_{dl} / \{C_{sp} \text{ per ECSA cm}^2\}$$

The commonly used C_{sp} value for a flat metal surface is varying between 0.02-0.06 mF/cm² and here 0.04 mF/cm² value has been consider but it should be noted that the C_{sp} value of such kind of Sn₄P₃ along with graphene-CNTs is not available, and thus the choice of 0.04 mF/cm² may create some errors.^{2,3,11,47} The ECSA value of Ni-Gr-CNTs-Sn₄P₃ is calculated to be 525 and 487 cm² for HER and OER, respectively.

The roughness factor (RF) is also calculated by using the following formula.⁴⁸

$$RF = \text{Catalyst active surface area (ECSA)} / \text{Geometric surface area}$$

The geometric surface area of the electrode is considered to be (1x1.2) cm² shown in Figure S28. The RF value of Ni-Gr-CNTs-Sn₄P₃ is calculated to be 437 and 406 for HER and OER, respectively.

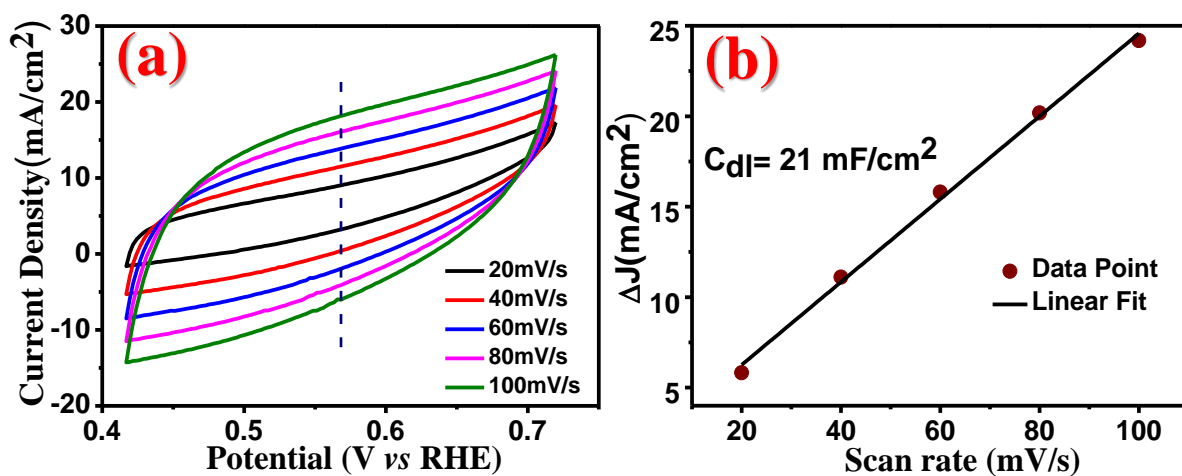


Figure S14. (a) Cyclic voltammograms of Ni-Gr-CNTs-Sn₄P₃ in the non-faradic region of 0.41 to 0.71 V (vs RHE) at different scan rates ranging from 20 mV/s to 100 mV/s with an interval of 20 mV/s in 0.5M H₂SO₄ and corresponding (b) capacitive currents density difference (ΔJ) as a function of scan rate has been plotted at 0.56 V (vs RHE) and the slope (C_{dl}) calculated from the linear fitting has been found to be 21 mF/cm². The estimated values of the C_{dl} indicate the significantly higher electro-active surface area of Ni-Gr-CNTs-Sn₄P₃.

Active site density (n) and Turn-over frequency (TOF) calculations

The active site density (n) is defined as the following formula^{49,50}:

$n = Q/2F$ in HER process.

$n = Q/4F$ in OER process.

Where Q is the voltammetric charge and F is the Faraday constant (F) ~ 96485 C/mol

The CV analysis has been carried out within a potential window of -0.2 V to 0.8 V (vs RHE) with a scan rate of 50 mV/s in 1M Phosphate buffer Saline shown in Figure S13 for the calculation of the voltammetric charge. The active site density of Ni-Gr-CNTs and Ni-Gr-CNTs-Sn₄P₃ are found to be 2.27×10^{-4} mol/cm² and 5.33×10^{-4} mol/cm² for HER and 1.13×10^{-4} mol/cm² and 2.66×10^{-4} mol/cm² for OER, respectively. Here, we consider all the active surface sites include Ni, Sn, P, C atom involved in the HER and OER activity.

Turn-over frequency (TOF) has been calculated using these active site densities (n) and it is given as^{49,50}:

$TOF = I / 2nF$ for HER process

$TOF = I / 4nF$ for OER process

where, I is the current density measured by the LSV plots in HER and OER processes.

The TOF of Ni-Gr-CNTs are 1s⁻¹, 2s⁻¹ at overpotentials of 301 mV and 427 mV and 264 mV and 362 mV for Ni-Gr-CNTs-Sn₄P₃ respectively in case of HER. Similarly, for OER the TOF of Ni-Gr-CNTs is calculated to be 0.5s⁻¹@340 mV, 1s⁻¹@400 mV but in the case of Ni-Gr-CNTs-Sn₄P₃, the TOF is achieved to be 0.5s⁻¹@310 mV, 1s⁻¹@410 mV.

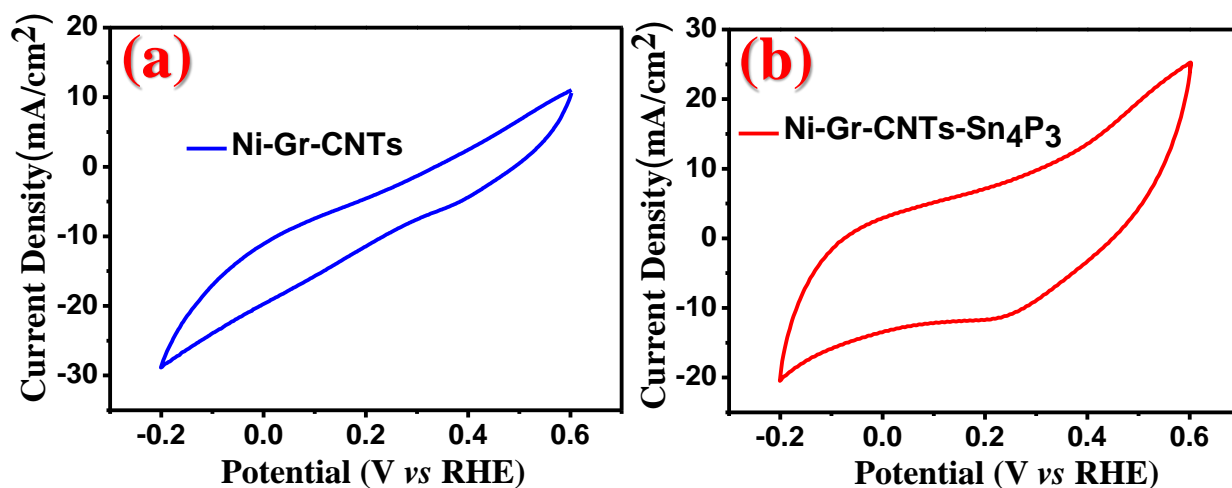


Figure S15. Cyclic voltammograms of (a) Ni-Gr-CNTs and (b) Ni-Gr-CNTs-Sn₄P₃ in 1M PBS (pH ~ 7) solution under the potential range between -0.2 V (vs RHE) to 0.6 V (vs RHE) at a scan rate of 50 mV/s helps to calculate the active site density and TOF.

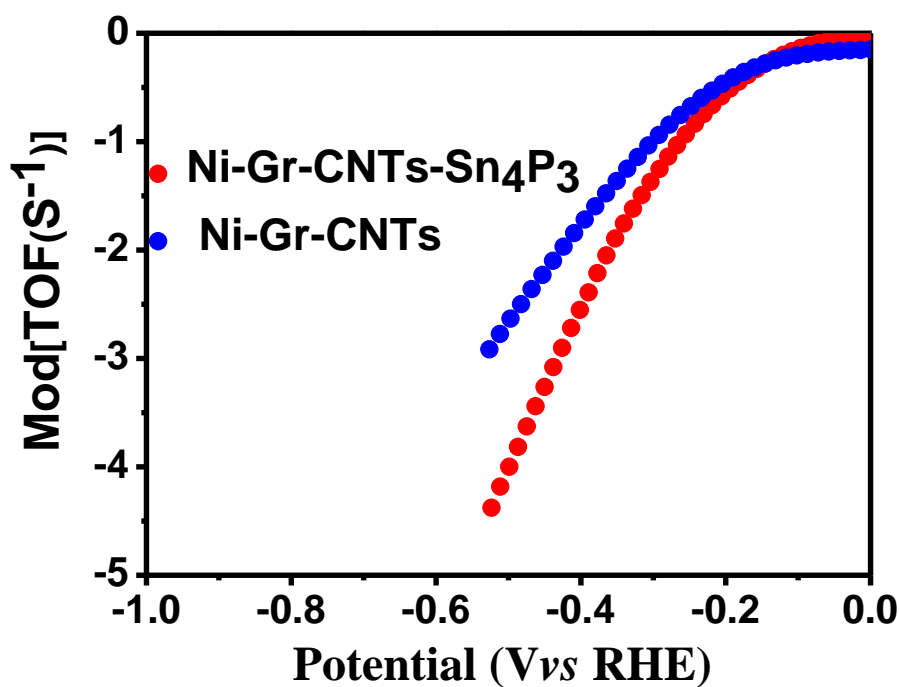


Figure S16. The TOF values have been calculated for Ni-Gr-CNTs-Sn₄P₃ (1s⁻¹@264 mV, 2s⁻¹@362 mV) and Ni-Gr-CNTs (1s⁻¹@301 mV, 2s⁻¹@ 427 mV); where the high TOF achieved at low applied potential indicates the good electrocatalytic activity of Ni-Gr-CNTs-Sn₄P₃ towards water splitting.

Table S6. OER catalytic performance of Ni-Gr-CNTs-Sn₄P₃ in comparison with the recently developed monometallic phosphide-based catalysts.

S.No	Catalysts	Medium	Current Density (mA/cm ²)	Overpotential (mV)	Tafel Slope (mV/dec)	Ref
1	Co ₂ P NN	1 M KOH	10	310	50	51
2	CoP NR	1 M KOH	10	340	71	52
3	CoP/C	1 M KOH	10	360	66	53
4	Ni ₂ P NWs	1 M KOH	10	400	60	54
5	Ni ₂ P NPs	1 M KOH	10	500	70	55
6	Co-P foam	1 M KOH	10	131	74	56
7	Ni ₅ P ₄ -NF	1 M KOH	10	290	40	41
8	Ni-P/CP	1 M KOH	10	190	73	57
9	CoP NR/NF	1 M KOH	10	390	65	58
10	Co-P film	1 M KOH	10	345	47	59
11	CoP NWs	1 M KOH	10	248	78	60
12	Co-P/NC	1 M KOH	10	354	52	61
13	Co ₂ P/CoNPC	1 M KOH	10	328	78	62
14	Co ₂ P NCs	1 M KOH	10	280	60.4	63
15	Ni ₁₂ P ₅ /NF	1M KOH	10	240	225	64
16	NiP ₂ /CC	1 M KOH	20	310	107.8	65
17	MOF-Ni ₂ P NS	1 M KOH	10	320	105	38
18	Co-PP/Au	1 M KOH	10	340	-	43
19	Ni ₁₂ P ₅ - O- WCNT	1 M KOH	10	280	62	66
20	Au/Ni ₁₂ P ₅ NPs	1 M KOH	10	340	49	67
21	Ni ₂ P@NSG	1 M KOH	10	240	43	44
22	Ni ₂ P	1 M KOH	10	280	115	68
23	FeP@Au	1 M KOH	10	320	56.8	69
24	Ni ₂ P@C/G	1 M KOH	10	285	44	70
25	Multishelled Ni ₂ P	1 M KOH	10	270	40.4	71
26	CoP-CNT	1M NaOH	10	330	50	72
27	C@Ni ₈ P ₃	1 M KOH	10	144	51	73
28	CoP ₃ NAs/CFP	1 M KOH	10	334	62	27
29	CoP ₂ /RGO	1 M KOH	10	300	96	74
30	Cuf@Ni ₅ P ₄	1 M KOH	20	210	127	36
31	Ni-Gr-CNTs- Sn ₄ P ₃	1 MKOH	20	169	88	This wor

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NN-nanoneedles, NR-nanoroad, C-carbon support, NWs-nanowire arrays, NPs-nanoparticles, CFP/CP- carbon fiber paper, NC-nanopolyhedrons composed, CoNPC- Co, N, P doped carbon, NCs- nanocrystals, NF- nickel foam, CC-Carbon Cloth, NS-nanosheet, Co-PP- cobalt phosphide phosphate films, O-WCNT-oxidized multiwall carbon nanotube, NSG- N-and S-doped graphene, C/G-carbon confined graphene, NAs- nanorod arrays

Table S7. The OER overpotential values at different current densities for all as-prepared samples in 1 M KOH solution.

Samples	Over Potential (mV) @20 mA/cm ²	Over Potential (mV) @50 mA/cm ²	Over Potential (mV) @100 mA/cm ²
Ni-Gr-CNTs-Sn ₄ P ₃	169	289	380
Ni-foam-RuO ₂	290	360	450
Ni-Gr-CNTs	320	390	-
Ni-foam	430	-	-

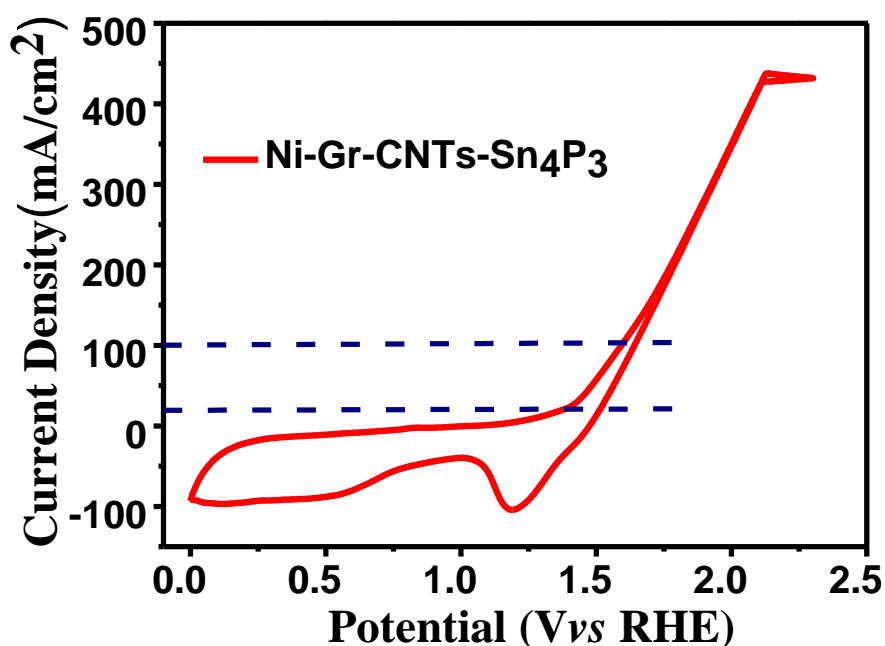


Figure S17. CV analysis of Ni-Gr-CNTs-Sn₄P₃ with a scan rate of 2 mV/s in the potential range of 0 to 2.25 V (vs RHE). It shows a very small oxidation peak at 1.45 V (vs RHE) and a prominent reduction peak at 1.19 V (vs RHE).

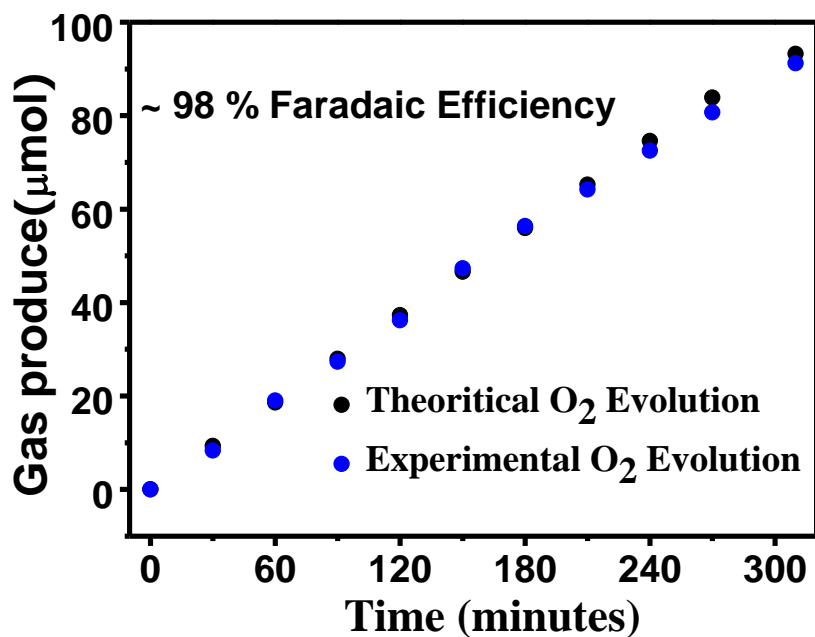


Figure S18. The experimentally and theoretically estimated amount of O₂ produced by the Ni-Gr-CNTs-Sn₄P₃ electrode at a fixed potential of 1.45 V in 1M KOH solution. A very small deviation in experimental and theoretical result indicates ~ 98% faradaic efficiency of O₂ production.

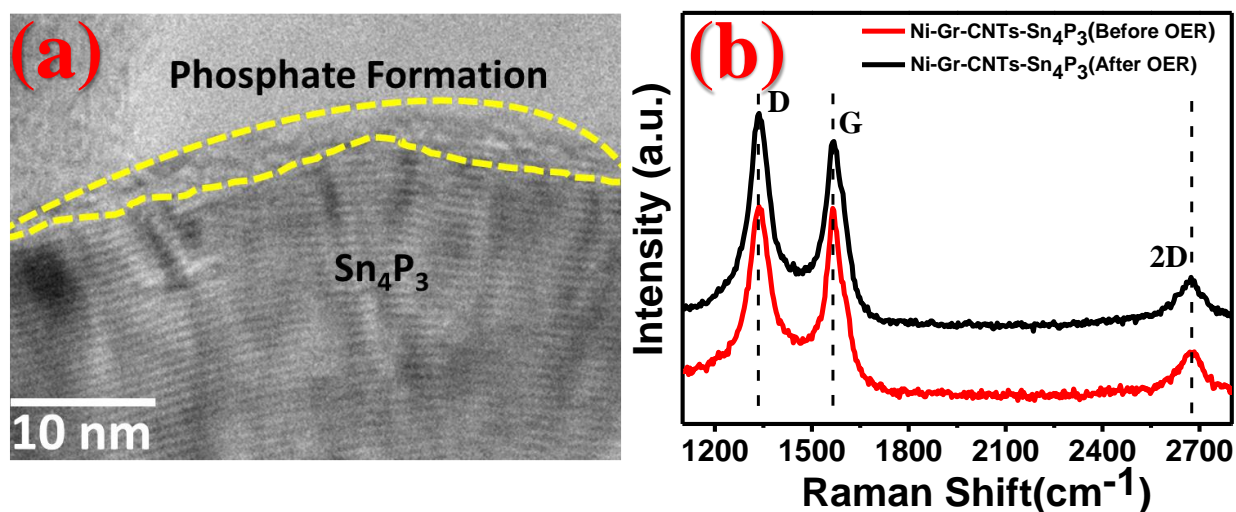


Figure S19. Post OER analysis of Ni-Gr-CNTs-Sn₄P₃ (a) HR-TEM and (b) Raman analysis indicates that surface oxidation of Sn₄P₃ doesn't impact on template carbon matrix in the Raman response.

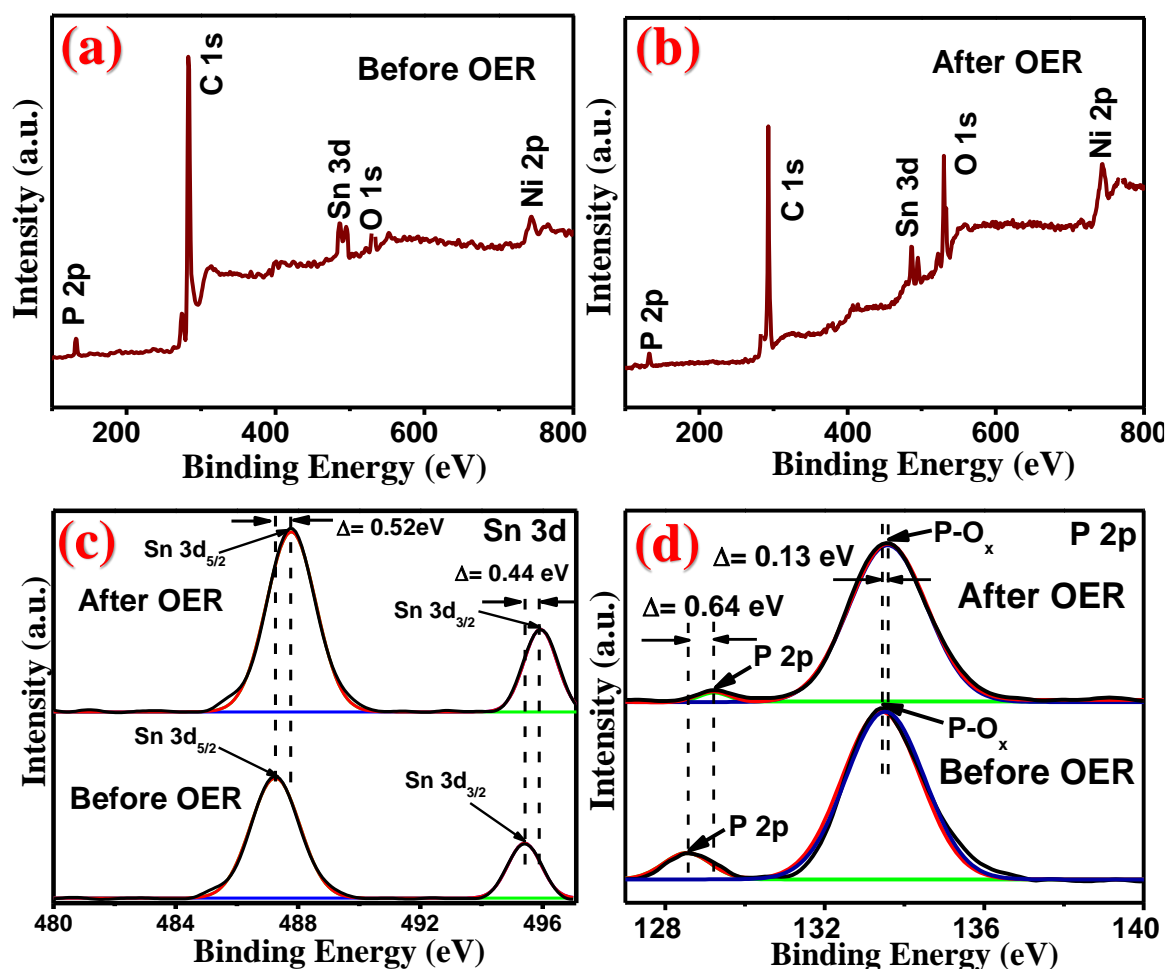


Figure S20. XPS survey spectrum of (a) before and (b) post OER process of Ni-Gr-CNTs-Sn₄P₃. The corresponding XPS high-resolution spectrum of (c) Sn 3d (d) P 2p under before and after OER process indicate that surface oxidation has a prominent impact on XPS response.

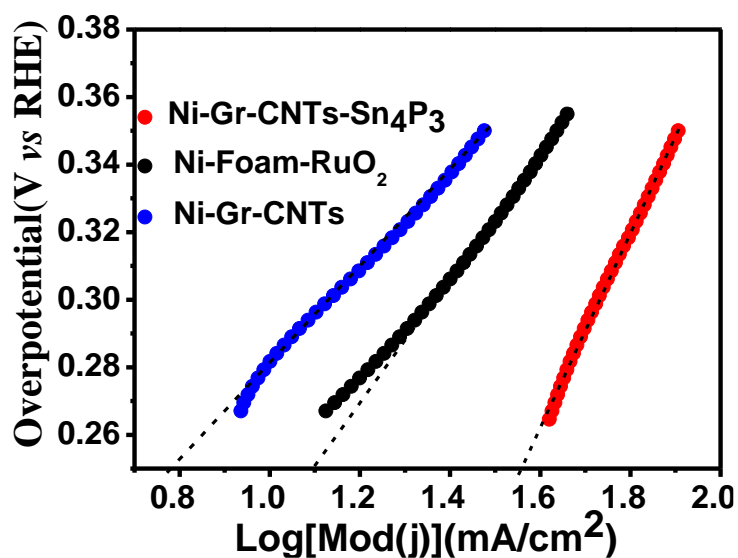


Figure S21. The exchange current density is calculated from the extrapolation of Tafel slope at the overpotential of 0 V. The exchange current density of Ni-Gr-CNTs-Sn₄P₃, Ni-RuO₂, and Ni-Gr-CNTs are calculated to be 4.7, 2.9, and 2.1 mA/cm², respectively.

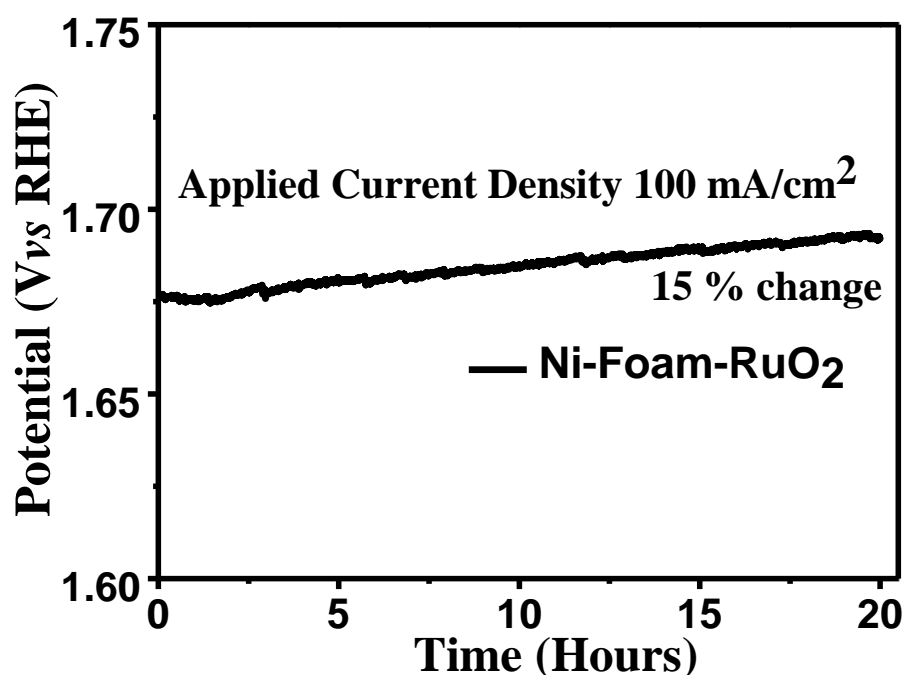


Figure S22. Chronopotentiometric analysis of the stability test at least for 20 hours with ~15% change of potential for Ni-foam-RuO₂ at constant applied current density of 100 mA/cm².

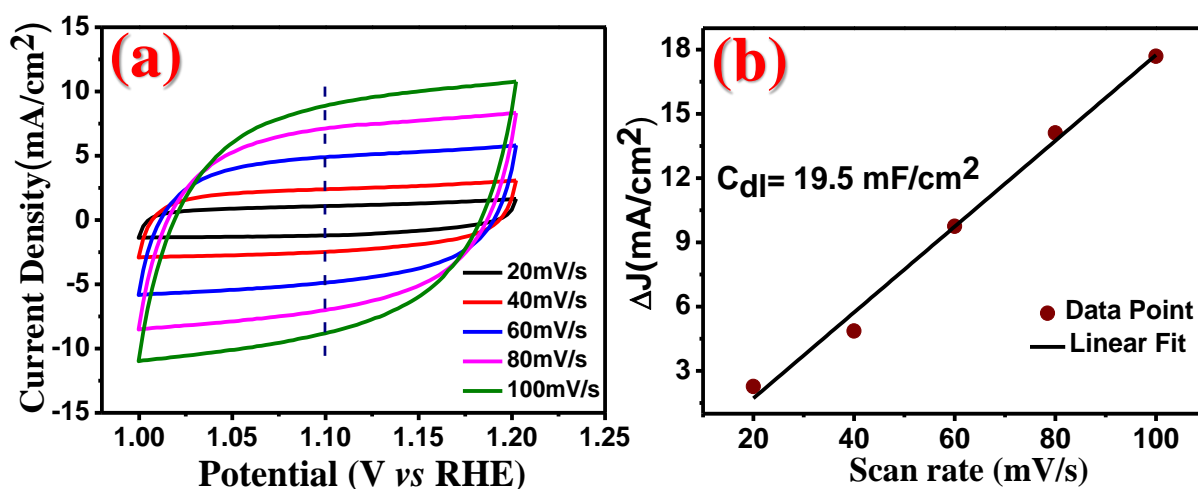


Figure S23. (a) Cyclic voltammograms of Ni-Gr-CNTs-Sn₄P₃ in the non-faradic region of 1 V to 1.2 V (vs RHE) at different scan rates ranging from 20 mV/s to 100 mV/s with an interval of 20 mV/s in 1M KOH and corresponding (b) capacitive currents density difference (ΔJ) as a function of scan rate has been plotted at 1.10 V (vs RHE) and from the slope, C_{dl} has been calculated to be 19.5 mF/cm². The estimated value of the C_{dl} indicates the significantly higher electroactive surface area of Ni-Gr-CNTs-Sn₄P₃ in the OER process.

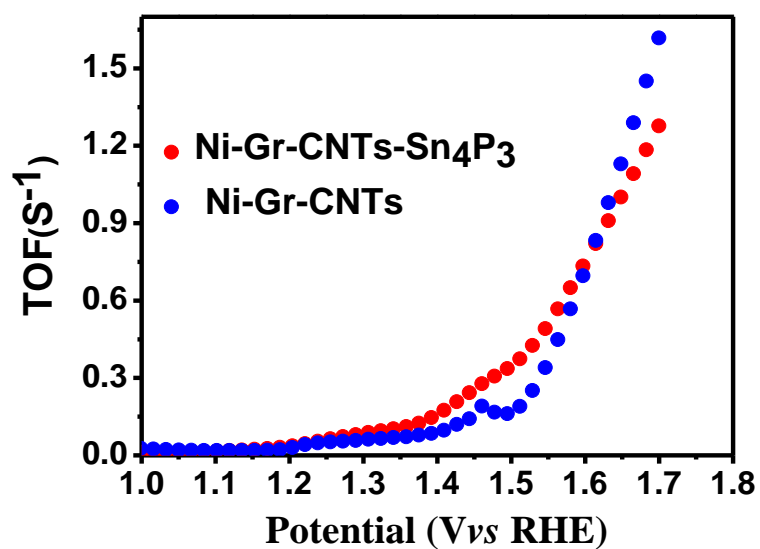


Figure S24. The TOF values of the Ni-Gr-CNTs-Sn₄P₃ (0.5s⁻¹@310 mV and 1s⁻¹@410 mV) and Ni-Gr-CNTs (0.5s⁻¹@340 mV and 1s⁻¹@400 mV) with varying OER potentials; where high TOF achieved at low applied potential indicates the good electrocatalytic activity of Ni-Gr-CNTs-Sn₄P₃ towards water splitting.

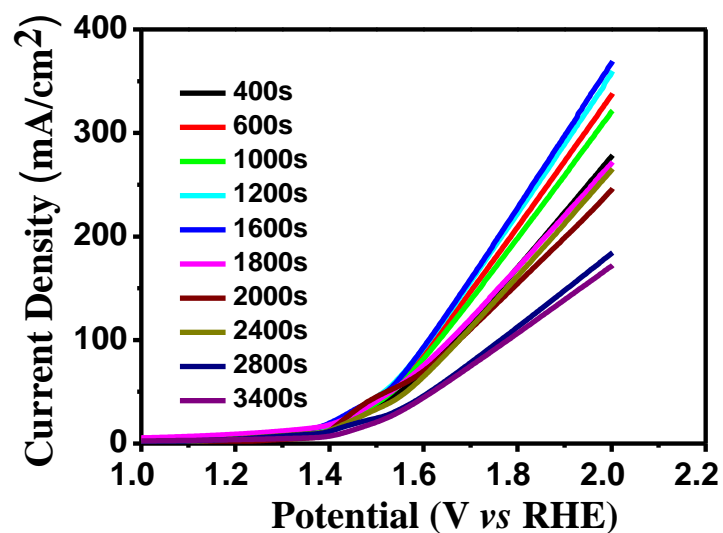


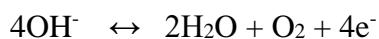
Figure S25. OER LSV polarization curves of Ni-Gr-CNTs-Sn₄P₃ at different deposition times of Sn in 1M KOH.

Table S8. Summary of overpotential values at different current densities for OER (1M KOH) in Ni-Gr-CNTs-Sn₄P₃ with the different deposition times of Sn.

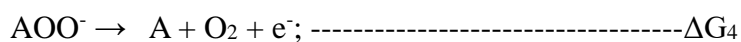
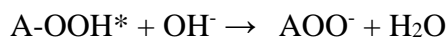
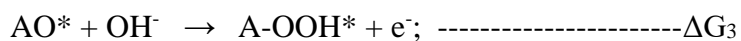
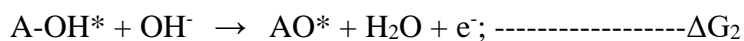
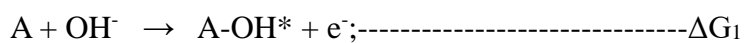
Sn deposition time	Over Potential (mV) @20 mA/cm ²	Over Potential (mV) @50 mA/cm ²	Over Potential (mV) @100 mA/cm ²
400 s	187	320	430
600 s	200	290	390
1000 s	186	300	400
1200 s	178	280	379
1600 s	169	289	375
1800 s	183	299	420
2000 s	193	296	440
2400 s	201	333	446
2800 s	230	382	534
3400 s	265	390	550

Oxygen evolution reaction mechanism:

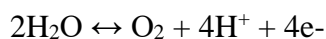
In alkaline medium, OER involved



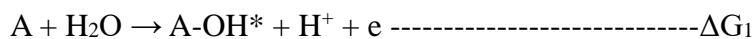
The above reaction is followed by four electrons transfer process in several steps.

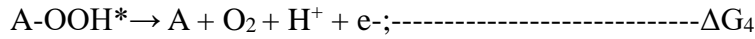
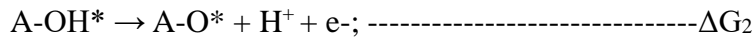


In acid medium, OER is given by:



The above reaction proceeds *via* several steps as follows:





where A denotes the active metal site on the surface

In acid or alkaline medium, OER takes place *via* generating OH*, O* and OOH* intermediate.

The adsorption energies of the intermediates at different active sites on the catalyst can be determined as:^{75,76}

$$\Delta E_{OH^*} = E_{OH^*} - E^* - (E_{H_2O} - 0.5E_{H_2})$$

$$\Delta E_{O^*} = E_{O^*} - E^* - (E_{H_2O} - E_{H_2})$$

$$\Delta E_{OOH^*} = E_{OOH^*} - E^* - (2E_{H_2O} - 1.5E_{H_2})$$

where E* is the energy of the catalytic surface. The adsorption energies in reference to the energy values of H₂O and H₂ molecules in the gas phase i.e., E_{H₂O} and E_{H₂}, have been calculated from DFT.

Further, the free-energies are calculated as^{75,76}

$$\Delta G = \Delta E + \Delta ZPE - T\Delta S - \Delta G_U + \Delta G_{pH}$$

Here, the small values of zero-point energy (ΔZPE) and entropy change (ΔS), the term ΔZPE and $T\Delta S$ are neglected and considered to be zero for the simplicity of the calculation. ΔG_U represents the applied electrode potential, that is $\Delta G_U = eU$ and $\Delta G_{pH} = K T \ln a_{H^+}$ where K as the Boltzmann constant, T is the temperature and a_{H^+} is the pH of the proton. The Gibbs free-energy change for each intermediate reaction is determined at standard conditions (pH= 0, T=298.15 K) as^{3,75}

$$\Delta G_1 = \Delta G_{OH^*}$$

$$\Delta G_2 = \Delta G_{O^*} - \Delta G_{OH^*}$$

$$\Delta G_3 = \Delta G_{O^*} - \Delta G_{OH^*}$$

$$\Delta G_4 = \Delta G_{O_2} - \Delta G_{OOH^*}$$

Here, the experimental value of 4.92 eV is considered for ΔG_{O_2} .

The theoretical overpotential can be calculated using the given equation:³

$$\eta^{OER} = \max [\Delta G_1, \Delta G_2, \Delta G_3, \Delta G_4] - 1.23 \text{ eV}$$

Hydrogen evolution reaction involves the H₂ generation at the cathode when the proton gains electron from the catalyst surface. The Gibbs free energy for HER is given as:³

$$\Delta G_{H^*} = \Delta E_{H^*} - T\Delta S + \Delta ZPE$$

$$\Delta G_{H^*} = \Delta E_{H^*} + 0.24 \text{ eV}$$

The correction term for adsorption of H-atom i.e., $\Delta ZPE - T\Delta S$ is taken as 0.24 eV.

Adhesion energy

Adhesion energy of the heterostructure is given as:⁷⁷

$$E_{\text{adhesion}} = \frac{1}{A} [E_{\text{Sn}_4\text{P}_3-\text{SWCNT}} - (E_{\text{Sn}_4\text{P}_3} - E_{\text{SWCNT}})]$$

where $E_{\text{Sn}_4\text{P}_3-\text{SWCNT}}$, $E_{\text{Sn}_4\text{P}_3}$ and E_{SWCNT} represent the total energies of the heterostructure, Sn_4P_3 and SWCNT, respectively. A is the total surface area (258.97 \AA^2) of the heterostructure. The XRD analysis represents the most prominent crystalline planes of (107) and (110) for as grown Sn_4P_3 (Figure 3a). Therefore, the free-energies have been computed for OER intermediates on these two different surfaces of Sn_4P_3 as shown in Figure S27. Sn-site is found to be more active as compared to P-site towards OER for both the surfaces and further, Sn-site on (110) surface exhibits the best OER activity. Hence, Sn_4P_3 -(110) is chosen to build the heterostructure with metallic SWCNT. The adsorption energies of the OER intermediates (OH^* , O^* and OOH^*) have been computed on Sn as well as P sites (Table S10).

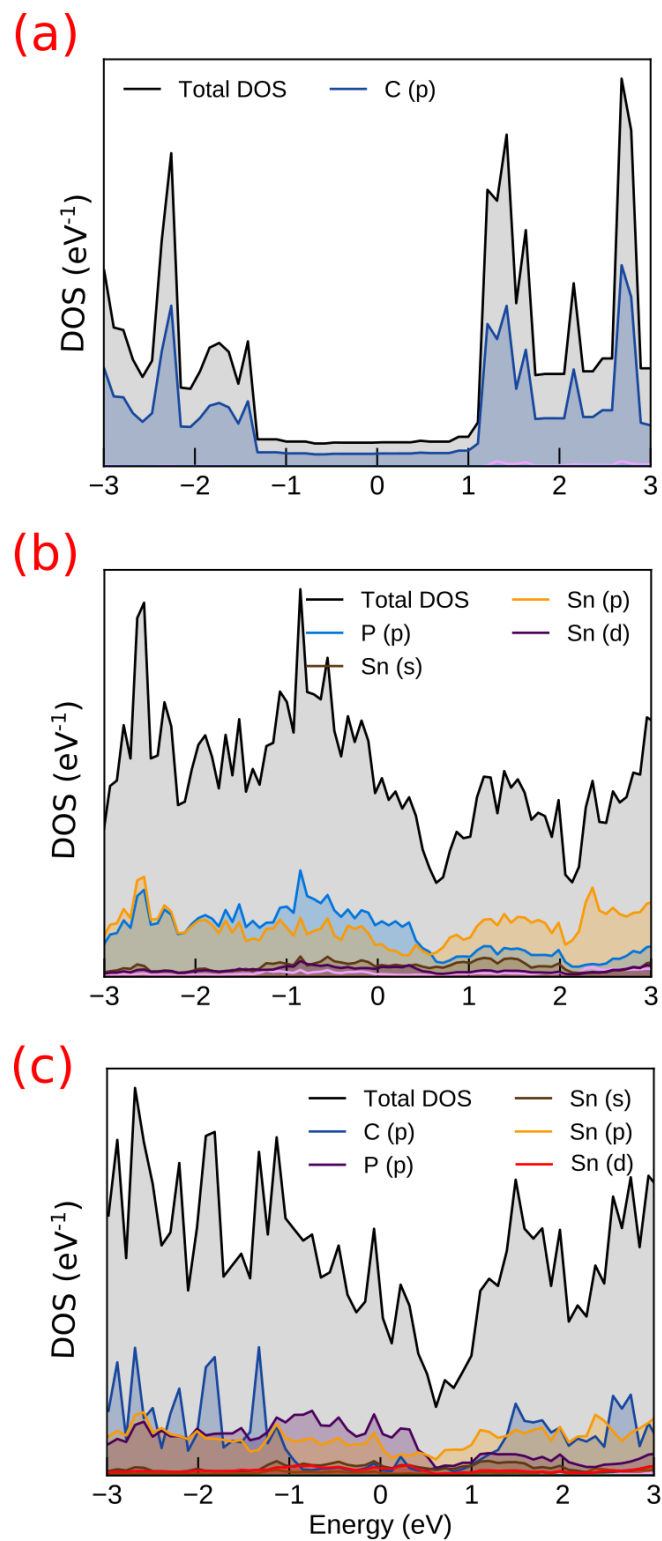


Figure S26. Projected DOS for (a) SWCNT, (b) Sn_4P_3 and (c) Sn_4P_3 -SWCNT heterostructure where the higher number of density of states of the heterostructure leads to a faster charge transfer process.

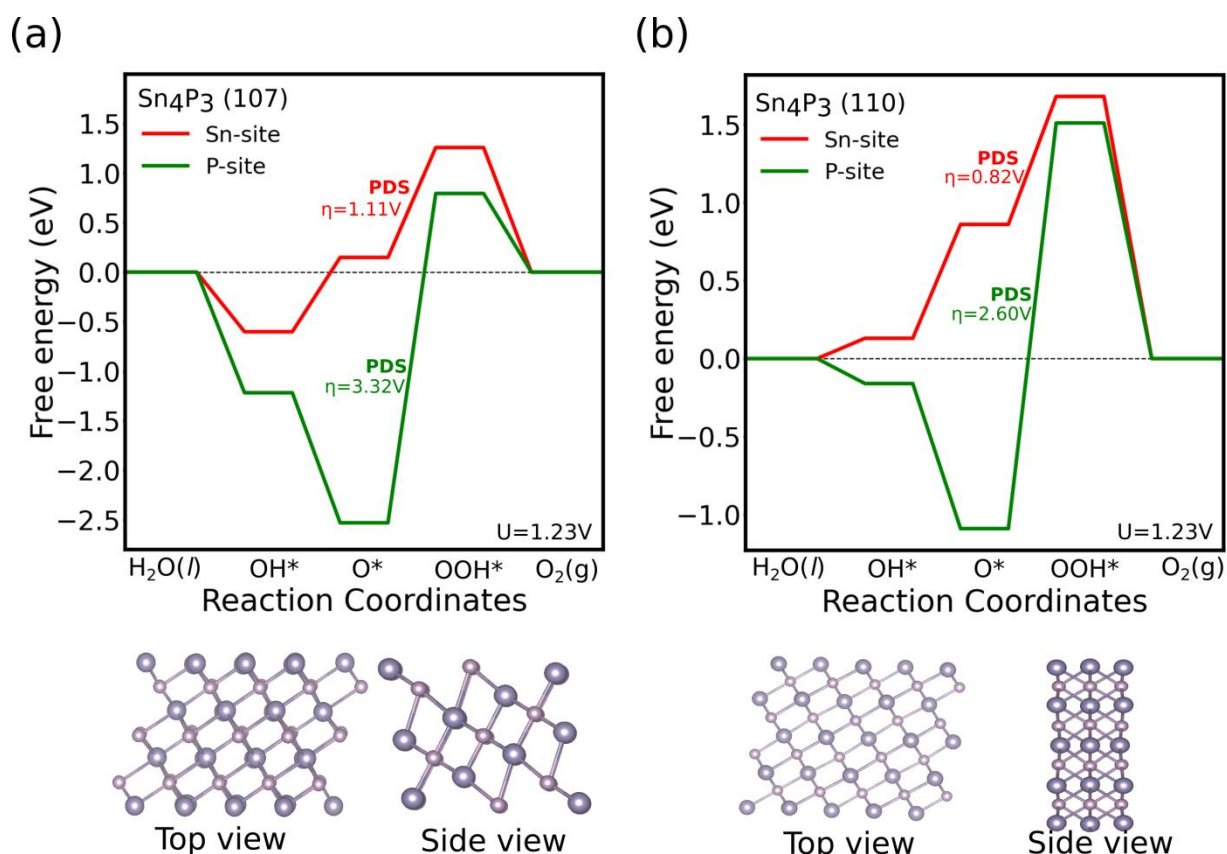


Figure S27. Gibbs free-energy diagrams of OER on the surface of (107) and (110) planes of Sn_4P_3 with their structural representations. Here, Sn_4P_3 (110) indicates more active plane towards OER process. Both P-site and Sn-site is active for OER reactions, out of which Sn-site is more active towards OER kinetics

Table S9. Total energy and adsorption energy of different adsorbates on the surface of pristine SWCNT, Sn_4P_3 and Sn_4P_3 -SWCNT heterostructure.

Surface	Total Energy (eV)	Adsorption energy (eV)
SWCNT	-519.30229533	-
SWCNT + OH*	-528.72603209	2.05
SWCNT + O*	-526.23640489	1.15
SWCNT + OOH*	-	5.35*
Sn_4P_3	-78.10384016	-
$\text{Sn}_4\text{P}_3@ \text{Sn} + \text{OH}^*$	-88.22411043	1.36
$\text{Sn}_4\text{P}_3@ \text{Sn} + \text{O}^*$	-82.86262287	3.32
$\text{Sn}_4\text{P}_3@ \text{Sn} + \text{OOH}^*$	-92.29533408	5.37
$\text{Sn}_4\text{P}_3@ \text{P} + \text{OH}^*$	-88.50840294	1.07
$\text{Sn}_4\text{P}_3@ \text{P} + \text{O}^*$	-92.46477395	1.37
$\text{Sn}_4\text{P}_3@ \text{P} + \text{OOH}^*$	-690.30542476	5.20
$\text{Sn}_4\text{P}_3\text{-SWCNT}@ \text{Sn} + \text{OH}^*$	-700.57853432	1.20
$\text{Sn}_4\text{P}_3\text{-SWCNT}@ \text{Sn} + \text{O}^*$	-695.10096994	3.29
$\text{Sn}_4\text{P}_3\text{-SWCNT}@ \text{Sn} + \text{OOH}^*$	-704.63422987	5.24

* For graphene and SWCNT, $\Delta E_{\text{OOH}^*} = \Delta E_{\text{OH}^*} + 3.3 \text{ eV}$.⁷⁶

Table 10. Gibb's free energy and overpotential values of pristine SWCNT, Sn₄P₃ and Sn₄P₃-SWCNT heterostructure.

Surface	ΔG_1 (eV)	ΔG_2 (eV)	ΔG_3 (eV)	ΔG_4 (eV)	η (V)
SWCNT	0.82	-2.13	2.98	-1.67	2.98
Sn ₄ P ₃	0.13	0.73	0.82	-1.68	0.82
Sn ₄ P ₃ -SWCNT	-0.023	0.853	0.72	-1.55	0.853

Table S11. Total energy and adsorption energy values of H* on the surface of pristine SWCNT, Sn₄P₃ and Sn₄P₃-SWCNT heterostructure.

Surface	Total Energy (eV)	Adsorption energy (eV)	Gibbs Energy (eV)
SWCNT + H*	-521.6763881	1.02	1.26
Sn ₄ P ₃ @Sn+ H*	-80.64409199	0.85	1.09
Sn ₄ P ₃ @P+ H*	-81.45881386	0.038	0.28
Sn ₄ P ₃ -SWCNT@P + H*	-693.7437643	-0.044	0.196

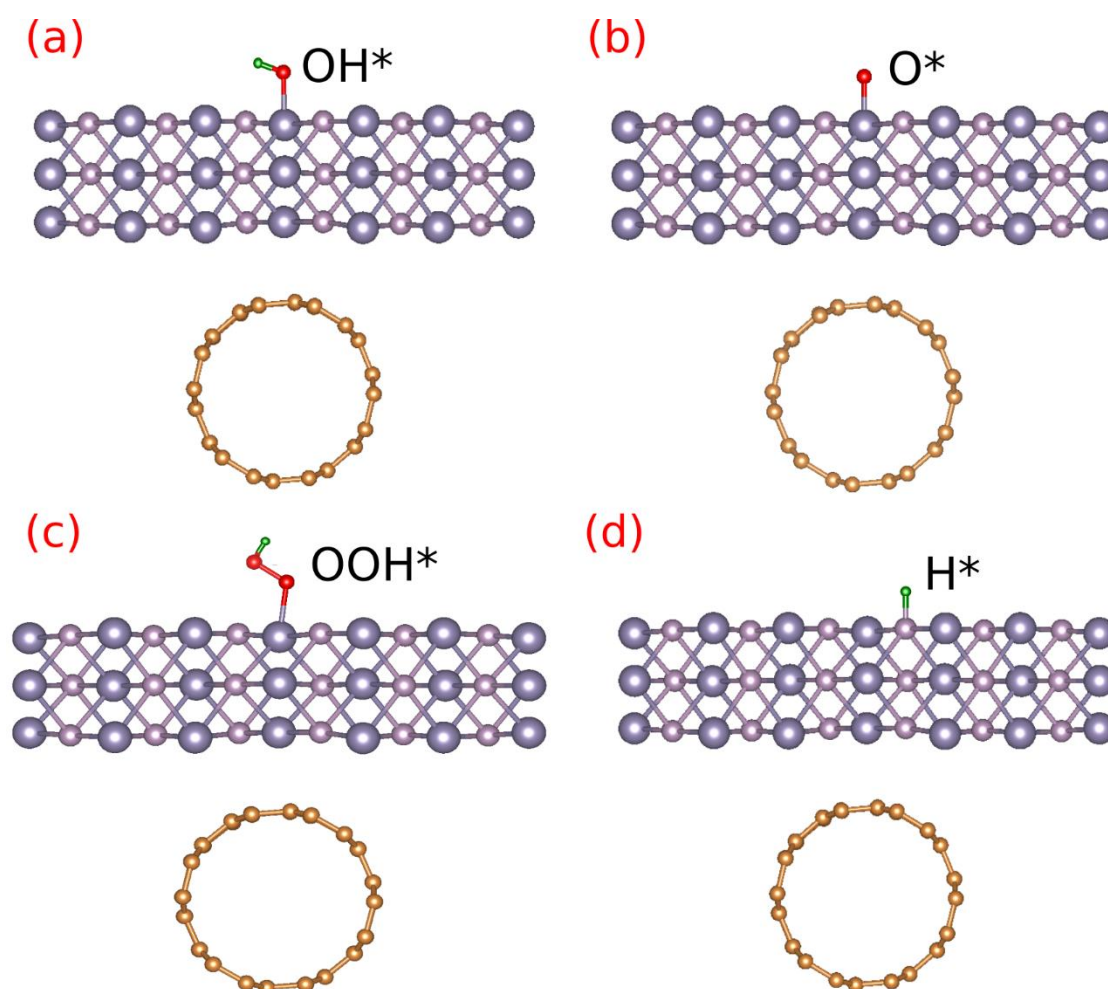


Figure S28. Structural representations of the adsorption of OER (OH*, O* and OOH*) and HER (H*) intermediates on the surface of Sn₄P₃-SWCNT heterostructure.

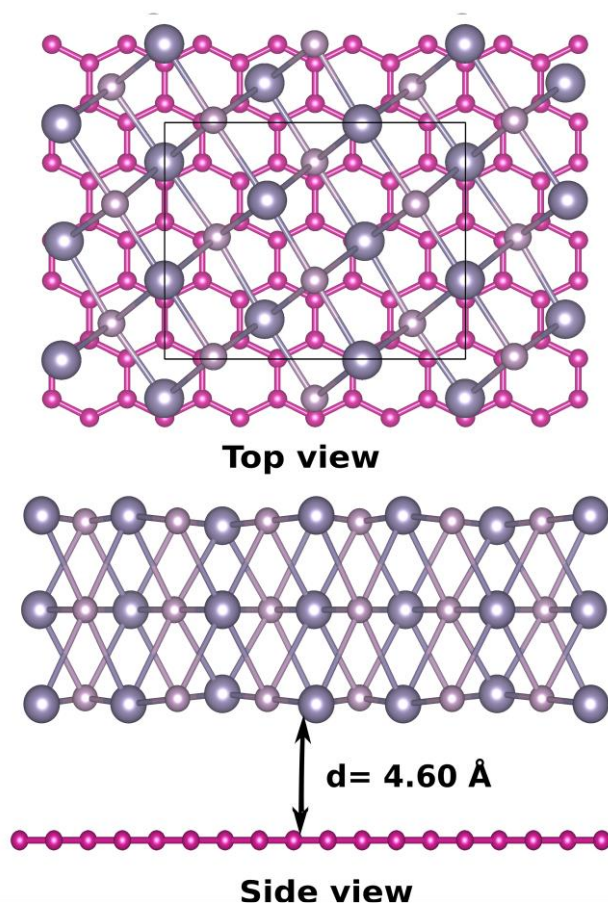


Figure S29. Schematic representation of top and side views of Sn_4P_3 -Graphene heterostructure. The Sn_4P_3 -Graphene heterostructure has been constructed using Sn_4P_3 (18 atoms) and graphene sheet (32 atoms). The optimized interlayer distance between Sn_4P_3 and graphene in heterostructure is found to be 4.6 Å.

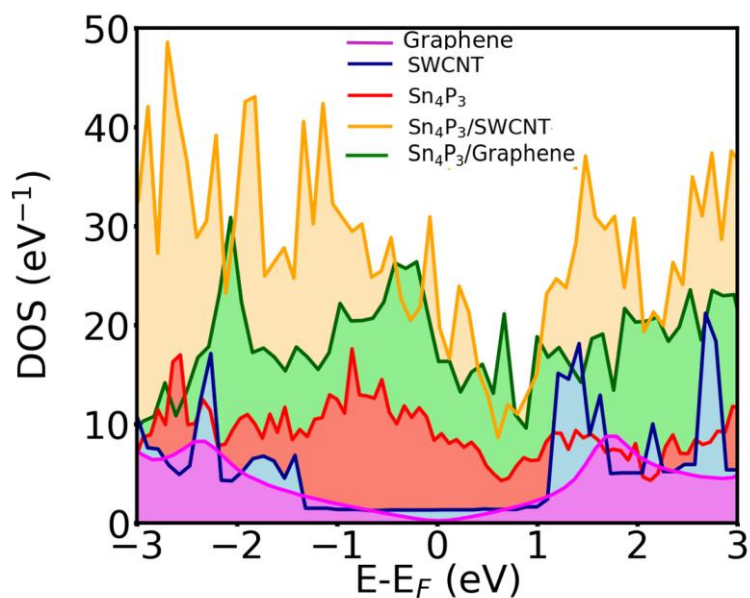


Figure S30. Density of states (DOS) plot for pristine Sn_4P_3 , pristine SWCNT, pure Graphene, Sn_4P_3 -SWCNT and Sn_4P_3 -Graphene heterostructure.

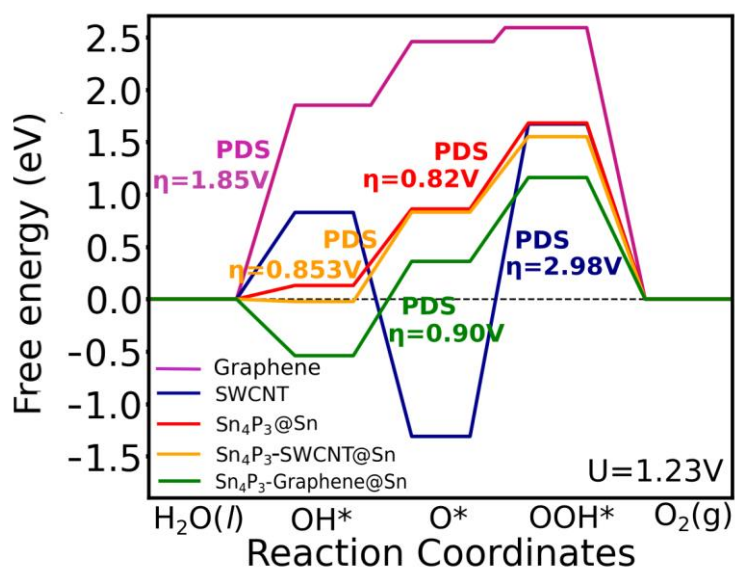


Figure S31. The calculated free energy for OER intermediates adsorbed on the surfaces of pristine SWCNT, Sn_4P_3 , pure graphene, Sn_4P_3 -SWCNT and Sn_4P_3 -Graphene heterostructures.

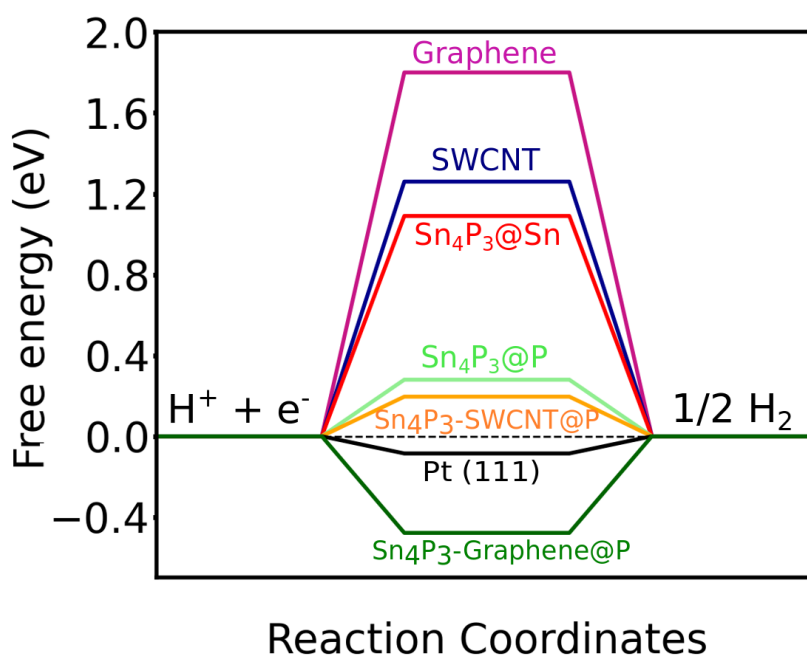


Figure S32. The calculated free energy for HER intermediate adsorbed on the surfaces of pristine SWCNT, pure graphene Sn_4P_3 , Sn_4P_3 -SWCNT and Sn_4P_3 -Graphene heterostructures.

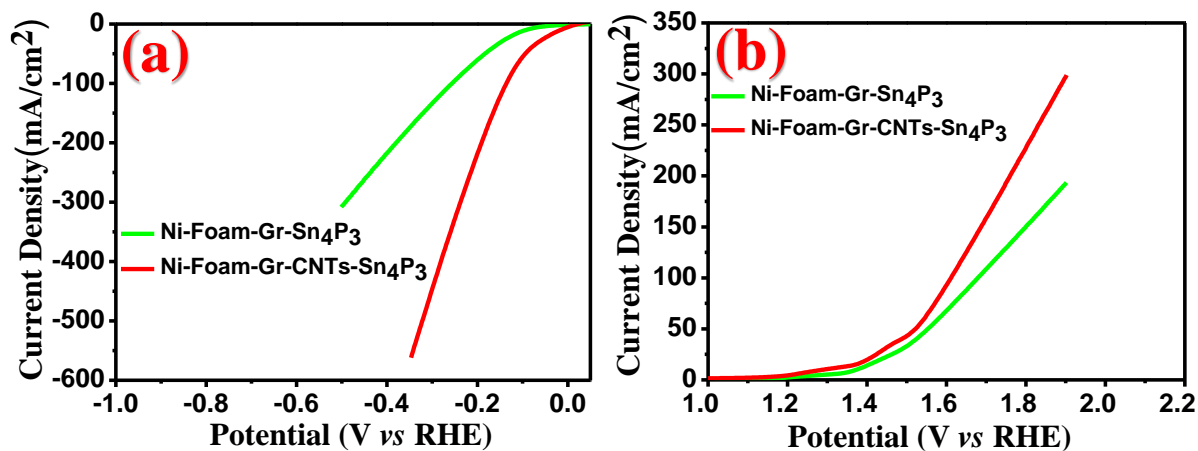


Figure S33. LSV polarization curve of Ni-Gr-Sn₄P₃ and Ni-Gr-CNTs-Sn₄P₃ samples for (a) HER and (b) OER

Overall water splitting mechanism:

The breaking of water is not a thermodynamically favourable reaction and its required 286 k/mol of energy input corresponding to a thermodynamic voltage of 1.23 V under standard conditions of room temperature and pressure. The water splitting mechanism is given as:

The total reaction: $2\text{H}_2\text{O} \rightarrow 2\text{H}_2 + \text{O}_2$

In acidic electrolyte: cathode surface: $4\text{H}^+ + 4\text{e}^- \rightarrow 2\text{H}_2$

anode surface: $2\text{H}_2\text{O} \rightarrow 4\text{H}^+ + \text{O}_2 + 4\text{e}^-$

In neutral and alkaline electrolytes:

cathode surface: $4\text{H}_2\text{O} + 4\text{e}^- \rightarrow 2\text{H}_2 + 4\text{OH}^-$

anode surface: $4\text{OH}^- \rightarrow 2\text{H}_2\text{O} + \text{O}_2 + 4\text{e}^-$

Table S12. Cell-voltage comparison of recently reported monometallic phosphide-based catalysts with the as-synthesized Ni-Gr-CNTs-Sn₄P₃ for overall water splitting in 1M KOH solution.

S.No.	Catalysts	Medium	Current Density (mA/cm ²)	Cell Voltage (V)	Ref
1	Cu ₃ P/NF	1 M KOH	10	1.63	21
2	Ni ₂ P/rGO/NF	1 M KOH	10	1.676	28
3	CoP/rGO	1 M KOH	10	1.70	33
4	Cu ₂ @Ni ₃ P ₄	1 M KOH	10	1.66	78
5	MOF Derived Ni ₂ P NS	1 M KOH	10	1.52	38
6	Ni ₂ P/Ni/NF	1 M KOH	10	1.49	40
7	Ni ₂ P@NSG	1 M KOH	10	1.572	44
8	Ni ₂ P NPs/NF	1 M KOH	10	1.63	55
9	Ni ₃ P ₄ /NF	1 M KOH	10	1.7	41
10	Ni-P@CP	1 M KOH	10	1.48	57
11	CoP-MNA/NF	1 M KOH	10	1.62	58
12	Co-P film	1 M KOH	10	1.56	59
13	CoP NWs	1 M KOH	10	1.56	60
14	Co ₂ P/CoNPC	1 M KOH	10	1.64	62
15	Co ₂ P NCs		10	1.56	63
16	Multishelled Ni ₂ P	1 M KOH	10	1.57	71
17	CoP-CNT	1 M KOH	10	1.52	72
18	C@Ni ₈ P ₃	1 M KOH	10	1.65	73
19	CoP ₂ /RGO	1 M KOH	10	1.56	74
20	N doped FeP	1 M KOH	100	1.72	79
21	Ni/Ni ₈ P ₃	1 M KOH	10	1.79	80
22	Mo-CoP ₃ /CC	1 M KOH	10	1.65	81
23	Fe-CoP/Ti	1 M KOH	10	1.60	82
24	FeP/CC	1 M KOH	10	1.69	83
25	Ni-Gr-CNTs-Sn ₄ P ₃	1 M KOH	10	1.482	This work

Estimation of the energy efficiency of water electrolyzer:

The water electrolysis proceeds via hydrogen evolution reaction (HER) on the cathode surface and the oxygen evolution reaction (OER) on the anode end. The minimal cell voltage E_{cell}^0 required for the electrolysis of water under standard condition (25° C temperature, 1 atm pressure) is given by the following equation⁶⁰

$$E_{cell}^0 = -\frac{\Delta G^0}{\eta F} \quad (1)$$

$$\Delta G^0 = -\eta F E_{cell}^0 \quad (2)$$

where ΔG^0 is the change in the Gibbs-free energy under standard conditions, η is the number of electrons transferred and F is the Faraday constant (F ~ 96485 C/mol)

$$\Delta G^0 = \Delta H^0 - TR\Delta n - T\Delta S^0 \quad (3)$$

For the electrolysis of water, the standard reaction enthalpy is $\Delta H^0 = 285.8 \text{ kJ mol}^{-1}$, $\Delta n = 1.5$, $\Delta S^0(\text{H}_2) = 130.6 \text{ J mol}^{-1}$, $\Delta S^0(\text{O}_2) = 205.1 \text{ J mol}^{-1}$, $\Delta S^0(\text{H}_2\text{O}) = 70 \text{ J mol}^{-1}\text{K}^{-1}$, $\Delta S_{Tot}^0 = 130.6 + \frac{1}{2}205.1 - 70 = 163.14 \text{ J mol}^{-1}\text{K}^{-1}$, and $\Delta G^0 = 237.2 \text{ kJ mol}^{-1}$. Thus, the minimum cell voltage for an open cell is

$$E_{cell}^0 = -\frac{\Delta G^0}{\eta F} = 1.23 \text{ V} \text{ according to equation (2).}$$

The maximum possible efficiency of an ideal open electrochemical cell is defined by the following equation:

$$\epsilon_{max} = \frac{\Delta H^0}{\Delta G} = -\frac{\Delta H^0}{nFE_{cell}} \quad (4)$$

where E_{cell} is the cell voltage to drive water splitting at the current of I ⁶⁰

$$E_{cell} = E_{cell}^0 + IR + \sum n \quad (5)$$

where R is the total ohmic series resistance in the cell which include the resistance of the external circuit, electrolyte, electrodes and membrane materials (if any); $\sum n$ is the sum of overpotentials including the activation overpotential at the two electrodes and the concentration overpotential.

The splitting of water by electrolysis is an endothermic reaction and the enthalpy change in this process is defined as follows⁶⁰:

$$\Delta H^0 = \Delta G^0 + T\Delta S = -\eta FE_{cell}^0 + T\Delta S \quad (6)$$

where ΔG^0 supplies energy in the form of applied electricity and the rest, $T\Delta S$, by heat.

In electrolysis, the cell voltage is always higher as compared to the theoretical voltage termed as reversible voltage $E_{cell}^0 = 1.23 \text{ V}$, where the voltage difference is converted into heat. When the cell voltage reaches 1.48 V , the heat generated by the overpotential and ohmic loss is used by the reaction and at this voltage, there is no heat generation or absorption to and from outside the system i.e., $\Delta S = 0$. Consequently, the cell voltage of 1.48 V is defined as “thermo-neutral potential” where the cell does not heat or cool and all-electric energy used for electrolysis is converted into heat content of evolved gas.⁶⁰ This voltage is used as the standard of 100% efficiency. As the thermo-neutral potential is a signature of 100% current efficiency in water electrolysis. Thus, the energy efficiency of water electrolysis can be calculated by dividing 1.48 V with the practical cell voltage (E_{cell}) and it can be derived by the following according equations (4) and (6):

$$\epsilon \cong \frac{1.48 \text{ V}}{E_{cell}} \quad (7)$$

Estimation of Faradaic efficiency:

Here, faradaic efficiency can be estimated by the gas chromatography (GC) technique *via* quantifying the gas products. It is expressed as:⁸⁴

$$\text{Faradaic Efficiency} = \frac{z \times F \times n}{I \times t}$$

z is the number of electrons involved in the HER and OER. Here, we consider that 2 electrons are involved in HER for one mole of hydrogen production and 4 electrons are involved in OER for one mole of oxygen production. n is the total number of moles of the gas evolved during water electrolysis reaction, F is the faraday constant ($F \sim 96485$ C/mol), I is the constant applied current and in our case it is fixed at ~ 10 mA/cm² during water splitting reaction. The quantification of produced gas during water electrolysis has been carried out every 20 minutes by taking $20 \mu\text{l}$ gas from the sealed vessel using an air-tight glass syringe followed by injection into the GC system. Before starting the quantification measurement, the solution is saturated with argon and N₂ gases for 1 hour each.

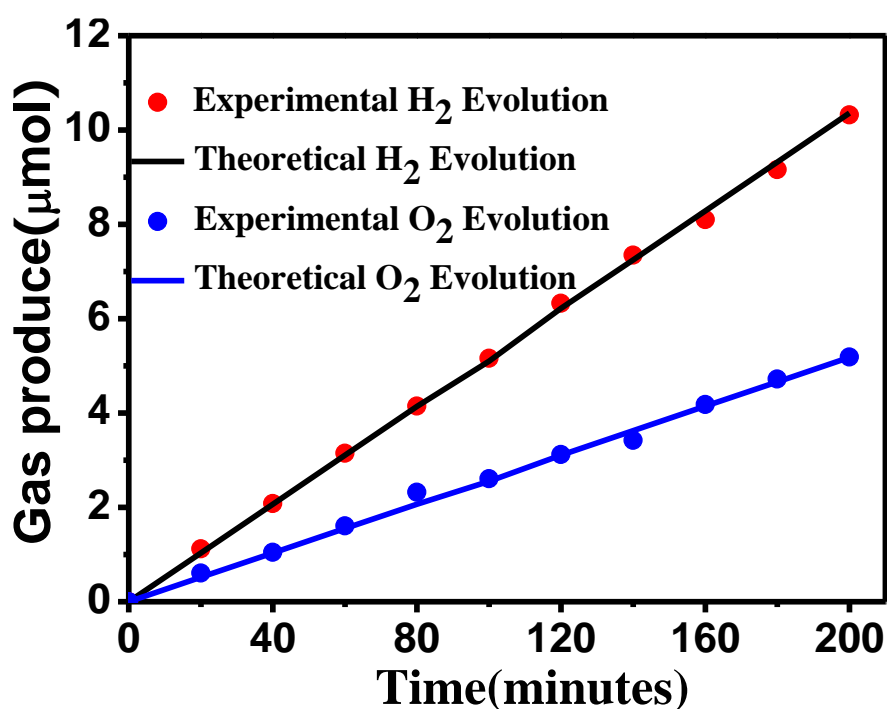


Figure S34. The experimentally and theoretically estimated amount of H₂ and O₂ produced by the Ni-Gr-CNTs-Sn₄P₃ electrode at a fixed current density of 10 mA/cm² in 1M KOH solution during water electrolysis. A very small deviation in experimental and theoretical result indicates ~ 99 % faradaic efficiency with the production of H₂ and O₂ at a molar ratio of 2:1.

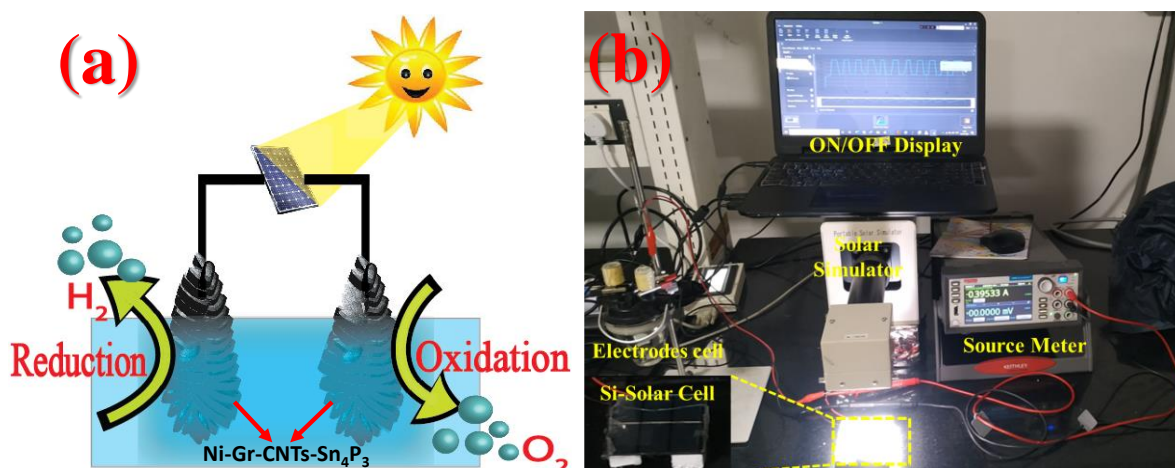


Figure S35. (a) Schematic illustration and (b) the optical photograph of the overall integrated set-up of the solar-driven water splitting.

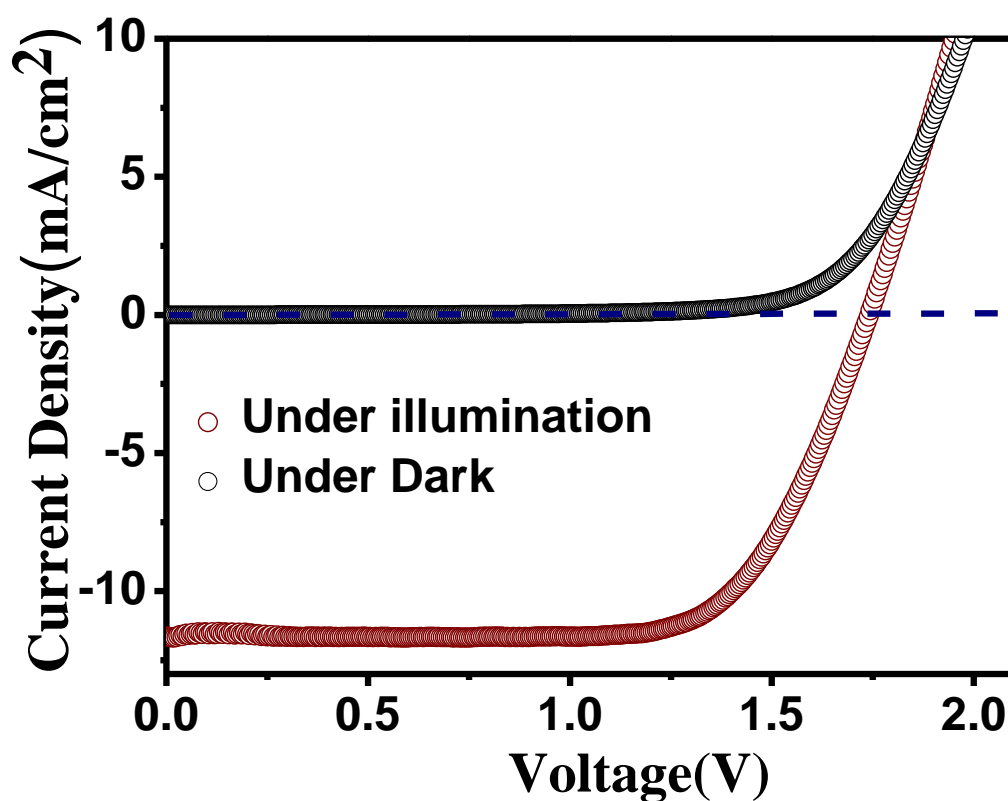


Figure S36. The I - V measurement of standard Si-solar cell under both dark and illumination (AM 1.5G, 100 mW/cm²) conditions.

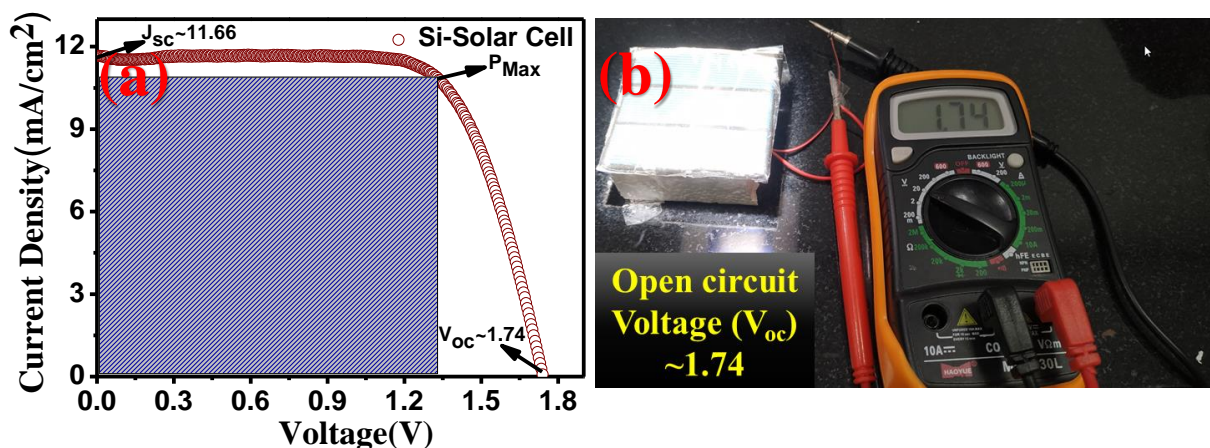


Figure S37. The I - V measurement of standard Si-solar cell in illumination (AM 1.5G, 100 mW/cm²) condition demonstrates a short-circuit photocurrent density (I_{sc}), open-circuit voltage (V_{oc}) and fill factor (FF) of 11.66 mA/cm², 1.74 V and 0.71, respectively with power conversion efficiency (PCE) of ~14.43% and (b) open-circuit voltage (V_{oc}) of Si-solar cell verified by multi-meter.

Table S13. Operating point and STH efficiency Ni-Gr-CNTs-Sn₄P₃ have been compared with the recently reported solar-driven electrocatalyst for overall water splitting.

S.No.	Solar cell category	Solar cell structure	Catalysts	Operating Point (mA/cm ²)	STH %	Ref
1	DSSC	FTO/TiO ₂ /Dye/PEDOT/FTO	WO ₃	2.52	3.10	85
2	DSSC	FTO/TiO ₂ /Dye/PEDOT/FTO	Fe ₂ O ₃	0.95	1.17	85
3	DSSC	FTO/TiO ₂ /Dye/Pt/FTO	WO ₃ /BiVO ₄	4.6	5.7	86
4	PSC	FTO/TiO ₂ /MAPbI ₃ /Spiro/Au	NiFe-LDH	10	12.3	87
5	PSC	FTO/TiO ₂ /MAPbI ₃ /Spiro/Au	BiVO ₄	2	2.5	88
6	PSC	FTO/TiO ₂ /MAPbI ₃ /Spiro/Au	FeNiOx/AI ₂ O ₃ /Fe ₂ O ₃	1.6	2.0	89
7	PSC	FTO/TiO ₂ /MAPbI ₃ /Spiro/Au	Mn-Fe ₂ O ₃	1.93	2.4	90
8	PSC	ITO/PEDOT/MAPbI ₃ /PCBM/Al	NiCo ₂ O ₄	5	6.2	91
9	PSC	FTO/NiMgLiO/MAPbI ₃ /PCBM/Ti(Nb)O _x /Ag	BiVO ₄	5.01	6.2	92
10	PSC	FTO/TiO ₂ /MAPbI ₃ /Carbon	CoP	5.5	6.7	93
11	PolSC	Triple Junction polymer	Pt	-	3.1	94
12	PSC	FTO/TiO ₂ /CH ₃ NH ₃ PbI _{3-x} Cl _x /Spiro/MoO ₃ /Au	NiFe alloy NP supported N, S-doped carbon	7.6	9.7	95
13	PSC	Based on CsPb _{0.9} Sn _{0.1} Br ₂	Ni _{0.5} Co _{0.5} P/CP	-	3.12	96
14	PSC	ITO/PEDOT: PSS/	Te/SnS ₂ /A	-	0.49	97

		$\text{CH}_3\text{NH}_3\text{PbI}_{3-x}\text{Cl}_x$ /PCBM/CBLs/Al	g			
15	Silicon	2-jn a-Si	W:BiVO ₄	4	4.9	92
16	Silicon	3-jn a-Si	Co- OEC/NiM o Zn	-	2.5	98
17	Silicon	2-jn planar -Si	NiFeSP/N F	-	9.2	99
18	Silicon	2-jn planar -Si	Ni-Gr- CNTs- Sn ₄ P ₃	8.89	10.8 2	Thi s wo rk

PSC- perovskite solar cells, PolSC- polymer solar cell, DSSC, Dye-Sensitized Solar Cells

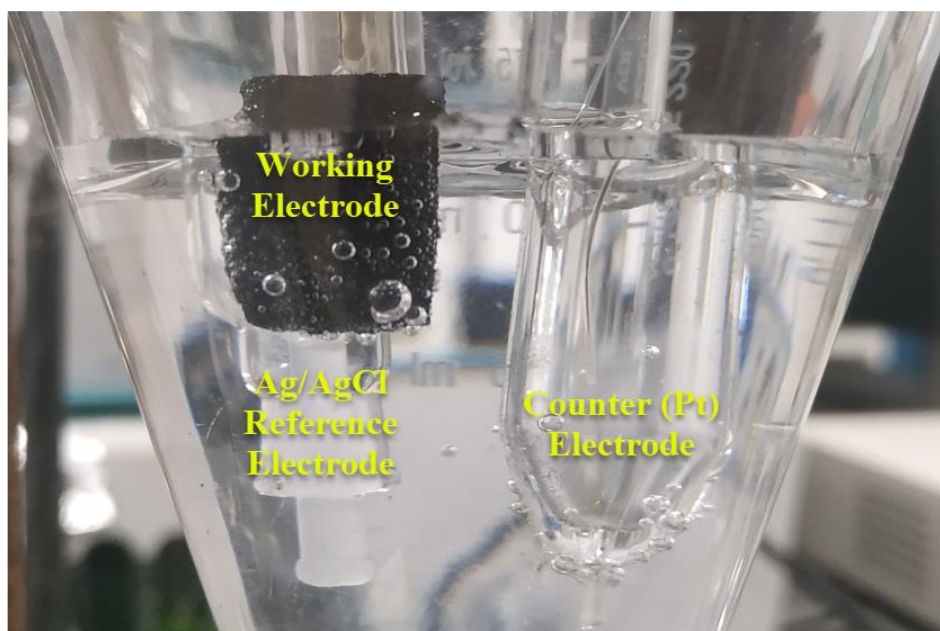


Figure S38. Optical photograph of the three-electrode setup for electrochemical measurement. Here, Pt wire acts as a counter electrode, Ni-Gr-CNTs-Sn₄P₃ is a working electrode and Ag/AgCl. The bubble formation on the working electrode surface confirms gas generation *via* water splitting.

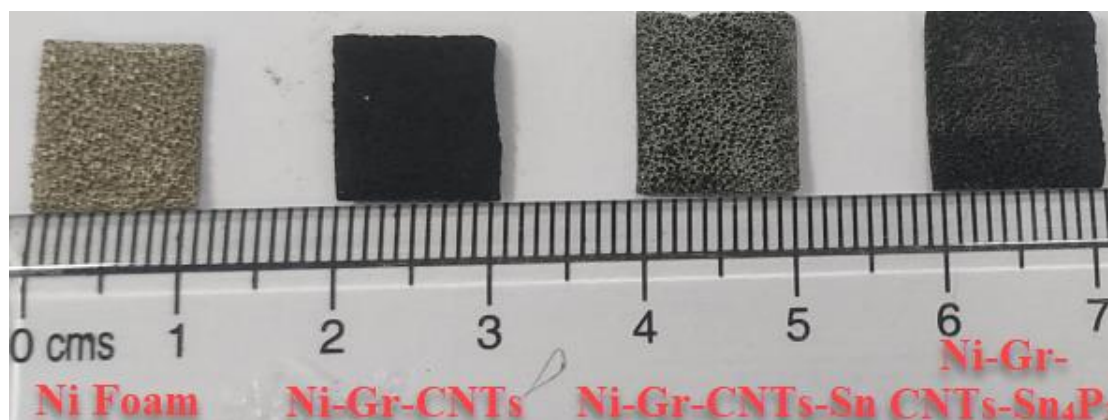


Figure S39. The optical photograph of the as-prepared electrodes where the effective working electrode area is $\sim (1 \times 1.2) \text{ cm}^2$ which is immersed in the electrolyte during the electrochemical measurement.

Supplementary videos:

Video S1: Under the chronoamperometric measurement, the continuous H₂ and O₂ bubbles are generated from the two similar electrodes of Ni-Gr-CNTs-Sn₄P₃ acting as cathode and anode, respectively. The video demonstrates that small-sized gas bubbles are continuously releasing from the electrode surface and its immediate disappearance confirms the super-aerophobic nature of the as-designed catalyst. This property helps to minimize the dead surface area and creates a better exposure of the active sites towards the electrolyte thereby enhancing the overall water splitting performance.

Video S2: The live video demonstrates the solar-driven overall water splitting which has been tested by integrating with a commercially available Si solar cell (open circuit voltage ~ 1.74 V). Here the water electrolyser is powered by the solar-cell for the continuous generation of H₂ and O₂ from the cathode and anode surface, respectively. This demonstrate a footprint towards the development of green fuel production *via* renewable energy resources which is a better alternative as compared to the existing photoelectrochemical technology.

Video S3: The live video demonstrates the super-hydrophilic nature of the catalyst as it is clearly observed that all the water droplets are getting absorbed and immediately seeps through the surface of Ni-Gr-CNTs-Sn₄P₃. The super-hydrophilic property of the catalyst creates more wettable surface as well as activates the electrocatalyst surface towards electrolyte and thereby improves the overall water splitting performance.

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