## **Associated Content**

## Instability of Hydrogenated TiO<sub>2</sub>

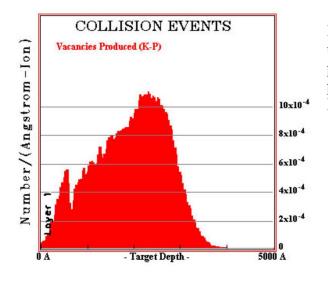
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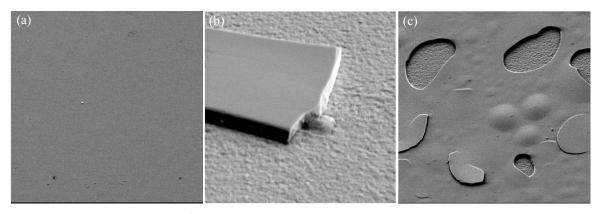
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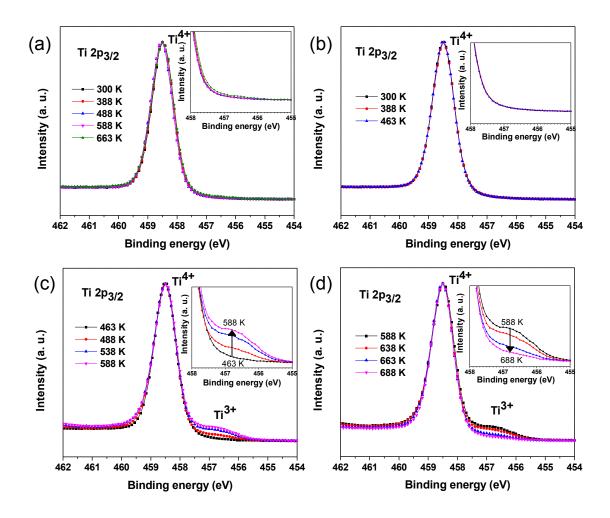
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**Figure S1**: Oxygen vacancy profile obtained using the Kinchin-Pease (K-P) model. <sup>1</sup> The profile corresponds to a peak defect generation probability of 1 in 1000 for every incident 45 KeV H<sup>+</sup>.



**Figure S2**: SEM images of H<sup>+</sup>-implanted surfaces (40 keV ion energy): (a)  $TiO_2(110)$  surface after annealing at 800 K, (b)  $SrTiO_3(100)$  after annealing at 870 K (image from <sup>2</sup>), and (c)  $ZrO_2(100)$  after annealing at 770 K (image from <sup>3</sup>). The H<sup>+</sup> doses were  $1.0x10^{17}$ ,  $5.0x10^{16}$ , and  $1.0x10^{17}$  cm<sup>-2</sup>, respectively.



**Figure S3.** The XPS Ti  $2p_{3/2}$  high resolution spectra of the (a) pristine single crystal  $TiO_2(110)$  sample in-situ annealed from room temperature to 663 K and H-implanted single crystal  $TiO_2(110)$  sample in-situ annealed from (b) 300 to 463 K (c) 463 to 588 K, and (d) 588 to 688 K indicating the reduction of  $Ti^{4+}$  to  $Ti^{3+}$  from 463 to 588 K and re-oxidation of  $Ti^{3+}$  to  $Ti^{4+}$  from 588 to 688 K for H-implanted  $TiO_2$  in contrast to pristine  $TiO_2$ . The enlarged Ti 2p regions associated with  $Ti^{3+}$  are shown in the inset of each spectrum.

The Ti  $2p_{3/2}$  peak of pristine  $TiO_2(110)$  (Figure S3a) at 758.5 eV, ascribable to  $Ti^{4+}$ , was used as the reference point for all other spectra. This peak did not change in position or shape for the unimplanted sample after annealing up to ~700 K. (As noted in the experimental section, temperature readings in the XPS system erred toward higher values.) At most, the extent of thermally generated surface  $Ti^{3+}$  in pristine  $TiO_2(110)$  was ~3%, which did not change

significantly during the annealing sequence of Figure S3a. The H-implanted  $TiO_2(110)$  sample showed similar behavior to the pristine sample up to 463 K (Figure S3b and its inset), however  $Ti^{3+}$  peak appeared at 757.5 eV after annealing beyond 463 K (Figure S3c and its inset). This indicates that the H-implanted surface was readily reduced as the implanted H moved through the interface. Again, this is in sharp contrast to the behavior of pristine  $TiO_2(110)$  surface (Figure S3a). The near-surface  $Ti^{3+}$  concentration increased to, at most, ~12% with annealing from 463 to 588 K (see inset of Figure S3c), suggesting that most of the implanted H departed as  $H_2$ .

When the implanted sample was further annealed from 588 to 688 K (Figure S3d), the  $Ti^{3+}$  concentration decreased to ~4% almost reaching the amount of  $Ti^{3+}$  on the pristine  $TiO_2(110)$  surface. This result is consistent with the known thermal properties of reduced  $TiO_2$  surfaces in vacuum, namely that excess surface reduction is transmitted at these temperatures into the bulk of rutile via diffusion of  $Ti^{3+}$  interstitials. <sup>4,5</sup>

## References

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