

~~Operando Observation and Detailed~~ ~~Chemical Modeling of the Bilayer Solid~~ Electrolyte Interphase

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<https://bit.ly/2vhquqk>



Acknowledgements

Collaborators

At the NCNR: Joseph A. Dura, Eric D. Rus, Paul Kienzle

At Colorado School of Mines: Christopher H. Lee (MS, 2016), Amy LeBar (MS, 2019), Center for High Performance Computing

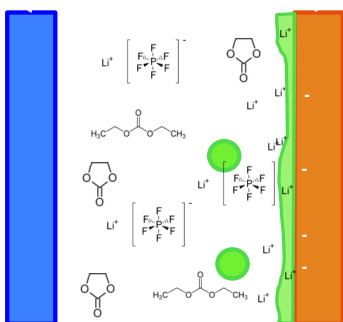
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The Solid Electrolyte Interphase (SEI)



• Because the electrolyte is a liquid, it is not possible to operate a battery without a protective layer.

- Electrolyte formulations increase passivity to validate the

- Material composition products Solid Electrolyte Interphase (SEI).

- Electrical modeling to balance SEI development 'all-in'.

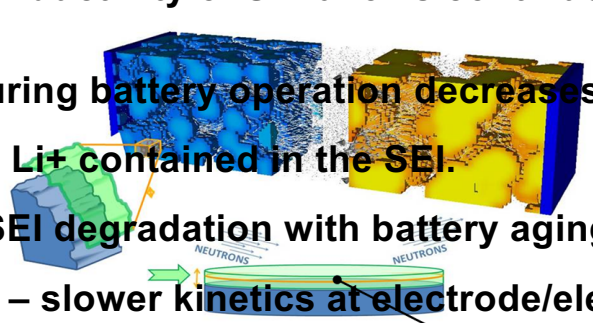
- Ion conductivity of SEI allows continued battery operation

• In reality, continued SEI growth during battery operation decreases device durability and efficiency

- Long-term capacity fade due to Li^+ contained in the SEI.

- Durability and safety issues – SEI degradation with battery aging.

- Higher resistance, lower power – slower kinetics at electrode/electrolyte interface.



Reduce electrolyte on working electrode
(non intercalating) to form SEI

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State of SEI Understanding (1/3): Many reactions are known; few are quantified



An, et al., Carbon, 2016

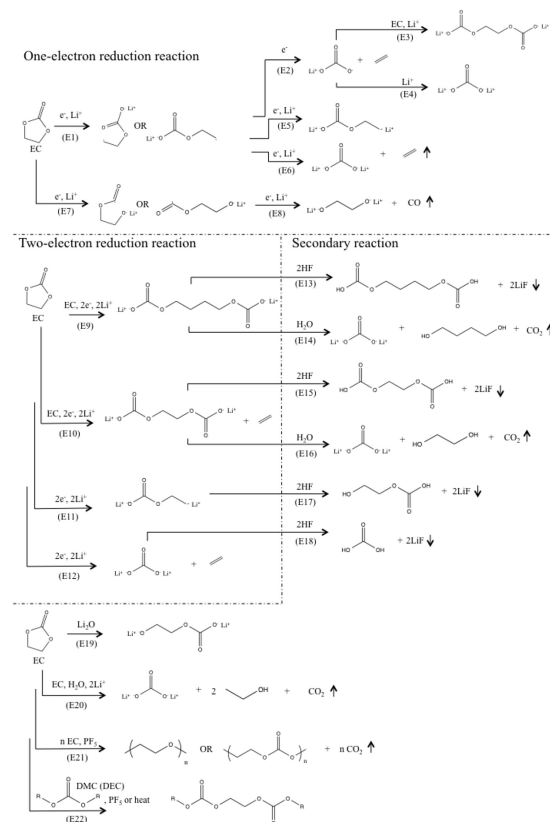
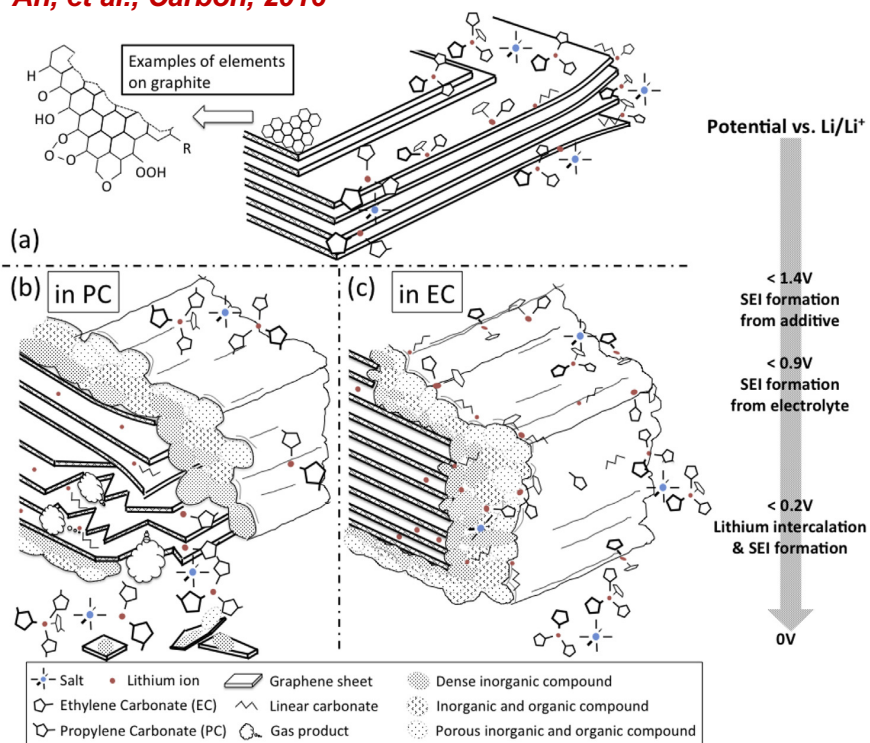


Fig. 8. Ethylene carbonate (EC) reduction process (reference groups in parentheses; details are shown in Table 2).

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One-electron reduction reaction

$$\text{R-O-C(=O)-O-R} \xrightarrow[\text{(L1)}]{\text{e}^-, \text{Li}^+} \text{R-O-C(=O)-O}^\bullet\text{Li}^+ + \text{R}^\bullet \xrightarrow[\text{(L2)}]{\text{e}^-} \text{R}^-$$

LC
(DMC, DEC)

Two-electron reduction reaction

$$\text{R-O-C(=O)-O-R} \xrightarrow[\text{(L3)}]{2\text{e}^-, 2\text{Li}^+} \text{Li}^+\text{O-C(=O)-O}^\bullet\text{Li}^+ + \text{R-R}$$

LC
(DMC, DEC)

Secondary reaction

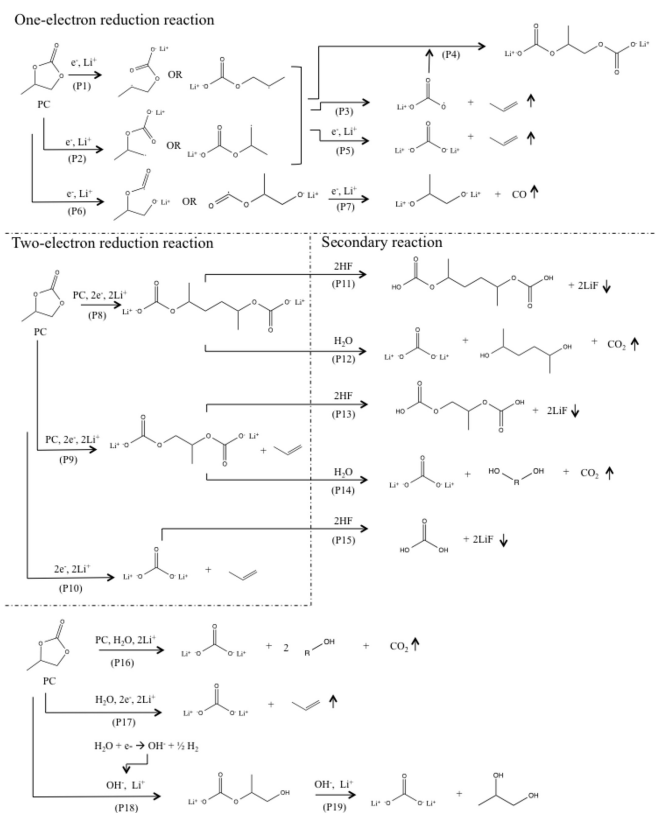
$$\begin{aligned} &\xrightarrow[\text{(L5)}]{2\text{HF}} 2\text{LiF} \downarrow + \text{HO-C(=O)-OH} \\ &\xrightarrow[\text{(L6)}]{\text{HF}} \text{LiF} \downarrow + \text{R-O-C(=O)-OH} \\ &\xrightarrow[\text{(L7)}]{\text{HF}} \text{LiF} \downarrow + \text{R-OH} + \text{CO}_2 \uparrow \end{aligned}$$

$$\text{R-O-C(=O)-O-R} \xrightarrow[\text{(L8)}]{\text{LC, H}_2\text{O}, 2\text{Li}^+} \text{Li}^+\text{O-C(=O)-O}^\bullet\text{Li}^+ + 2 \text{R-OH} + \text{CO}_2 \uparrow$$

LC
(DMC, DEC)

$$\text{R-O-C(=O)-O-R} \xrightarrow[\text{(L9)}]{\text{EC, PF}_3 \text{ or heat}} \text{R-O-C(=O)-O-CH}_2\text{CH}_2\text{O-C(=O)-R}$$

EC
(DMC, DEC)

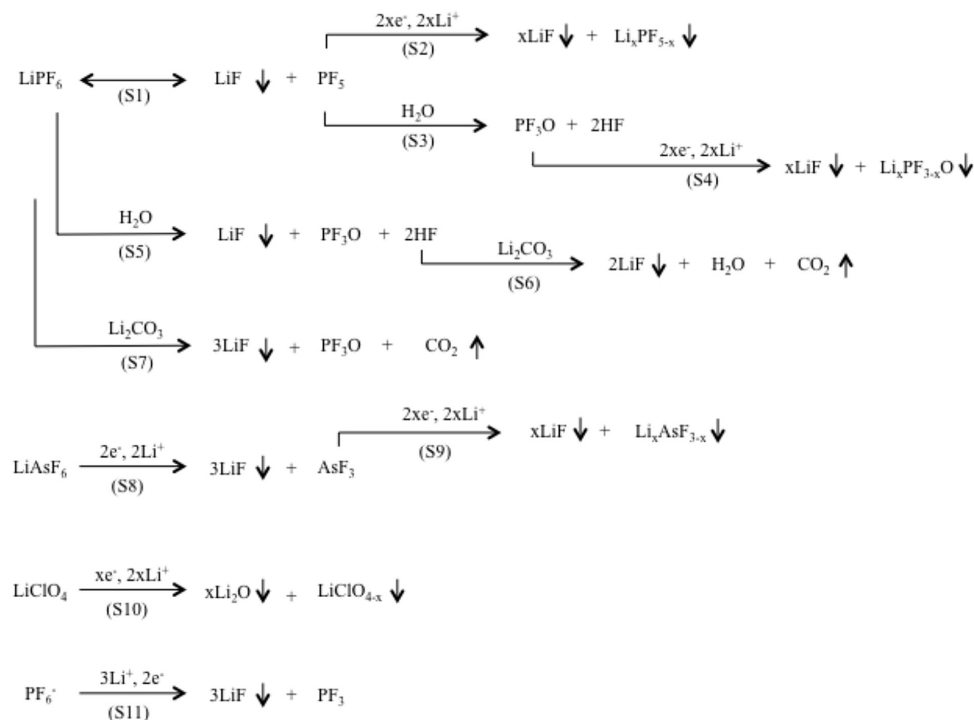


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State of SEI Understanding (3/3): Many reactions are known; few are quantified



An, et al., Carbon, 2016



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There are 2 primary needs, for mechanistic understanding of SEI growth & Evolution



Need #1: Operando validation data with high spatio-chemical resolution.

Need #2: Detailed electrochemical modeling tools, with flexible and robust mechanism implementation

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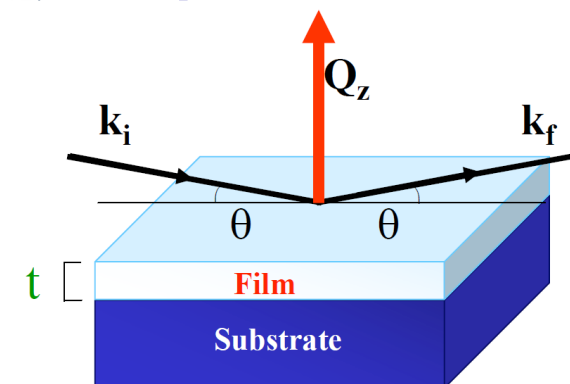
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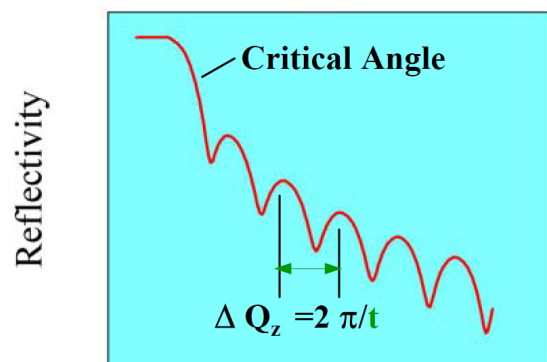
Neutron Reflectometry Provides Sub-nm Resolution Depth Profiles



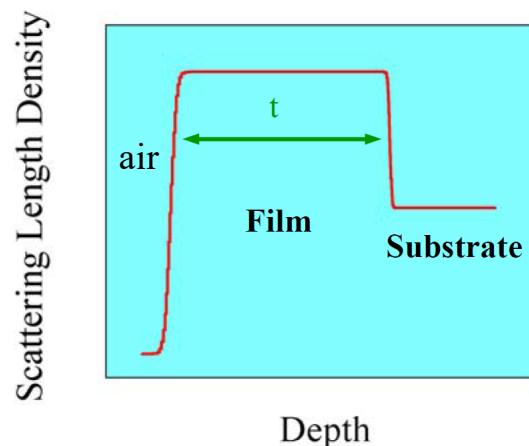
- Measures Reflected Intensity vs. grazing angle θ
- Oscillations with period 2π / layer thickness
- NR Provides Depth Profile of the SLD
- SLD related to Composition:

$$SLD(z) = \sum_j SLD_j V_j$$

$$SLD(z) = \sum_i b_i n_i$$



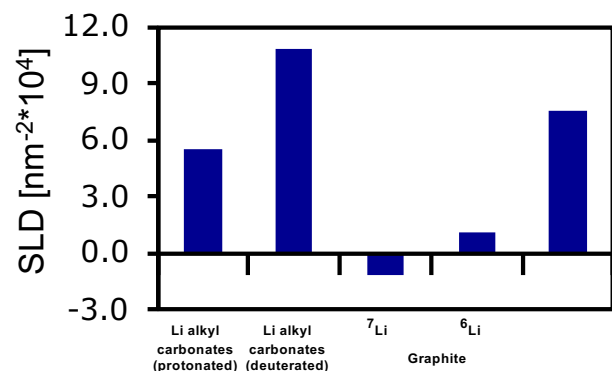
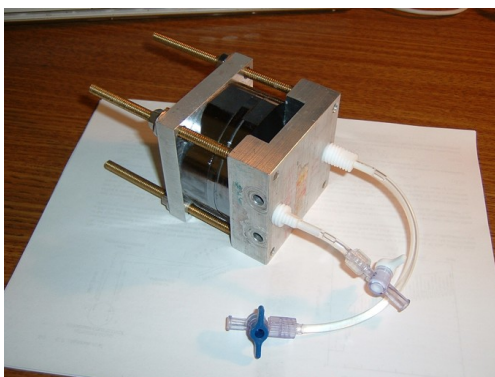
Fitting

<https://bit.ly/2vhquqk> Q_z



Neutron Reflectometry: Challenges and Advantages for Electrochemical Interfaces



NR has several inherent advantages as an *operando* probe for electrochemical systems

- Sensitive to structures with sub-Å resolution.
- Neutrons transmit through many solids with little attenuation - simplifies sample env.
- High SLD contrast for relevant isotopes.

Challenges:

- Requires large, very flat ($r < 20 \text{ Å}$) surfaces
- Long collection times (several hours) for best resolution.
- Complicated data fitting/analysis - no unique solution to a single NR spectrum.

<https://bit.ly/2vhquqk>

Experiment 1: NR on Copper Reveals a thin, 4 nm Layer That is Rich in Li



NR at OCV and 250 mV vs. Li/Li+ fit simultaneously

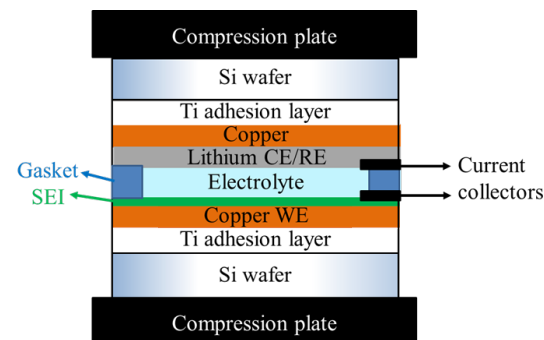
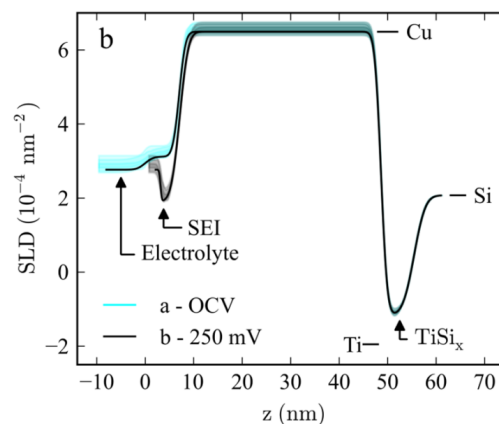
Constrain substrate, WE parameters.

NR measurements at OCV reveal an initial CuCO_3 + $\text{Cu}(\text{OH})_2$ layer.

10 initial cycles remove this layer and form an SEI that is 4.0 nm thick at 250 mV vs. Li.

The SEI is easily observed via NR, and is rich in Li-containing molecules (low SLD).

Owejan, et al., Chem. Mater., 2014



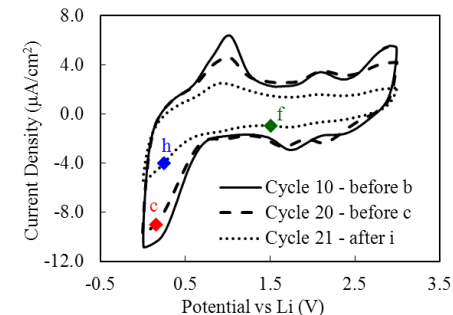
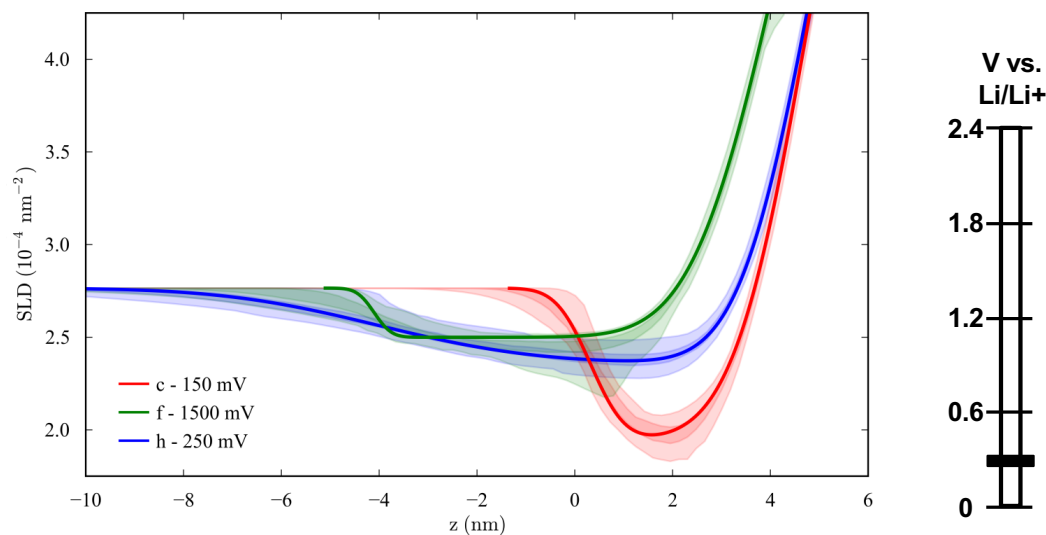
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Electrochemical cycling reveals SEI “Breathing.”



NR results during potentiostatic holds reveal changes to the SEI thickness, composition, and structure with cycling.

- SLD decreases (Li increases) with decreasing WE potential.
- SEI thickness continues to increase with additional processing.
- Some test points show composition gradients; others show mixing.



Owejan, et al., Chem. Mater., 2014

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Experiment 2: W Anode Enhances SEI Sensitivity

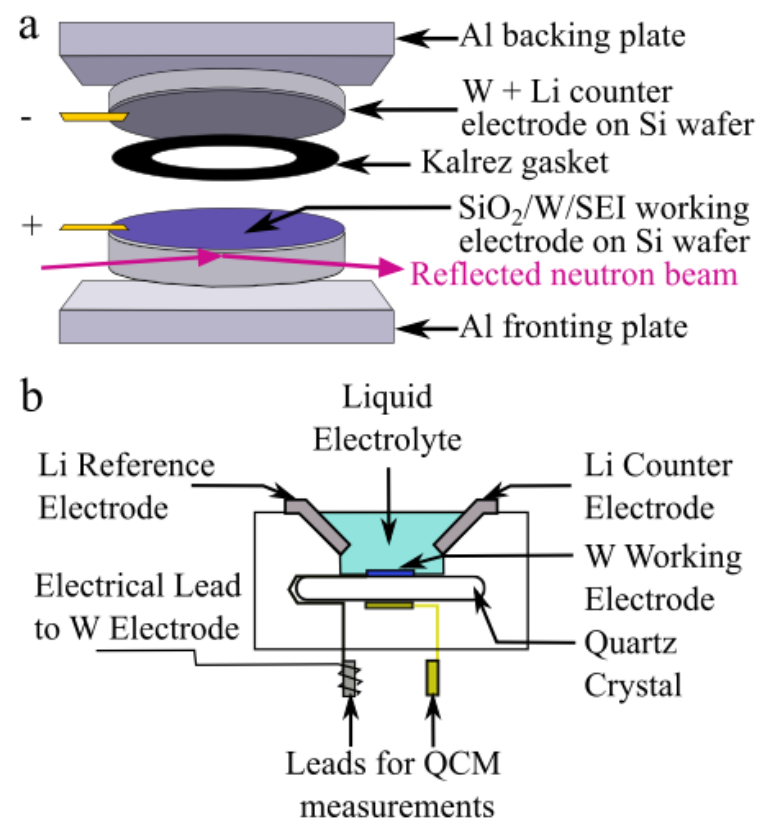


NR Measurements

- NR in He: structure w/ no SEI
- NR during hold at reducing potential (0.56 V)
- Measure in electrolyte, no electrochemistry

Electrochemistry + QCM-D

- Quartz crystal microbalance: highly resolved mass uptake ($\sim 2 \text{ ng/cm}^2$).
- Simultaneous cyclic voltammetry



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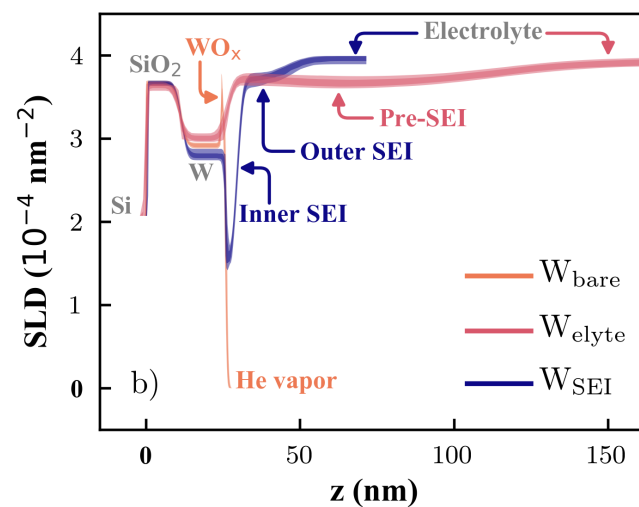
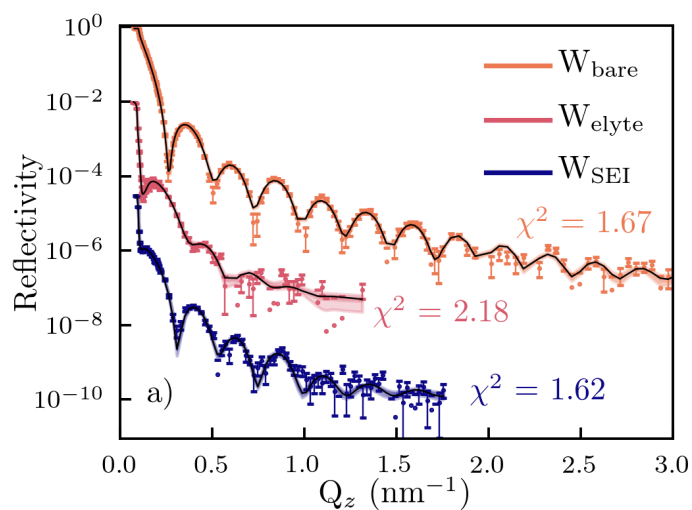
NR Directly Observed Hypothesized Two-layer SEI



Simultaneously fit NR data in He and after SEI growth

Constrains substrate layer parameters for greater confidence

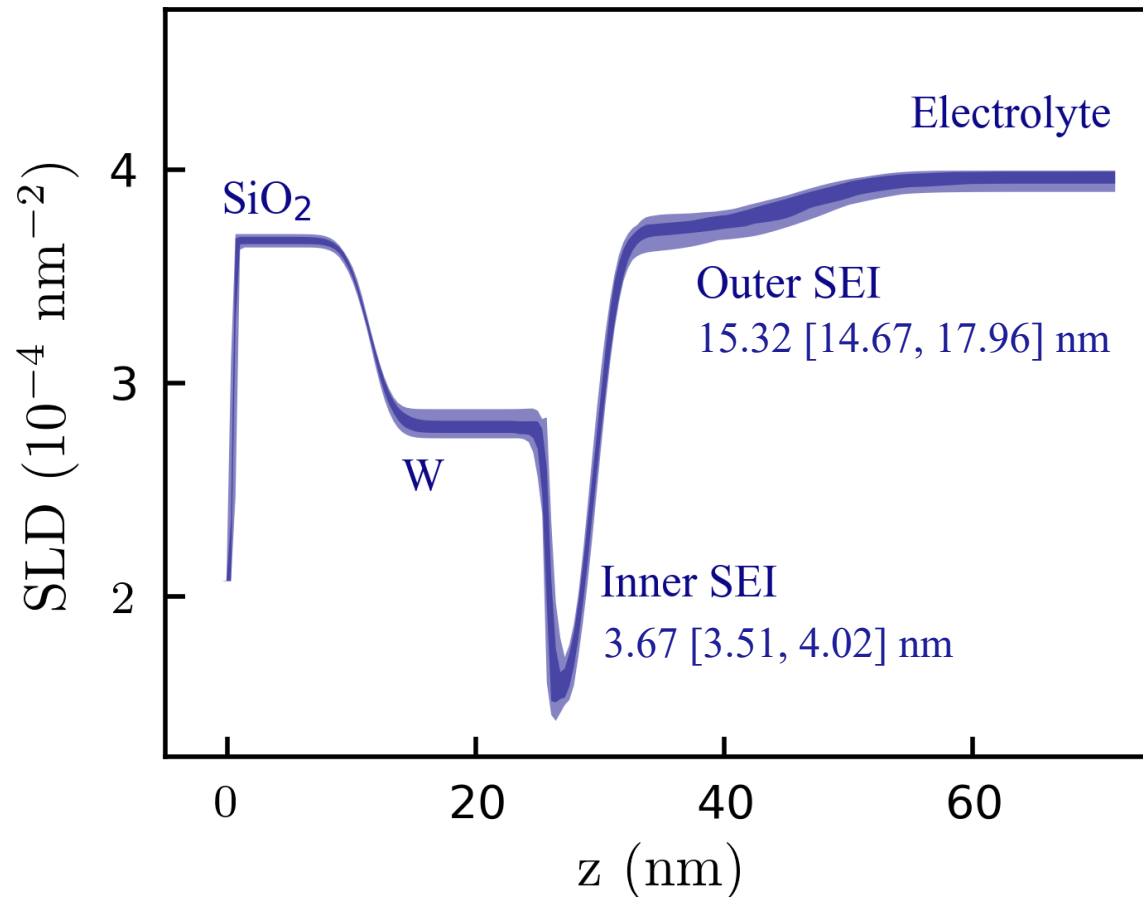
Can be used for element conservation.



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Lee, et al., J. Power Sources., 2019





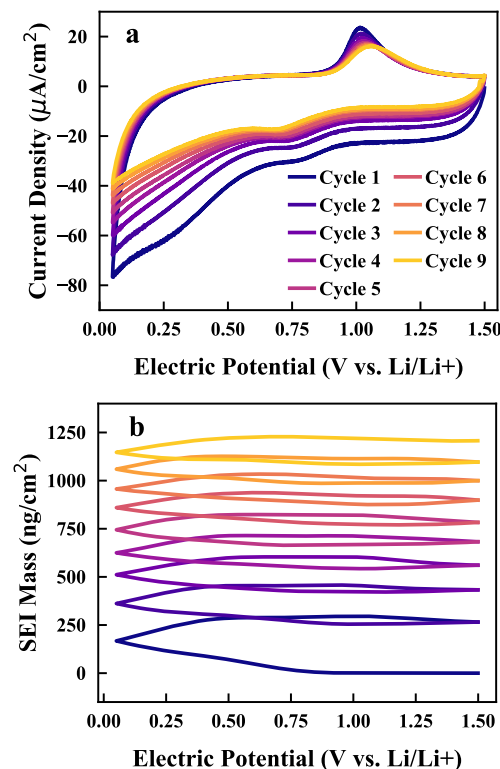
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Lee, et al., J. Power Sources., 2019

EQCM-D Shows Growth, Re-dissolution of SEI



- Reduction peaks at 800 mV and 275 mV, vs. Li/Li⁺.
- Oxidation peak at 1.1 V.
- CV qualitatively similar to that on non-intercalating Cu.
- Current decreases as a function of cycle number, decrease in mass gain per cycle: passivation: SEI passivation
- Mass uptake *and* mass loss during each cycle: SEI breathing



Lee, et al., J. Power Sources., 2019

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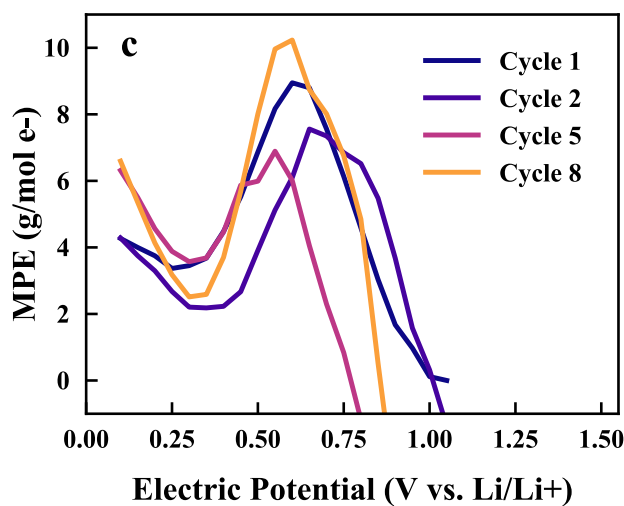
Mass Per Electron (MPE) Elucidates SEI Chemical Evolution with Cycling



- Two main processes
 - 750 mV: higher MPE
 - Below 300 mV: lower MPE

$$m.p.e = \frac{F \Delta m}{\Delta Q}$$

- Large “extraneous current” prevents quantitative species ID.



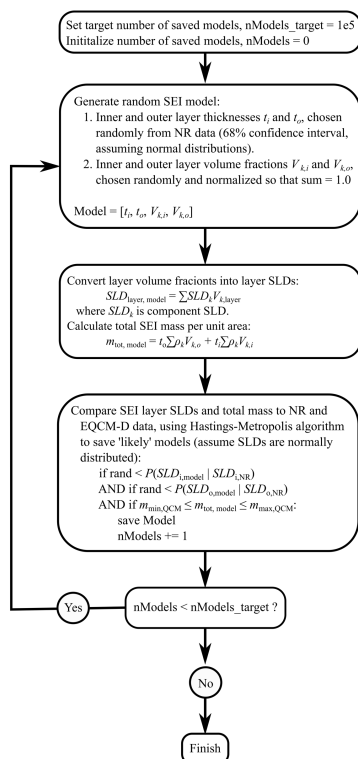
Compound	m.p.e (g/mol e ⁻)
Li	7
Li ₂ O	15
LiO ₂	19
LiOH	24
LiF	26
Li ₂ CO ₃	37
(CH ₂ OCO ₂ Li) ₂	81

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Lee, et al., *J. Power Sources*, 2019



Monte Carlo Simulations ID Chemical Compositions Consistent with the Data



User Inputs

Generate Random Model

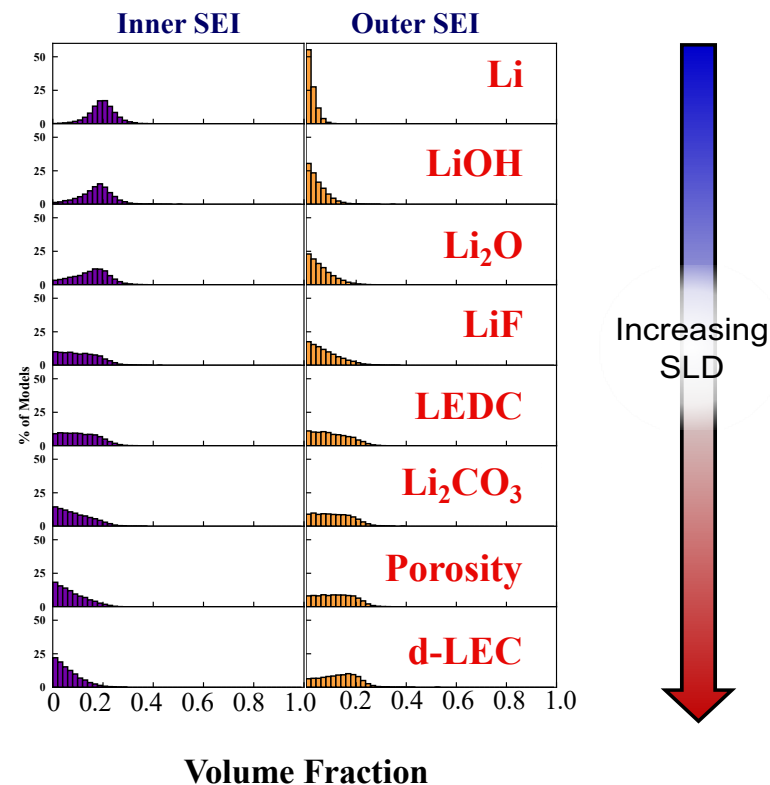
- Layer thicknesses
- Species volume fractions

Calculate SEI mass and layer SLDs

Compare SEI properties to data

- Screen out “bad” masses
- Calculate probability of observing SLDs
- Metropolis-Hastings algorithm

Accumulate population of “acceptable” models.



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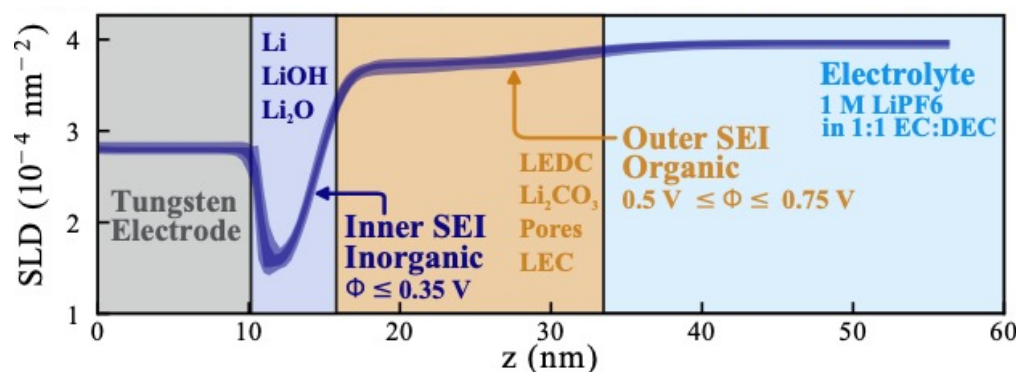
Lee, et al., J. Power Sources, 2019



Conclusions and Outlook



- **Direct, operando observation of two-layer SEI on non-intercalating anode.**
 - Thin, dense inner layer: inorganic, Li-rich compounds
 - Thicker, porous outer layer: organic compounds
 - Outlook: Li-active substrates, interrogate SEI improvement strategies.
- **EQCM-D provides insights into formation mechanisms**
 - Organics deposit at higher potentials; secondary rxns form inorganics at lower potential.
 - Outlook: Operando data as a platform for mode/mechanism validation



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Preliminary Numerical Simulations

