Conductive Polymers for Bidirectional Neural Interfaces:**Fundamentals Aspects and in Vivo Applications**

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Abstract

Next-generation neural interfaces for bidirectional communication with the central nervous system (CNS) aim to achieve the intimate integration with the neural tissue with minimal neuroinflammatory response, high spatio-temporal resolution, very high sensitivity and readout stability. The design and manufacturing of devices for low power/low noise neural recording and safe and energy-efficient stimulation that are, at the same time, conformable to the brain, with matched mechanical properties and biocompatibility, is a convergence area of research where neuroscientists, materials scientists and nanotechnologists operate synergically. The biotic-abiotic neural interface, however, remains a formidable challenge that prompts for new materials platforms and innovation in device layouts. Conductive polymers (CP) are attractive materials to be interfaced with the neural tissue and to be used as sensing/stimulating electrodes because of their mixed ionic-electronic conductivity, their low contact impedance, high charge storage capacitance, chemical versatility and biocompatibility. This manuscript reviews the state-of-the-art of CP-based neural probes for extracellular recording and stimulation, focusing on those technological approaches that were successfully demonstrated in vivo. The aim is to highlight the most reliable and ready-for-clinical-use solutions, in terms of materials technology and recording performance, other than spot major limitations and identify future trends in this field.

1. Neural interfaces for extracellular recording and stimulation

One of the main technological challenges is the design of brain interfaces suitable to establish an effective bidirectional communication with the central nervous system (CNS) by enabling *in situ* transduction, modulation, and decoding of neural signals in a unique, not distributed, “resident” volume of the brain^[1–6]. Reproducing these functions with a device would dramatically change the way we conceive brain-machine interfaces aimed at improving patients’ quality of life, paving the way for novel therapeutic routes based on controlled stimulation, inhibition, repair or replacement of neural circuits^[7]. Different methods are available for recording and stimulating neuronal activity, which differ in terms of spatial/temporal resolution of the signal and device invasiveness (**Fig. 1a**).

Figure 1. Extracellular recording methods. a) Different brain signals can be recorded through non-invasive or invasive approaches. EEG signals are recorded from the scalp; LFP signals can

be recorded from μ -ECoG arrays placed on the cortex surface. b) MuSa array intended for human studies (courtesy of Prof. Thomas Stieglitz, University of Freiburg). Single-unit spikes are commonly recorded by penetrating probes such as c) microelectrode arrays (Utah array), d) single-shank (modified and reproduced under terms of the CC-BY license, Copyright 2017, published by Springer Nature^[8]) or e) injectable nanomesh devices, reproduced with permission, Copyright 2018, John Wiley and Sons.^[9]

1.1. Brain recording

Functionally relevant information can be obtained by means of large electrodes placed on the scalp, resulting in a non-invasive technique called electroencephalography (EEG) (**Table 1**). However, EEG (but the same applies to magnetoencephalography, MEG) provides rather weak and noisy electrical signals that generally encompass the activity of extended brain areas^[10] or reflect functional brain networks^[11]. The detection with EEG of high frequency activity (>70 Hz) is also hampered by the attenuation/filtering phenomena produced by the scalp and the skull, interposed between the electrodes and the brain. Therefore, more invasive devices such as ultra-conformable electrocorticography (ECoG) and μ -ECoG (**Fig. 1b**) arrays, placed on the surface of the CNS, or intracortical probes (**Fig. 1c-e**) should be adopted^[12–15]. Indeed, extracellular electrodes are the most viable interfaces to record LFPs at high spatial-temporal resolution and high signal-to-noise ratio^[16], action potentials (spikes) from a population of nearest neurons ("multi-unit" recording), and in some instances also from single neurons (single-unit activity)^[11,17,18]. Spikes are usually recorded by the means of penetrating probes; however, it is also possible to detect them also by the surface of the cortex using ultra-conformable μ -ECoG arrays, as it has been recently demonstrated.^[19–21] In addition, integrated stimulation allows to assess anatomical and functional connectivity at diverse scales of interactions^[22,23].

Table 1. Clinical applications of brain recording techniques

Site	Form	Material	Contact Size	Recorded signals	Recording sample	Clinical application
Scalp	Cup electrodes	Ag/AgCl	10 mm / 314 mm ² / 3.1 cm ²	EEG		Diagnosis of epilepsy and of regional dysfunctions [24]
Sub-cutaneous	Needle electrodes	Pt-Ir	35 mm ²	EEG		Long-term monitoring of seizures and interictal epileptic activity [25]
Subdural	Grid electrodes Stripe electrodes	Pt-Ir / Steel	4.15 mm ²	LFP		Presurgical mapping and localization of the epileptogenic region [26]
Intra-cerebral	Cylinder-shaped multicontact electrodes	Pt-Ir / Steel	5.0 mm ² / 8.1 mm ²	LFP		Presurgical localization of the epileptogenic region [27]
	Hybrid electrodes with microwires	Pt-Ir	wire: 3000 μm ³	LFP, unit activity		Studies in cognition [18] and neuro-physiology [26,27]
	Hybrid electrodes with shafts tetrodes	Pt-Ir		LFP, unit activity		

1.2 Neural stimulation

1.2.1. Stimulation and charge balance

Neural stimulation means the injection of current into tissue with the purpose of triggering a specific neural response. When the stimulus reaches over a certain threshold, the impulse can trigger action potentials in neurons nearby, meaning the artificially generated stimulation is translated to a real neural signal in the nervous system. If the electrodes are sufficiently small

and close enough to the target structure, it is possible to selectively stimulate even down to the level of a single neurons in a system. The typical stimulus is a square current pulse of a few hundred μ -seconds duration or, using common terminology, the pulse would be one “phase” of the stimulation. The threshold necessary to elicit an action potential would in this case be defined by the current amplitude. As the threshold, in turn, is linked to the duration of the pulse, so that longer pulses require lower current, it is common to note charge/phase instead of current to define the stimulation. In order to prevent build-up of electrochemical reactions by-products at the interface, it is common to stimulate in bi-phasic manner, meaning the depolarizing impulse (usually cathodic) is immediately followed by the opposite pulse, retrieving the injected charge. The net-charge transferred over such a bi-phasic pulse will be zero, yet the outcome on the stimulated neural structure will be evoking an action potential. Stimulation pulse parameters vary depending on the adopted electrode materials^[28] and must be safe with respect to the electrodes as well as the neural target tissue^[29]. Simple rules of thumb exist to limit biological damage^[30], however electrochemical and biological processes are highly complex and limits should be interpreted rather as an educated guess than as hard numbers.

1.2.2. Clinical neurostimulation

In addition to action potentials, stimulation can be used to influence the state of the whole system. Such neuromodulation typically employs signals with slower dynamics than the bi-phasic pulses described above, as the intention is not eliciting action potentials in single neurons, instead to influence the network on system level (**Table 2**). Common terms to describe the signals are alternating current stimulation (ACS), typically a sinusoidal signal in the 0.1-200 Hz range, and direct current stimulation (DCS). For the latter, the DC stimulation is commonly “dosed” in asymmetric pulses, but with pulse lengths closer to the seconds to minute range, than the sub-millisecond. It is important to note that even if this broad variety of pulse types, from biphasic to DC, is clinically relevant and grouped under the general term

“neurostimulation”, these pulses pose very different challenges from the perspective of the electrode material. Short bi-phasic pulses have the advantage that it is possible to inject charge mainly via capacitive mechanisms, meaning the reaction kinetics is perfectly symmetric in the anodic and cathodic phase. Longer pulses are only possible by engaging additional faradaic charge injection mechanisms on the surface, and therefore put the electrodes at higher risk for corrosion or similar electrochemically driven degradation.

Table 2. Clinical applications of stimulating electrodes

Stimulation mode	Pulse type & parameters	Localization	Mechanism of action	Clinical application
DC stimulation	DC/pseudo DC	Transcranial 1-4 mA, 20 min/day	Neuronal de-/hyperpolarization	Cognitive enhancement [31] Depression [32] Focal epilepsy ^[33,34]
Low frequency stimulation (<10 Hz)		Transcranial 0.5-10 Hz, 1-4 mA	Induction of longterm depression (LTD)	Focal epilepsy (EASEE-Device) ^[34]
		Subortical 2-4 Hz		Focal epilepsy ^[35,36]
High frequency stimulation (>100 Hz)		Transcranial 100-130 Hz, 1-12 mA	Activation of hypoactive brain areas Interference with ictal pattern generation	Depression [32] Focal epilepsy ^[34]
		Epicortical 30-90 Hz 100/200 Hz	Gating modulation Interference with ictal	Pain ^[37] Focal epilepsy

		Amplitude 80% motor threshold	pattern generation	(RNS- Device ^[38])
		Intracortical 30-40 Hz/ 130 Hz, 2-5V	Targeted neurotransmitter release Reduction of network recruitability / Interference with ictal pattern generation	Pain ^[37] Psychiatric disorders (Obsessional- compulsive disorders, depression) ^[39] Focal epilepsy (Thalamic stimulation ^[40] , focus stimulation ^[38])
Patterned stimulation (e.g. theta burst stimulation)		Transcranial theta 5Hz, amplitude: 80% motor threshold Intracortical Theta 4-5 Hz, 4-5 mA	Activation of physiological patterns in memory formation and spatial navigation	Cognitive enhancement ^[41,42]

1.3. Main issues of implantable probes

A major issue with invasive probes (more relevant for intracortical probes) is the deterioration with time of the recording/stimulation performance due to the occurring of the “foreign body reaction” (FBR)^[43–45], eventually leading to the encapsulation of the probe by an insulating layer mainly composed of activated glia and astrocyte cells (see **Box 1**). The formation of the so-called “glial scar”, together with concurrent reduction in the number of neurons in the proximity of the probe (within a radius of ca. 100 μm around the probe)^[43] due to cell death or migration, deplete the amplitude of the recorded signals within a few weeks^[46,47]. Apart from biotic factors, the quality of recording can be compromised also by abiotic mechanisms. These include cyclic stresses applied to the brain tissue due to shear micro-motion at the interface between the probe or the (stiff) connector and the (soft) brain tissue due to brain pulsations and

micro-movements^[48], inability of the probe to conform to the cortex surface (in particular for epicortical or subdural devices) and delamination or chemical leaching from the electrode surface especially when coatings are applied^[8,49]. In order to limit the impact of the FBR and maintain stable, high quality chronic recording and stimulation, technological advances are oriented towards the manufacturing of highly-flexible compliant devices^[14] and minimally-invasive probes such as penetrating carbon microfibers^[50] or injectable nanomesh probes^[51,52] (**Fig. 1e**). In this regard, it was demonstrated that implants with lateral size < 10 micron (i.e. corresponding to the average size of a neuron soma) cause an almost negligible inflammatory response, enabling stable high-quality recording in chronic scenarios^[53]. This can be ascribed to reduced tissue displacement (intracortical probes) or depression (epicortical probes), limited blood-brain-barrier (BBB) disruption and vasogenic edema, and minimally impaired transport of signaling molecules^[53]. Besides, in order to minimize the mechanical mismatch at the nervous tissue/neural probe interface, materials research is devoted to the design of neural probes using relatively soft (polyimide PI, parylene PAR), soft (e.g. polydimethylsiloxane PDMS) and ultra-soft (i.e. hydrogels) materials, showing Young's modulus from ca. 6 orders of magnitude larger to similar values to that of the brain tissue (0.01-10 kPa), respectively^[54,55]. However, it is worthwhile reminding that the combination of modulus and probe geometry/cross-section (which is the device's moment of inertia) confers high compliance to the probe. It is indeed possible to create highly flexible probes with relatively stiff materials showing bending stiffness comparable to that of brain tissue (0.1–100 pN·m)^[14] leading to only minor chronic tissue shears and negligible upregulation of inflammatory cytokines^[51,56]. These considerations are especially useful when considering conjugated materials and specifically conductive polymers, as it will be discussed in detail below.

Box 1. Neural tissue reaction to probe implantation

Breaching of the blood-brain-barrier (BBB) and of cell membrane, mechanical strain and inflammation upon probe implantation, are regarded as the first steps of the biochemical signaling cascade of events targeting a variety of cell types (**Figure B1**). The inflammatory cascade leads to acute and later on chronic host tissue response, that eventually lowers the quality of recording, and may cause implant mobilization and failure^[56].

Figure B1. Invasive devices elicit a localized immune response. a) Cellular components in healthy brain tissue. Axons signal both oligodendrocytes and NG2 glia and neuronal modulation via glutamatergic and GABAergic neurotransmitter release. b) Acute injury after device insertion. Disrupted blood vessels leak inflammatory factors. Microglia, astrocytes, and NG2 glia become activated and adhere to the surface of the electrode. c) Chronic immune response to implanted devices. Glial cells form a chemical and mechanical barrier around the device, preventing the transmission of ions, charged solutes, and neurochemical signals, compromising the quality of signal recording. Reprinted with permission from ^[57], Copyright 2017, American Chemical Society.

Upon BBB rupture, blood-serum proteins (such as albumin, globulins, fibrin/fibrinogen), blood cells, pro-inflammatory molecules and cytokines are spilled in the brain parenchyma around the probe, where they activate the inflammatory pathways of nearby microglia and

astrocytes. Activated microglia and monocytes start to encapsulate implant with lamellipodia within the first hour from implantation; during the next 12 hours activated microglia become motile and move towards the injury site where they complete the formation of a thin encapsulation sheath^[56,58]. Simultaneously, the upregulation of pro-inflammatory cytokines drives nearby neurons towards excitotoxicity and neurodegeneration. Tight junctions among glial cells limit ionic exchange with the electrode surface, giving rise to the dramatic increase of the electrochemical impedance usually observed during the first week from implantation^[59]. Astrocytes, the most abundant cells in brain providing metabolic support to neurons and guaranteeing processing and transfer of synaptic information across neuronal networks^[60], are also highly activated during the first week and give rise to a compact layer around the implant during week 2 and 3^[50], further worsening the recording performance. Other than microglia and astrocytes, non-neuronal cells such as pericytes, NG2-glia (oligodendrocyte precursors), monocytes and leukocytes (infiltrating upon BBB disruption) were demonstrated to participate to the gliosis formation process^[58,61,62]. The expression of cytokines and chemokines is a hallmark of chronic inflammatory response. Cytokine production can begin to occur as early as day 1; for example, acute inflammation at 0–3 days involves TNF α , IL-1 β , and IL-6 while early chronic tissue response (3–7 days) may involve IL-10, TGF- β , and PDGF. The gliosis process is generally assumed to reach a steady state three to six months post-implantation, greatly affecting both extracellular recording and neurostimulation quality and efficacy^[54,63].

1.4. Towards high spatio-temporal resolution

Microelectrode miniaturization and fabrication of high-density micro-electrocorticography (μ -ECoG) arrays allow neuroscientists to record with unprecedented spatial and temporal resolution the orchestrated/synchronized activity underlying neural networks functions as well

as to precisely stimulate selected neuronal ensembles^[64–66]. Nevertheless, smaller microelectrodes exhibit higher electrolyte resistance and lower double layer capacitance compared to larger microelectrodes, which results in higher impedance^[28,53]. Since thermal noise, one of the main sources of noise during recording, scales with the electrode impedance (see **Box 2**), the output SNR is expected to be reduced upon decreasing the electrode size^[67,68]. However, signal loss through shunt pathways as well as electronic noise due to the amplifier and flicker noise may greatly affect the SNR in vivo under certain circumstances and recording setups^[69]. Other issues that arise as a consequence of electrode miniaturization might be the electrical coupling between adjacent lines (crosstalk)^[70], the chemical instability of the electrode surface and integration hurdles due to high density small connectors and leads^[71,72]. In view of these arguments, efforts are devoted to the modification/functionalization of the microelectrode surface with the aim to lower the overall impedance and achieve high-quality recordings especially in case of high-density small recording sites. In this framework, one of the most investigated approaches is to increase the roughness of the electrode surface, to increase its electroactive area and lower its impedance. In this way, flake-like gold or nanostructured platinum electrodes were realized, showing order of magnitude lower impedance values than pristine smooth electrodes^[73–75]. Alternatively, smooth or porous iron oxide layers can be grown on the electrode surface by electrochemical oxidation or sputtering, respectively^[76]. However, activated metal oxide layers can change their chemistry and passivate with time in chronic experiments, leading to the deterioration of recording capability over time^[28]. In addition, the release of potentially harmful metal ions in the peri-implant environment by electrochemical processes has not been deeply investigated so far ^[46,49,77].

In this complex scenario, where mechanical, chemical, electrochemical and biological aspects are intertwined, the choice of materials for fabricating the electrodes is extremely challenging because it is unlikely that one material encompasses all desirable properties. In the past two

decades, conductive polymers (CPs) emerged as an attractive class of materials due to their highly porous structure contributing a large electroactive area, low impedance, mixed ionic-electronic conduction, high charge storage capacitance (CSC), other biocompatibility and possibility of bio-functionalization. Therefore, a great deal of interest has been generated around CPs in the neuroelectronics community, driven by the prospect of achieving high performance probes for bidirectional neural interfaces.

2. Conductive polymers

2.1. Properties and advantages of conductive polymers

Conductive polymers (CPs), represent a versatile class of materials suitable for a wide range of applications, from microelectronics industry (including solar cells, antistatic coatings, energy storage technology)^[78,79] up to biomedical applications^[80]. CPs are conjugated polymers that upon doping exhibit conductivity exceeding 1000 S/cm, i.e. comparable to that of metals or highly doped inorganic semiconductors (**Fig. 2**).

Figure 2. Organic (semi-)conductors for neural applications. In the top left, polyaniline (PA), polypyrrole (PPy) and polythiophene (PT), three of the most common CPs for neural applications and progenitors of state-of-the-art materials. Modern derivatives of polythiophene are also shown, including the widely investigated PEDOT:PSS (bottom, blue

background) and a family of variously engineered polythiophenes (center/right, green background).

They are characterized by a system of extensively delocalized π -molecular orbitals, and due to delocalization, charge-separated states of these molecules (i.e. their oxidized or reduced form) are stable and long-living. These features make the charged states to behave as “quasi-particles”, as they possess the ability to drift and diffuse across the material. For these reasons, these charged states are also termed charge-carriers. CPs are obtained via oxidative coupling of the monomers either by chemical synthesis (condensation or addition polymerization reactions), electrochemical polymerization or photo-induced oxidation^[81]. As in organic semiconductors, charge transport in CPs occurs via hopping between localized Wannier states, conversely to what happens in metals, in which electrons are free to occupy any position in the conduction band. The term “doping”, lexically borrowed by the semiconductor field, refers to the generation of charge carriers in the CPs, either via injection of extra electrons (i.e. reduction or n-type doping) or removal of electrons, creating so-called “holes” (i.e. oxidation or p-type doping) that can be transported across the conjugated polymer backbone upon application of an external bias field. In the simplest and most common scenario, i.e. conduction in a p-type CP film between two electrodes upon application of a bias, holes are injected (electrons are removed) at the positive electrode and are collected at the negative one, in a cascade of localized oxidation reactions that proceed from the positive to the negative electrode. Doping can be performed either chemically or electrochemically^[82] during CP synthesis, generating charge carriers that are usually in the form of polarons (i.e. radical anions/cations) or bipolarons (i.e. dianions/dications). During device operation it is possible to electrostatically control the amount of mobile charge carriers, allowing switching between insulating and conductive regimes^[83]. The conductive properties of CPs depend on many intrinsic and extrinsic parameters such as charge carrier density; delocalization lengthscale of π -electrons and

structural defects; intra-and inter-chain interactions; interaction of the charge carriers with other charges (e.g. (counter)ions, static charge) and dipoles (either from polarization of the material, water, solvent) or reactive species (oxygen, peroxides, superoxides); morphology, crystallinity, synthesis and processing parameters^[84–87]. The high chemical versatility, the compliant mechanical properties, the compatibility with a large variety of cell type, the low impedance over a wider frequency bandwidth compared to metal or metal oxide based coatings, make CPs ideal materials to boost the recording performance of conventional metal-based neural probes **(Box 2)**^[88–91]. A particularly appealing feature of CPs for neural stimulation and recording is the mixed electronic and ionic transport exhibited by several CPs^[92]. Mixed transport in CPs is leveraged in a plethora of applications including electrolyte-gated organic devices such as organic electrochemical transistor (OECT)^[93–95] and organic electronic ion pumps^[80,96]. It is also worthwhile mentioning that several CPs, due to their porous structure and high ionic conductivity, commonly exhibit higher charge storage capacity (CSC) and charge injection capacitance limit (CIL) values compared to metal and metal oxide electrodes^[89], which is a desired characteristic in view of neural stimulation applications^[28,97]. Nevertheless, poor adhesion to noble metals like Pt, PtIr, Au, swelling and conductivity drifting especially under cyclic stress conditions, strongly limited the application of CPs for chronic applications to date^[98,99]. Noteworthy, it has been recently reported the possibility to guarantee durable adhesion by modifying the surface of the metal electrodes, e.g. by increasing the roughness as in the case of nanostructured Pt or gold electrodes, or by growing a adhesion promoter layer of porous IrOx, capable to establish a stronger bonding with the CP, via formation of a carbide bond^[74,76]. Furthermore, as stability of the tissue-electrode interface can be improved by lowering the mechanical mismatch between the electrode material and the neural tissue, soft three-dimensional (3D) hydrogels based on conducting polymers have been recently proposed and tested both in vitro and in vivo^[100–102].

2.2 PEDOT:PSS: the workhorse material for bioelectronics

There is a wide library of CPs nowadays available, however the most used CP is PEDOT:PSS, a polymer blend of poly(3,4-ethylenedioxythiophene)(PEDOT) and poly-(styrene sulfonate) (PSS). The oxidation and polymerization of the EDOT monomer in the presence of the negatively charged PSS counter ion results in an intrinsically highly doped p-type semiconductor, which behaves as a moderate conductive material. ($< 1 \text{ S/cm}$)^[103]. Upon mixing with small molecules such as ethylene glycol or dimethyl sulfoxide (known as secondary dopants)^[104] or treatment with strong acidic solution, PEDOT:PSS increases its conductivity up to 10^3 - 10^4 S/cm ^[105,106]. Because of its mixed ionic/electronic conduction mode, PEDOT:PSS is the first choice as the signal transducing component in bioelectronics devices operated in aqueous electrolytes or hydrogels such as biosensors and transducers based on OECTs, low-impedance multi-electrode arrays (MEAs) interfaced with the nervous system, up to neuromorphic logic-based devices and sensors^[107–111]. The mixed ionic/electronic conductivity in PEDOT:PSS results from the presence of two physically distinct, albeit intimately interpenetrated, phases, one acting as hole-transport region (PEDOT-rich phase) and the other as ion transport region (PSS-rich phase) (**Figure 3a**). The lengthscale of multiscale phase separation is from nanometer to mesoscopic length scales and can be observed by atomic force microscopy (**Fig. 3b**)^[112]. The consequent formation of the electrical double layer at the interface between these two domains gives rise to the large effective capacitance that is one of the most desirable properties of PEDOT:PSS. The capacitance of PEDOT:PSS is linearly correlated to its volume for relatively small amount of polymer, typical of those of interest in electrolyte-gated organic electrochemical transistors (EGOTs) and microelectrodes^[113,114]. Deviations from this trend at larger polymer volumes were however reported, due for instance to incomplete film hydration^[115] and saturation of the electroactive area^{[116][116–120]}. When synthesized via chemical route, PEDOT⁺ and PSS⁻ form polymeric particles where positively

charged PEDOT chains are surrounded by poly-anionic PSS shells, the latter providing sufficient stability for PEDOT⁺ chains to be dispersed in water^[121,122]. Commercial water-dispersed PEDOT:PSS formulations offer the great advantage of being a commodity as they allow the easy manufacturing by means of low-cost additive techniques for a plethora of applications^[8,121,123], of antistatic coatings^[124] and devices such as transistor^[125,126], memristors^[127], solar cells^[128,129], OLEDs^[130], supercapacitors^[131] and artificial synapses^[94,111]. Besides, highly conductive PEDOT/PSS films can be grown through the oxidative electrochemical deposition directly on the top of dense neuron-sized microelectrodes arrays, thus allowing coating in a mask-less process^[14].

Figure 2. PEDOT:PSS nano-morphology. (a) 2D model encompassing two distinct yet highly interconnected phases consisting of PEDOT-rich and PSS-rich domains, as proposed by Volkov and co.^[112]. Reproduced with permission, Copyright 2017, John Wiley and Sons. (b) Atomic force microscopy (AFM) topography (i) and phase images of spin-cast PEDOT:PSS thin films. Scale bar is 100 nm. Darker areas in the phase image (small phase angle deviation) can be associated to soft hygroscopic PSS-rich areas, whereas brighter areas (larger phase angle deviation) can be associated to “harder” PEDOT-rich areas (authors’ unpublished data; scale bar is 100 nm).

Box 2. The electrochemical tissue-electrode interface

Electrochemical impedance spectroscopy (EIS) represents one of the most important tool of analysis since it is known that neural biological activity exhibits a characteristic frequency of 1 kHz thus, the impedance values at 1 kHz are often related to the high quality of a neural electrode.^[28] Nevertheless, impedance data are preferably collected over a wide range of frequencies, typically from 10 kHz down to below 1 Hz, in order to gain insights also on the charge transfer dynamics along the conductive films due to the diffusion of ions.^[97] In general, a dramatic decrease of the impedance, especially over the low frequency domain, represents the typical footprint of a conductive polymer film efficiently deposited on a metal-based electrode. This is due to the fact that CPs exhibit both electronic as well as ionic conductance, with respect to metal based electrodes, thereby enabling a dramatic reduction of the imaginary part of the impedance Z_I , which can be expressed as: $Z_I = -1/(2\pi \cdot f \cdot C_{dl})$, where f is the frequency and C_{dl} is the double layer capacitance. One of the first studies of PEDOT/PSS electrodeposited on neural microelectrodes was reported by D. Martin et al. in 2003.^[132] They showed how PEDOT:PSS coating can decrease the impedance modulus $|Z|$ by almost two orders of magnitude when compared to the uncoated metal electrodes. It was also reported that small gold microelectrodes (15 μm in diameter) coated with PEDOT:PSS were able to record higher-quality neural activity and exhibited a much higher signal-to-noise ratio (SNR) than uncoated Au electrodes.^[133] To understand the effect of PEDOT:PSS on the SNR, a brief digression on the possible sources of (biological and non-biological) noise is necessary. Non-biological sources include electronic noise from the amplifying circuit, the injection noise at the electrolyte-electrode interface, and the thermal noise, which is the major contribution correlated to the electrochemical properties of the coating. Thermal noise (V_{rms}^{th}) is directly related to the real part of the impedance $Z_R = \text{Re}\{Z\}$ of the recording microelectrode and can be expressed as:

$$V_{rms}^{th} = \sqrt{4k_b T Z_R \Delta f} \quad (1)$$

where k_b is Boltzmann constant, T is the absolute temperature, and Δf is the recording bandwidth. To explain the role of the electrode interface on Z_R , we discuss the widely adopted equivalent circuit consisting of the solution resistance R_s connected in series with an electronic capacitance C_{el} , which is linked to the space charge of conductive film, and a finite-length Warburg impedance Z_D , as depicted in **Figure B2** (model A)^[132,134]

The Warburg impedance Z_D accounts for ion diffusion through PEDOT:PSS pores and can be expressed by:

$$Z_D = R_D \frac{\coth \sqrt{j 2 \pi f \tau_D}}{\sqrt{j 2 \pi f \tau_D}} \quad (2)$$

where R_D is the diffusion resistance, j is the imaginary unit ($-j^2 = 1$), and τ_D is the ion diffusion characteristic time.^[135] The Z_D element is related to the diffusional pseudo-capacitance C_D , according to the formula $\tau_D = R_D C_D$. The leading terms in the expansion of eq. 2 for $f \tau_D \rightarrow 0$ are $\text{Re}\{Z_D\} = R_D/3$ and $\text{Im}\{Z_D\} = -1/(2\pi \cdot f \cdot C_D)$. In general, the $\text{Re}\{Z\}$ is frequency independent in the case of PEDOT/PSS coatings, unlike the uncoated and flat electrodes^[28], and the resulting impedance plot is dominated by a resistive behavior in the high frequencies, and a capacitance contribution in the middle to low frequency domain. In addition, larger electroactive surface area (ESA), if compared to flat and non-porous metal electrodes, reduces $R_s = \rho l/\text{ESA}$ where ρ is the solution resistivity and l is the electrode distance^[73]. Thus, altogether this results in lower $\text{Re}\{Z\}$ values for PEDOT/PSS coated microelectrodes, so it is advantageous to produce a decrease in thermal noise. As far as the capacitance of PEDOT/PSS is concerned, according to model A of Figure B2, it can be expressed as $C_{dl} = [1/C_D + 1/C_{el}]^{-1}$. Its porous structure enables fast and efficient ion diffusion inside the material, thus yielding larger C_{dl} if compared to flat metal electrodes. This improvement results highly effective in reducing the dangerous polarization voltages which are typically

reached upon fast current pulses adopted during neural stimulating protocols. When PEDOT:PSS coated microelectrodes are used for in vivo experiments, a further element should be added to the above mentioned circuit model. This is due to the intimate interaction between the neural microelectrode and the surrounding inflamed neural tissue which gives rise to tissue encapsulation and gliosis. For example, the circuit model adopted by Cui et al. (Figure B2, model B) to describe the impedance traces for in vivo electrochemical analysis with PEDOT coated neural microelectrodes, consisted of a solution resistance in series with the above described Warburg element and a charge transfer element (a charge transfer resistance R_{CT} in parallel with a constant phase element CPE). Interestingly, the magnitude of the charge transfer resistance can be used as a parameter to describe the extent of the neural probe encapsulated by the surrounding tissue^[136].

Figure B2. Typical equivalent circuit models adopted to describe the electrochemical impedance interface for PEDOT/PSS coated microelectrodes during in vitro (model A), or in vivo (model B) studies.

To sum up, model A represents the typical scenario that occurs for in vitro studies, when the relatively simple interface of PEDOT:PSS||electrolyte can be fitted with the equivalent RCT. Besides, model B describes the typical situation for an in vivo experimental EIS, where PEDOT/PSS interacts non only with a water based electrolyte solution but also with a surrounding neural tissue, which is also biologically reactive.

2.3 Beyond PEDOT:PSS

Despite the main advantages and interesting features exhibited by PEDOT:PSS, some shortcomings have been reported especially with regard to long-term in vivo applications, including limited adhesion, swelling and degradation by hydrolysis when exposed to physiological oxidants^[137,138]. To overcome these problems, composites with carbon nanotubes (CNTs) have been proposed during the last decade, by either doping PEDOT with CNTs during electrodeposition or directly mixed with PEDOT:PSS. The addition of CNTs proved to be a promising approach to prevent material spallation and cracking within the PEDOT:PSS coatings as well as PEDOT adhesion during chronic recording and stimulation experiments. At the same time, the excellent electrical properties allow better electrostimulation performance compared to bare PEDOT (see *section 4.1*). Another promising strategy to increase the adhesion of PEDOT coating is based on the physical/chemical modification of the metal or metal oxide underlying electrode surface (see *section 3.2*).

In addition to mechanical issues, the limited charge injection, drift instability and the high affinity of PEDOT:PSS for divalent cations (which may hinder interactions with other ionic species in extracellular and cerebrospinal fluids, thus limiting the possibility to modulate the input signals from the brain), has led to an extensive exploration of alternative dopants to PSS. To date, several categories of anionic dopants have been investigated, ranging from small ions (e.g. perchlorate or tetrafluoroborate), to acidic molecules (e.g. p-toluenesulfonic acid, pTS), corticoids (e.g. Dexamethasone) and polymers (e.g. Nafion). Examples of application of PEDOT:X for recording and stimulation experiments are given in *section 4.2*.

Noteworthy, alternative materials to PEDOT are currently being investigated for neuroelectronic applications, even not directly as electrode material. For example, 2,7-dioctyl[1]benzothieno[3,2-b][1]benzothiophene (C8-BTBT-C8) has been used to construct synaptic transistors^[139,140] and a light-stimulated devices with both memory and learning

capability^[141]. Light-stimulated writing/reading neuromorphic memory has also been proposed in light-gated organic bio-hybrid transistors (LEGOT) based on TIPS-pentacene^[142]. The fused thiophene diketopyrrolopyrrole (FT4-DPP)–based conjugated polymer showed small hysteresis whereas the well-known poly(3-hexylthiophene-2,5-diyl) (P3HT) produced large hysteresis and relatively long memory retention^[143]. Improved retention time has also been observed with organic semiconductors in which oligoethylene side chains have been inserted for cation coordination^[144]. In fact, n- and p-type conjugated polymers with oligoethylene glycol side chains appear to be extremely interesting as the electronic/ionic conductivity, specificity towards ions and stiffness can be tuned by appropriately modifying the side chain length and grafting density^[144]. Furthermore, the incorporation of bioactive functional groups can easily allow bio-recognition towards molecular signals of interest (stress, inflammation) and its transduction into the electronic properties of the active material^[145].

3. Neural interfaces based on PEDOT:PSS

3.1. Extracellular recording

3.1.1. μ ECoG arrays

Highly conformable μ ECoG arrays featuring PEDOT:PSS-coated Au microelectrodes were reported by Khodagholy and coworkers in 2011^[146]. Due to intrinsic lower impedance, PEDOT:PSS coated microelectrodes outperformed Au microelectrodes in the recording of bicuculline-triggered sharp-wave events (i.e. mimicking epileptic spikes). Interestingly, PEDOT-coated electrodes provided similar recording quality as standard intracortical silicon probes featuring Ir microelectrodes. A few years later, the same research group developed the “Neurogrid”, a flexible array of neuron-size density PEDOT:PSS coated microelectrodes^[20]. Neurogrid enabled mapping of action potentials from the surface either of the cortex or the hippocampus in freely moving rats^[20] as well as local field potentials (LFPs) and action

potential waveforms of individual neurons (spikes) from the cortical surface of human subjects undergoing epilepsy surgery (**Fig. 4a, b**)^[21].

Figure 4. μ ECoG arrays based on PEDOT:PSS. (a) Photomicrograph of 240-channel NeuroGrid device (left, scale bar is 1 mm); magnified microscope image of 10 mm² electrodes arranged in 2 x 2 tetrodes, (top-right, scale bar is 1 mm); high-density patterning of conducting polymer-coated electrodes, comprising a single tetrode, (bottom-right, scale bar is 20 mm). Reprinted with permission from^[21] (Copyright 2016, American Association for the Advancement of Science). (b) Neurogrid device placed on the surface of rat somatosensory cortex. Reproduced with permission^[147]. Copyright 2016, John Wiley and Sons. c) Schematic representation of the fabrication and assembly of a 32 channel MuSA array featuring 10 μ m and 100 μ m recording sites made of Pt, IrOx and PEDOT:PSS coatings on the metal electrode(i); MuSA device wrapped around a pipette (ii); details of the electrode area, connection pads and interconnection tracks of the MuSA array (iii). Scale bars = 200 μ m. (d) conformable MuSA array on the rat cortex. Scale bar = 1 mm. Reprinted with permission from ref. ^[14]

Probe conformability to the surface of the brain is an essential prerequisite to obtain a stable and high quality recording^[14] and record both spikes as well as LFPs with μ ECoG grids^[19]. Our groups investigated the relationship among polyimide μ ECoG conformability, depression of

the rat cortex after implantation and impedance and recording quality in chronic scenario^[14]. Highly flexible Multi-Species Array (MuSA) implants of various thickness (4µm, 8µm, 12µm) and conformability, featuring both 10 and 100 µm size microelectrodes of different materials (Pt, IrOx and PEDOT:PSS) were designed and manufactured (**Figure 4c, d**). Despite showing similar values during the first weeks from implantation, the impedance of poorly conformable arrays began to significantly increase starting from week 3. Histological analysis revealed that the brain depression caused by poorly conformable arrays was significantly higher when compared to thinner devices. Noteworthy, the Peak/Mean Ratio (PMR) was higher for the most conformable probes, regardless of the electrode material^[14]. The possibility to record neural signals with high SNR by arrays of OECTs, was reported for the first time by Khodagholy and co. in 2013^[90]. The key concept for the adoption of OECTs in place of PEDOT:PSS coated electrodes is in the superlinear amplification of the voltage changes into the OECT channel current (see **Box 3**)^[148]. Highly-conformable ECoG arrays of PEDOT:PSS-based OECTs, implanted in the somatosensory cortex of epileptic rat models, enabled the detection of both bicuculline-induced and spontaneous spikes with larger SNR compared to PEDOT:PSS µECoG electrodes as well as intracortical Ir microelectrodes^[90].

Box 3. ECoG arrays based on organic transistors

To overcome the inherent limitation of higher noise subsequent to miniaturization, transistor-based architectures capable of providing *in situ* amplification of ECoG and µECoG signals represent an emerging alternative of extremely high interest. The key difference between a transistor and an electrode is that the former is an active device, forming a circuit that acts as a voltage-controlled current amplifier. A voltage-controlled current amplifier transforms voltage modulations at the gate electrode into current modulations at the drain electrode, while amplifying *in situ* the signal at the same time (**Fig. B3**).

Figure B3. The rationale behind the use of organic transistors for neural recording. a) electrical connection in an organic transistor; b) schematics and connections layout of EGOT devices; c) In situ vs. ex situ amplification by means of transistor-based neural probes. n is the biological noise, s is the neurophysiological signal, n_{line} is the line noise. Capital letters indicate the corresponding amplified signals.

When electrodes are used for in vivo recording, an ex situ amplification is required, which involves also the line noise amplification. Downstream amplification thus results in increased line noise which affects the signal-to-noise ratio. Instead when transistors are used, an in situ amplification of the neurophysiological signal and biological noise is carried out. The line noise undergoes no amplification in this case, but is simply transferred as it is. The result is a higher signal-to-noise ratio. Electrolyte-gated Organic Transistors (EGOTs), working either in depletion or in enhancement/accumulation mode, feature an organic semiconductive channel directly exposed to the electrolyte. PEDOT:PSS OECTs were used to transduce epileptiform activity in rats, providing increased SNR with respect to conventional surface microelectrodes^[90,126]. The high transconductance resulting from the mixed ionic-electronic

transport in organic semiconductors, the easy tunability of device properties and the possibility to work in the wet environment avoiding encapsulation layers and to fabricate large-area devices by printing and additive techniques, all make EGOT technology suitable for translation to neural recording application^[80].

Despite these promising results, the use of EGOTs for electrophysiological applications is still limited by two factors: i) EGOTs are operated in the common source configuration, which implies the application of a bias in the brain; ii) EGOT working principle implies that higher amplification gain results in decreased bandwidth, and vice versa. If the former issue can be circumvented by careful device design in order to achieve maximum gain at zero bias^[149] or by implementing alternative connection schemes, the latter is ubiquitous in EGOTs and hard to avoid with materials currently in use. Noteworthy, EGOT devices were demonstrated to be highly selective and ultra-sensitive biosensors for the detection and diagnosis of inflammatory cytokines, anti-drug antibodies, and neurotransmitters in complex matrices^[150–152]. In addition, organic devices operated in a two- or three-terminal architecture with voltage pulses at fixed amplitude and frequency produce an output current response with either increasing or decreasing amplitude depending on the frequency. These devices can also mimic the response of neuronal synapses, embodying in one device some of the functions of neural circuitry, such as short-term plasticity (STP), spike-timing-dependent plasticity (STDP), long-term potentiation (LTP), and long-term depression (LTD). These properties are attractive for developing artificial synapses in neuromorphic circuits on the one hand, and prosthetic integration into the brain circuitry on the other hand.^[94,117,153,154]. Therefore, EGOT-based arrays should be envisaged to be endowed with different functionalities, allowing the design of adaptive multimodal probes capable of on-site amplification, signal analysis (including data filtering, pattern recognition and artefact detection), drug release and sensing of molecules/proteins of interest.

3.1.2. Intracortical probes

In one of the early works regarding the use of CPs coating for intracortical probes, PEDOT:PSS was electrodeposited on penetrating Pt/Ir microwire arrays in order to perform chronic recording from rat somatosensory cortex^[89]. PEDOT-coated electrodes steadily showed lower impedance ($\sim 200 \Omega$) than bare ones, translating into a modest improvement in SNR ratio compared to bare microelectrodes. More recently, the use of highly-flexible nanomesh PEDOT:PSS arrays enabled the recording of spontaneous and evoked single-unit activity of neurons across layers of the mouse visual cortex^[155]. Due to the extremely softness of the parylene-based probe, a temporary stiffening coating was necessary to facilitate the insertion of the nanomesh arrays into the mouse brain. 3D-printing has been also investigated as a straightforward method to directly pattern PEDOT:PSS micro-electrodes on highly-compliant substrates such as polydimethylsiloxane (PDMS) (**Fig. 5a**)^[156]. The probe was implanted into the mouse dorsal hippocampus and it continuously recorded neural activities in freely moving mice, including LFPs and action potentials. Very recently, Cea and co. developed a penetrating array based on innovative electrochemical transistors, named enhancement-mode internal ion-gated organic electrochemical transistor (e-IGT), characterized by high hole mobility, large transconductance, large capacitance and long-term operational stability^[126]. Recording using a common-source configuration from the somatosensory cortex of a freely moving rat allowed them to detect with high fidelity LFP showing typical spectral features of non-rapid eye movement (NREM) sleep, rapid eye movement (REM) sleep and wakefulness (WAKE) as well as single-unit activity.

3.1.3. Multimodal probes

Additional insights about brain communication networks can be obtained by hybrid approaches where temporally-resolved electrical recording is coupled with spatially-resolved optical

methods. By leveraging the low impedance of multi-arrays of PEDOT:PSS electrodes and the optical transparency of a thin parylene substrate, Donahue and co. simultaneously recorded neural signals and two-photon calcium imaging from the same location in the cortex of epileptic rats (**Fig. 5b**)^[157].

Figure 5. 3D printed and multimodal intracortical probes based on PEDOT:PSS. a) Image of an all-PEDOT 3D-printed soft neural probe (i), scale bar is 1 mm; images of the implanted probe in mouse cortex (ii); LFPs and spike signals recorded under freely moving conditions (iii). Reproduced and modified under terms of the CC-BY license^[156]. Copyright 2020, published by Springer Nature. (b) Layout of the recording setup with 2P-compatible electrode array, showing the infrared laser beam passing through the organic electronic probe placed on the cortex under the objective (i); electrophysiological local field potential (LFP) recording (ii) and fluorescence variation arising from fluctuations in intracellular calcium concentration (iii). Reproduced and modified under terms of the CC-BY license^[157]. Copyright 2018, Donahue et al, published by Society for Neuroscience.

High correlation between the local variations of fluorescence, due to fluctuations of intracellular calcium concentrations, and the local electrophysiological activity (i.e. LFPs) was demonstrated (**Fig. 5e**). Transparent PEDOT-based nanomesh MEAs were also developed, allowing both wide-field epifluorescence and two-photon Ca^{2+} imaging of visual rat cortex. Visual-evoked potentials (VEPs) from multi-unit activity were detected; interestingly, no significant

inflammation of the cortex due to the MEA implantation was observed after 20 days^[158]. Ultra-flexible and transparent arrays of PEDOT: PSS based OECTs were also developed for high-resolution mapping μ ECoG signals evoked by blue laser light stimulation targeted on the rat's brain surface through a fiber-guided system^[159].

For clinical applications, the chance of simultaneously recording neural activity and performing on-demand drug release is highly attractive, especially for the treatment of drug-resistant neurological disorders such as epilepsy^[160,161] (see **Box 4**). In this view, efforts were made to integrate conformable ECoG devices with microfluidic-based ion pumps (μ FIP) that actuate electrophoretic drug delivery^[162,163]. The μ FIP device, endowed with small apertures coated with a PSS-based exchange membrane for the controlled release of the anti-epileptic drug (i.e. GABA) upon application of a small voltage, was flanked by 32 PEDOT:PSS-coated gold microelectrodes for *in situ* recording of signals with high SNR (**Fig. 5f**)^[163]. Seizure-like events, induced by local injection of 4-aminopyridine (4AP) into the hippocampus of anesthetized mice, triggered the immediate release of GABA, leading to full silencing of additional seizure events.

Box 4. Loco-regional drug delivery from CP coatings

Drug delivery from PEDOT is a loco regional approach aimed to reduce the inflammatory response of the surrounding tissue while preserving optimal electrochemical properties (see Box 1). A main advantage of using a drug delivery system based on PEDOT instead of the common microfluidics, for instance, relies on the fact that it enables release on-demand^[164–166]. Dexamethasone (Dex) and its water-soluble prodrug dexamethasone sodium phosphate (DEX-P) are extremely potent anti-inflammatory and immunosuppressive corticosteroids and were incorporated in neural probes coated with PEDOT to reduce the adverse reaction

of the surrounding tissue. For example, Martin and co-workers electrodeposited PEDOT on a PLGA layer charged with Dex. In this case, the release of the drug was controlled by actuation of PEDOT when subjected to external electrical stimulation^[167]. Nevertheless, the typical approach adopted to prepare PEDOT based drug delivery systems consists of the use of the negatively charged drugs, which act as counter ions and can be incorporated during the electrodeposition of the polymeric coating. This is summarized in **Figure B4**.

Figure B4. Current approaches for drug delivery from CPs. Incorporation of anionic drugs via a) electrostatic interaction or b) covalent immobilization. c) On-demand release of cationic drugs from organic electrochemical ion pumps. Reproduced with permission^[170] Copyright 2016, John Wiley and Sons. d) On-demand release of drugs using microfluidic ion pumps. Reproduced with permission^[163]. Copyright 2018, American Association for the Advancement of Science.

The negatively charged dexamethasone-phosphate (DEX-P), which is a pro-drug of Dex, has been extensively incorporated in PEDOT based neural implants^[136,168,169]. Cui group coated neural microelectrodes with PEDOT electrodeposited in conjunction of oxidized multiwalled

carbon nanotubes and Dex-P. These microelectrodes were used to for stimulation of dorsal root ganglions. They observed that the controlled release of the drug resulted in less neuronal death/damage if compared to uncoated electrodes^[136]. Following a similar strategy, the same group prepared intracortical neural probes which were tested both in vitro and in vivo, over a period of 11 days in animal models. They found that coated electrodes successfully recorded neural activity throughout the implantation period, setting the stage for the long-term evaluation of electrically controlled drug release coatings for neural interface applications^[136]. It was reported that active release of Dex-P from PEDOT can be controlled over a time frame of up to four weeks during in vitro experiments. Indeed, the active drug release should extend for several weeks after implantation in order to mitigate the inflammatory response in chronic neural experiments. In addition, a release of $0.5 \mu\text{g cm}^{-2}$ of Dex corresponds to a concentration of about $1 \mu\text{M}$ within a $500\text{-}\mu\text{m}$ radius from the neural microelectrode. This ensures an amount of drug higher than the therapeutic bioactive concentration of Dex, estimated in the range of $0.2 \mu\text{M}$ ^[171]. Recently, Boehler and co-workers demonstrated that the active release of Dex-P can reduce the inflammation around the implanted neural microelectrodes after 12 weeks from implantation. Authors suggested that the combination of flexible probe technology with anti-inflammatory releasing coatings should be further developed in the future as an effective approach to fabricate long-term stable chronic neural interfaces^[168]. Recently, tauroursodeoxycholic acid (TUDCA) was reported to exert neuroprotective function. In particular, this bile acid can reduce microglial and astrocyte cell activation which are normally induced by neural devices implantation^[172,173]. In light of this, TUDCA was incorporated in PEDOT to prepare an alternative drug delivery system for long term neural recording brain implants^[174]. A self adjusting drug delivery system was proposed by Carli and co-workers. In this study, Dex was

incorporated in PEDOT through covalent bonds, thereby enabling drug release based on chemical bond hydrolysis rather than electrochemical triggers as discussed above. The release of Dex was observed (in vitro) over a period of at least 3 weeks and was accelerated in the presence of hydrolytic enzymes, resembling the biochemical environment of inflamed tissues^[175]. Recently, new devices enabling on-demand drug delivery such as Organic Electrochemical Ion Pumps (OEIPs) and Microfluidic Ion Pumps (μ FIPs) were proposed. In some case these devices were incorporated in neural probes, allowing simultaneous recording and release of GABA on demand.^[162,163] Compared to the OEIPs, μ FIP significantly reduces the voltage required to deliver ions (drugs) and simplifies re-loading from the ion reservoir^[162,163]

Table 3. PEDOT:PSS-based neural probes for neural recording.

Ref	Year	PEDOT:PSS deposition	Probe substrate	Recording site size	Underlayer	In vitro EIS @ 1KHz (Ω)	In vivo	
							Model/ Chronic or Acute	Notes
ECoG recording								
[146]	2011	SC*	Par. C [#]	(20x20μm ²)	Au	-	Rat/A	Better peak definition of 1-10 and 30Hz bands compared to Au
[90]	2013	P	Par. C	- 12x12 μm ² - OECT channel: W: 15 μm, L: 6 μm	Au	-	Rat/A	SNR (dB): - Bicuculline- induced model 44 (OECT), 24.2 (microelectrode); - GAERS model (7-11Hz) 52.7 (OECT), 30.2 (microelectrode), 32.0 (Ir penetrating electrodes)- GAERS model (4-14Hz) 22.3 (OECT), 13.5 (microelectrode), 18.2 (Ir penetrating electrodes)
[20]	2015	SC	Par. C	10 μm ²	Pt/ Au	-	Rat/C	Stable SNR (6-9 dB) during all the experimental time in rats (10 days). Recording of LFPs, spiking activity of layer 1 interneurons, pyramidal cells and fast firing deep interneurons
[20]	2015	SC	Par. C	ME 10 μm ²	Pt/ Au	-	Human/A	Recording of cortical slow oscillation (beta-frequency) and spiking activity
[21]	2016	SC	Par. C	10 μm ²	Ti/ Pt/ Au	1·10 ⁴ - 1·10 ⁵	Human/A	High spatiotemporal resolution of LFPs and single spike activity

[159]	2017	SC	Par. C	OECT channel: W: 70 μm , L: 20 μm	Au	$1 \cdot 10^4$	Rat/A	Mapping of neuronal activation potentials with 1 mm ² spatial resolution. SNR: 13 dB
[157]	2018	SC	Par. C	25x25 μm^2	Au	$2.6 \cdot 10^4$	Mouse/A	Contextual recording of LFPs and visualization of correlated calcium activity
[158]	2018	ED	Par. C	$\varnothing$: 20 μm	Au	$5.4 \cdot 10^3$ (140 μm), $1.21 \cdot 10^4$ (80 μm), $4.34 \cdot 10^4$ (40 μm), $1.3 \cdot 10^5$ (20 μm)	Mouse/C	Contextual two-photon Ca ⁺⁺ imaging and electrophysiological visually-evoked recording. Significant impedance increase during the first day from implantation, then steady for all the experiment (2 weeks) Multiunit activity recording in the 0.3–7 kHz band. No cortex inflammation within 20 days.
[162]	2018	SC	Par. C	15x15 μm^2	Au	$7 \cdot 10^3$	Mouse/A	Electrode impedance not influenced by operation of the μFIP
[126]	2020	P	Par. C	OECT channel 5<W<500 μm 5<L<500 μm	Au	-	Rat/C	High definition recording of LFPs Absence of macroscopic tissue damage, neuronal loss or gliosis.
[14]	2020	ED	PI	$\varnothing$: 10 and 100 μm	Pt/ Ir Ox	$4.4 \cdot 10^5$	Rat/C	In vivo impedance increases for all electrodes in poorly conformable arrays compared to highly conformable ones. Peak/Mean Ratio greater in IrOx/PEDOT:PSS electrodes than Pt and IrOx ones. Multi-unit activity recorded from both IrOx and PEDOT:PSS up to 12 weeks.

Intracortical recording

[89]	2011	ED	Teflon	$\varnothing$: 25 μm	Pt/ Ir	$1 \cdot 10^4$ (~ 15 t. < bare el.)	Rat/C	Small improvement in SNR over two weeks of chronic implantation. Impedance of coated electrodes is steadily lower than that of bare electrodes. Rapid increase of impedance during the first week due to gliosis.
[176]	2015	ED	Par. C	$\varnothing$: 20 μm	Ti/ Au	$1 \cdot 10^4$ (~ 70 t. < bare el.)	Mouse/C	Higher SNR compared to bare Au electrodes. Action potential recorded.
[163]	2018	DC	Par C	$\varnothing$: 15 μm	Au	-	Mouse/A	Contextual recording of seizure-like events (SLE) and GABA delivery.
[155]	2019	P	Par. C	15x15 μm^2	Au	$1.5 \cdot 10^4$	Mouse/A	High-fidelity recordings of single-unit activities from different cortical layers in the brain. Spontaneous and visual-evoked spike detection across the array. SNR: 29 dB
[156]	2020	3D	PDMS	$\varnothing$: 15 μm	-	$5 \cdot 10^4$ - $1.5 \cdot 10^5$	Mouse/C	Continuous recording of neural activities in a freely moving mouse, including LFPs and single spikes.
[126]	2020	P	Par. C	OECT channel 5<W<500 μm 5<L<500 μm	Au	-	Rat/A	Detection of action potentials from individual neurons.

* P: photolithography; SC: spin coating; ED: electrochemical deposition; DC: dip coating; SP: screen printing; 3D: 3D printing- # Par. C: Parylene C.

3.2. In vivo stimulation

PEDOT-based electrodes have, thanks to their inherent hybrid charge transfer properties, excellent capabilities to cope with the challenge of ion-electron transduction at the neural interface (see Box 2). The metal substrate may inject the current in electronic form into the PEDOT-film, while the PEDOT-film transduce this charge to the surrounding electrolyte in the

form of ions. Charge transfer at the interface, between CP and electrolyte, in this way differs from that of a metal, as the metal can inject charge via the double layer capacitance and via electrochemical reactions (e.g. oxidation) which will be specific to the material and the electrolyte^[97]. The surface of the CP will couple capacitively to the electrolyte, just like a metal electrode. However, as the material is semi-permeable, the internal surface area of the material contributes with a much larger ionic pseudocapacitance contributed by the bulk of the CP film^[177–179]. This bulk may inject or absorb small ions from/to the electrolyte in response to a shift in polarity. Thus, this reaction is not specific to certain reactants but can involve any ions available at the interface that are small enough to penetrate the film. The dynamics of the stimulation pulse, and the diffusion properties of the specific film, will determine to which extent this larger bulk pseudo-capacitance can contribute to charge transfer.

With the typical short pulses during bi-phasic stimulation^[97] only the most superficial layer of the material will in practice be involved in charge injection, which typically results in a charge injection capacity in the range of 2-3 mC/cm² for PEDOT based electrodes^[89,179–181]. With increased pulse durations also deeper layers will contribute to charge injection. For instance, Green et al ^[179] quantified this effect showing that the CIC of PEDOT/pTS increased close to linearly, from 2 to 4 mC/cm², when the pulse duration was increased from 200 to 800 μ s. This large pseudo-capacitance is also the reason for the enormous discrepancy between reported CSC and CIC of CP films. For instance, in the work of Venkatraman (2011), the CSC for PEDOT was estimated to 120 mC/cm² while the CIC for the same electrodes was limited to 3 mC/cm²^[89]. This can be understood by considering that CSC measures charge exchanged for a sweep at 100 mV/s, meaning a full charge/discharge cycle of a CV takes $\sim$ 3 s to complete. CIC accounts for a pulse of a few hundred μ s meaning the time allowed for charge and discharge during CV is >7000 longer than for a typical neurostimulation pulse.

The most obvious benefit of using CPs for stimulating applications is the reduced risk for electrochemical side-reactions as the drastically reduced electrochemical impedance (compared to e.g. a smooth metal) means a reduction of the voltage needed to drive a certain current. In other words, it is possible to stimulate at higher currents before exceeding the ‘water window’ or alternatively use a smaller electrode. As smaller electrodes are crucial for increasing precision of the stimulation, CPs thereby may enable future generations of highly selective neurostimulation applications^[182]. Furthermore, the reduced voltage transient would help reduce the stimulation artefacts in recordings immediately following an injecting pulse^[89]. Despite these clear beneficial qualities, relatively few studies exist where CP electrodes are used for micro-stimulation in vivo. An overview of the available in vivo stimulation studies is provided in **Table 4**. One likely explanation why the stimulation qualities of CPs are not further exploited could be the frequently reported difficulty in establishing a strong bond between the coating and the electrode substrate^[89,183–185]. Stimulation challenge the interface both at the CP/electrolyte and the CP/metal boundary, meaning reliable adhesion is crucial to have a long lasting benefit of the CP coating. In the last 5 years several teams have contributed solutions to this problem, greatly improving the perspective for using PEDOT also for demanding stimulation applications in vivo^[76,186,187]. In fact, stability of PEDOT/PSS during stimulation in vitro has currently been demonstrated for > 7 billion bi-phasic pulses at a fairly high charge injection of 2 mC/cm² without degradation (unpublished data, C Boehler). This finding is encouraging as it demonstrates for that the electrochemical stability of PEDOT/PSS itself is very high, as long as a strong bond is supplied to the underlying interface.

A major benefit of CPs, which this far largely has been overlooked, is their ability to contribute to more efficient stimulation in the low frequency and DC regime, where metal electrodes typically fail. Dijk et al explored the relation between PEDOT:PSS film thickness and stimulation properties showing an increase in safe stimulation current with increasing film

[89]	2011	ED	Micro-wire	Ø: 25 µm and 75 µm	Pt-Ir	3 mC/cm ²	Rat/C	2 weeks, rat cortex, cathodic first bi-phasic stimulation. 10 µA /200 µs and microwire electrodes. Reduced voltage transient compared to PtIr, facilitating safe stimulation.
[179]	2013	ED	PDMS	Ø: 380 µm	Pt	2 mC/cm ²	Cat/A	Suprachoroidal stimulation, analysing the recordings in the visual cortex. Neural responses elicited at charge injection of 76 nC using PEDOT/pTS compared to 85 nC for Pt. At 100 µA, 100 µs charge injection voltage excursion was reduced to 1.5 V compared to 3V for Pt.
[181]	2017	ED	PI	Ø: 50 µm	Au	2 mC/cm ²	Rat/A	Typical pulsing parameters 20 µA, 100 µs. Reproducible neural response for 1 mC/cm ² .
[190]	2017	SA	PI	Ø: 10 µm	Au	-	Rat/A	Bi-phasic pulses using an Mn ²⁺ functionalized PEDOT gel coating. The gel covers a large area of the probe, not only the individual microelectrodes.
[191]	2019	ED	Micro-wires	325 µm ²	Pt-Ir		Rat/C	Bi-phasic charged-balanced at 20 µA, 90 µs (30 µC/cm ²). Stimulation in subthalamic nucleus over 7 and 15 days. PEDOT:BF ₄ .
[184]	2020	ED	PDMS	Ø: 0.31-0.37 mm	Pt	4 mC/cm ²	Rat/C	Bi-phasic charged balanced pulses at 200 µA, 300 µs duration (19.3 µC/cm ²). 5 weeks cochlear stimulation. Comparison of conducting hydrogel (CH) to bare Pt. Some of the CH coating was delaminated and triggered an increased tissue response.

PNS (including spinal cord) stimulation

[192]	2015	ED	Micro-wire	Ø: 100 µm	Au	-	Rat/A	Intramuscular stimulation, pulse duration 100 µs, amplitudes ranging from 0.1-3 mA
[180]	2015	ED	Micro-wire	-	Pt-Ir	-	Rat/C	Spinal cord stimulation, 10 days, comparing PEDOT/CNTs with and without additional dexamethasone. Stimulation at 20 µA, 200 µs pulse duration.
[193]	2017	ED	PDMS	-	-	-	Rat/A	Pulsed stimulation of sciatic nerve, pulse duration in ms range.
[183]	2019	ED	Carbon fibers	Ø: 7 µm 250µm cylindrical	C	80µA, 200 µs	Rat/A	Bi-phasic stimulation, 10-100 µA, 200 µs, in the spinal cord. Coated fibers were more efficient for stimulation (higher active response). Effect of CP was lost after 6000 pulses.
[194]	2020	SC	Visco-plastic PDMS	-	-	-	Rat/C	Voltage pulses up to 1 V (threshold response at 100 mV) to estimate conduction velocity. Total study time, 8 weeks.

* SA Self-Assembly

4. Neural probes based on CNTs, PEDOT:X and PPy

CNTs were added to PEDOT in order to improve its electrical and mechanical properties of the conductive polymer and therefore the performance of epicortical and intracortical probes both in acute and chronic scenarios (a summary can be found in **Table 5**). For instance, intracortical silicon shafts featuring gold microelectrodes coated with carboxyl-functionalized MWCNT/PEDOT:PSS composite were used to record acoustically-evoked single spikes with 2.5 times higher SNR compared to bare electrodes^[195] or to provide stable recording of visually-evoked single-unit and multi-units activity up to 12 weeks^[196]. The superior chronic recording

performance of PEDOT/CNTs composite electrodes compared to PEDOT were ascribed to the highly-porous structure and subsequent better integration with the neural tissue.

Figure 6. Neural probes based on alternative PEDOT dopants or composites. a) Scanning electron microscopy image of a pHEMA-coated PEDOT:PSS/CNT microsphere after recordings in rat; b) impedance modulus of a pHEMA-coated PEDOT-PSS-CNT microsphere before and after recording sessions. Reproduced under terms of the CC-BY license^[197]. Copyright 2016, Castagnola, Maggiolini, Ceseracciu, Ciarpella, Zucchini, De Faveri, Fadiga and Ricci, published by Frontiers in Neuroscience. c) A polyimide neural probe featuring PEDOT/Dex-coated microelectrodes: i) optical image of the layout, ii) probe insertion process using an optical fiber as guiding tool, iii) final placement of the intracortical electrodes, iv) image of the implant on site. Reproduced with permission^[168]. Copyright 2017, Elsevier

In order to improve the long-term stability of the probe and decrease the mechanical mismatch between probe and neural tissue, Castagnola and co. engineered a new kind of neural probe based on PEDOT:PSS-CNT-coated gold microspheres grown on a Pt wire, surrounded by a soft hydrogel made of poly(2-hydroxyethyl methacrylate) (pHEMA), added to avoid direct contact between the CNTs and the tissue (**Fig. 6a**)^[197]. The deposition of high surface area PEDOT:PSS-CNT onto the gold nanospheres 4-fold lowered the 1 kHz impedance compared

to the bare gold (**Fig. 6b**). The slowly-resorbable pHEMA hydrogel preserved the electrochemical performance and high quality recording of action potential from the PEDOT-PSS-CNT coated devices during both acute and chronic experiment on rat model. Only a minor occurrence of the inflammatory tissue reaction was observed^[197]. Somatosensory evoked potentials (SEPs) and spontaneous activity acute recordings were recently recorded from the rat cortex μ ECoG arrays featuring PEDOT:PSS co-deposited with oxidized single walled carbon nanohorns (ox-SWCNHs) or oxidized multi walled carbon nanotubes (ox-MWCNTs)^[73]. In another study, high quality SEPs were detected from the mouse cortex by means of μ ECoG arrays featuring relatively large gold microelectrodes (100 and 200 μ m) coated with carboxylated MWCNT/PEDOT:PSS composite^[198]. Due to the large gain in impedance compared to bare electrodes (3.4 fold in the LFP band, i.e. 1-250 Hz), PEDOT/CNT electrodes were able to record SEP signals with ca. 3-fold higher SNR compared to bare electrodes.

Ludwig and co. investigated the possibility to record low noise neural activity from rat motor cortex by depositing PEDOT doped with tetraethylammonium perchlorate (PEDOT:TEAP) on gold electrodes of an intracortical Si probe^[133]. Single activity was detected only from PEDOT-coated electrodes due to the lower impedance. However, the average number of recorded units for PEDOT sites was highest in the days immediately following surgery and then diminished over the week following surgery due to the progress of the immune response. Similarly, low impedance PEDOT:tetrafluoroborate (PEDOT:TFB) electrodeposited on gold microelectrodes of a single-shank silicon probe, recorded signals from the rat motor cortex with higher SNR compared to bare electrodes only during the first weeks from implantation^[199]. After 12 weeks of recording, only one third of PEDOT:TFB sites still recorded unit activity. PEDOT:Nafion copolymer has been recently proposed by Carli and co.^[200,201]. Nafion is a linear copolymer derived from tetrafluoroethylene and either perfluorosulfonic acid or perfluorocarboxylic acid

monomers^[202–204]. Its biocompatibility has been demonstrated both in vitro and in vivo ^[205,206]. PEDOT:Nafion electrodeposited on porous gold microelectrodes exhibited slightly lower impedance in vitro compared to PEDOT:PSS; this transduced into comparable noise level and SNR when recording in rodent model^[200]. Boehler and co. investigated the impact on chronic recording and foreign body reaction of controlled release of the anti-inflammatory corticosteroid Dexamethasone (Dex) from flexible neural probes implanted in the rat hippocampus (**Fig. 6c**)^[168]. The drug was released on a weekly basis over 12 weeks of implantation from electrodeposited PEDOT/Dex microelectrodes by applying a cyclic voltammetry signal in fully awake animals. Dex-functionalized probes provided stable low frequency (between 2 and 15 Hz) LFPs recordings and impedance characteristics over the entire chronic study, similarly to PEDOT:PSS functionalized control probes. Noteworthy, histological evaluation after 12 weeks of implantation revealed an overall low degree of inflammation around all flexible probes whereas electrodes exposed to active drug release protocols exhibited neurons closer to the electrode sites compared to controls^[168].

Over the last years PPy has been one of the most studied CP thanks to its numerous desirable features like easy oxidation, water solubility, high electrical conductivity and commercial availability as water dispersion^[207]. Moreover, PPy film can be synthesized either chemically or electrochemically^[208–210]. In particular, electrochemical oxidation of pyrrole monomers is the most diffuse approach to deposit PPy films on the surface of neural probe microelectrodes^[210,211]. PPy/MWCNTs composites were developed by Baranauskas and co. to improve the recording performance of PPy-based electrodes.^[212] Low impedance PPy/MWCNTs electrodes were implanted in the vibrissa region of rat somatosensory cortex and allowed to record single unit activity, LFPs and multi-unit activity with an up to fourfold SNR compared to uncoated microelectrodes^[212]. Qi and co. recently developed a highly-stretchable, ultra-soft (Young's modulus ca. 450 kPa) PMDS-based MEAs featuring PPy

electrodes^[193]. A dense layer of PPy nanowires, deposited between the PPy coating and the underlying PDMS by anodic oxidation, was introduced to limit the detachment of PPy from the PDMS substrate by decreasing of the mechanical mismatch between the two components. Once implanted epicortically on the visual cortex of normal and epileptic rats, PPy MEAs succeeded into recording normal rhythms and epileptic EcoG signals^[193].

Table 5. Summary of neural probes based on alternative PEDOT dopants/composites and on PPy.

Ref	Year	CP	Deposition method	Probe substrate	Recording site	Underlayer	In vitro EIS @ 1KHz (Ω)	In vivo	
								Model/Chroni	Notes
								c or Acute	
ECoG									
[198]	2014	PEDOT/MWCNT	ED	PI	100x100 and 200x200 μm^2	Ti/Au	$1 \cdot 10^5$ (~100 t. < bare el.)	Rat/A	Detection of SEPs with 3-fold higher SNR compared to bare electrodes. SNR: 46 ± 13 (spontaneous activity) 77 ± 15 , 89 ± 7 , 98 ± 30 (whiskers stimuli at 0.8, 1.1 and 1.3 V resp.)
[193]	2017	PPy	ED	PDMS	-	PPy	$1 \cdot 10^2$ (~20 t. < Au el.)	Rat/A	Recording of LFPs in normal rats and epileptic activity (10.5-11.5 Hz) in epileptic rats ECoG
[200]	2019	PEDOT: Nafion	ED	PI	$\varnothing$: 140 μm	Porous Au	$1.7 \cdot 10^3$ (~60 t. < bare. el)	Rat/A	Slightly lower impedance in vitro and similar SNR (8-33dB) in acute recording of spontaneous LFP activity compared to PEDOT:PSS.
Intracortical									
[212]	2011	PPy/CNT	ED	PDMS	$\varnothing$: 20 μm	Pt/W	$9.7 \cdot 10^3$ (~36 t. < bare el.)	Rat/A	Increases the SNR of spiking signals and of gamma-range LFPs, without decreasing the SNR in lower frequency LFP bands.
[133]	2011	PEDOT: TEAP [#]	ED	Si	$\varnothing$: 15 μm	Au	$3.7 \cdot 10^5$ (~25 t. < bare el.)	Rat/C	Foreign body reaction greatly affected activity recording for both coated and uncoated electrodes. However, the low starting impedance of coated sites allowed them to record more quality signals than uncoated one.

[195]	2013	PEDOT/ MWCNT	ED	SOI*	Ø:30 µm	Au	$1.2 \cdot 10^4$ (~ 108 t. < bare el.)	Rat/A	Detection of single unit activity following acoustic stimuli including characteristic frequency (CF) and minimum threshold (MT). 2.5 times higher SNR (15 dB) compared to uncoated.
[136]	2015	PEDOT/ MWCNT /DEX	ED	Par. C	Ø:12 µm	Pt/ Ir	$27.6 \cdot 10^4$ (~ 2 t. < bare el.)	Rat/C	Only spars unit activity recorded. Same SNR (1-4 dB) and noise level between coated and uncoated. Delayed worsening of impedance due to gliosis of coated porous electrodes compared to uncoated ones
[199]	2015	PEDOT: TFB	ED	Si	Ø: 15 µm	Ir/ Au	$4 \cdot 10^4$ (~ 4 t. < Au el.)	Rat/C	Unit recording up to 12 weeks. SNR of PEDOT:TFB sites (3-25 dB) was higher than that of Au sites only during the first two weeks. Impedance in vivo (@ 1kHz) significantly increased during the first week likely due to microglia activation and probe insulation.
[197]	2016	PEDOT/ MWCNT	ED	Pt	Ø: 50 µm	Pt/ Au	PEDOT: PSS/CNT (1.0±0.2)	Rat/A &C	Similar impedance in vitro and SNR of pHEMA coated and uncoated probe. Detection of action potentials up to 28 days. Limited occurrence of inflammatory response, no obvious neuron loss. SNR ~ 1.7 dB
[196]	2016	PEDOT/ CNT	ED	Si	Ø: 18 µm	Au	$9 \cdot 10^4$ (~ 90 t. < bare el.; 1.1 t. < PEDOT: PSS el.)	Mouse/ C	Visually evoked single-unit (SU) and multi-units (MU) activity invariantly recorded up to 12 weeks with PEDOT/CNTs microelectrodes. Rapid worsening of the recorded signals with PEDOT:PSS microelectrodes. Rapid impedance increase during the first 3 months from surgery. SNR ~ 3 dB
[168]	2017	PEDOT/ Dex	ED	PI	15x15 µm ² and 50x50 µm ²	Pt/ IrOx	.	Rat/C	Similar LFPs recording and impedance in vivo as PEDOT:PSS electrodes. Low degree of inflammation. Probes exposed to active drug release protocols did have neurons closer to the electrode sites compared to controls.
[73]	2018	PEDOT/ ox- SWCNH, PEDOT/ ox- MWCNT	ED	PI	Ø:60 µm	Au	3.8 ± 0.4 (CNHs) 3.5 ± 0.4 (CNTs)	Rat/A	SEP and spontaneous activity recorder. No differences between CNHs (SNR: 2.0 ± 1.5 dB) and CNTs (SNR: 2.1 ± 1.2 dB).
[213]	2020	PEDOT: pTS	ED	C	Ø:6.8 µm	C	$1.9 \cdot 10^4$ (~ 205 t. < bare el.)	Rat/A &C	PEDOT:pTS electrodes exhibited a decrease in recording yield and an increase in 1 kHz impedance. Negligible tissue damage due to extra small thickness of the carbon fiber.

* Si on insulator, # tetraethyleammonium perchlorate

5. Outlook

Stable, high-quality bidirectional communication with neural tissue relies on a number of biotic and abiotic factors that have neither been fully explored nor completely identified. Among them, impaired charge transfer capability at the electrode/electrolyte interface, micro-motions

involving both the probe and the connectors, suboptimal integration with surrounding tissue and occurrence of chronic gliosis, remain formidable hurdles. Therefore, the fabrication of long-lasting neural implantable devices represents a major challenge for materials scientists. The main strategies to address these issues are geared towards engineering the structural aspects of the probe, such as geometry, conformability and mechanical properties^[14]. However, the electrode surface is the actual electrochemical and electrical interface between the probe and the brain, and therefore a key component in the design of the most, robust and lasting neural probe. From this perspective, conductive polymers represent a very attractive arrow in the bow of materials scientists to improve the recording and stimulation performance of current electrodes. (**Figure 7**).

Figure 7. Sketch of the main routes via CPs that can be exploited to build the next generation of bidirectional interfaces. Desired properties are in green, while properties not directly derived from the use of CPs but dependent on the probe features are in italic. Relative sections or boxes in the text are reported within brackets.

One of the most obvious reasons for using CPs is the ability to drastically lower the impedance of the metal electrode and hence the thermal noise of the system. This can be particularly relevant in the case of high-density probes and small recording sites, which are inherently characterized by high impedance. Nevertheless, how such a pronounced drop in impedance obtained in vitro reflects a real advantage in terms of quality and specificity of the neural signals recorded in vivo, especially during chronic studies, is still a matter of discussion. This can be ascribed to the fact that i) below a certain low impedance value, the shunt loss is minimised regardless of the value of the coating impedance; ii) many factors, including the acute/chronic occurrence of the glial scar, affect the overall recorded signals following still poorly understood pathways, making hard to disentangle the contribution of the CP coating alone to the recorded signal^[59,69,214].

Importantly, CP coatings offer other advantages besides low impedance, that still make them highly attractive for neuroelectronic applications. For example, CPs represent a softer (albeit not “as soft as brain”) and more suitable tissue-probe interface than metals or metal oxides. This favors stronger neural cells-electrode interactions, faster and safer charge transfer across the tissue-electrode interface and therefore more stable and durable probes in chronic scenarios. In this context, the recent development of ultra-soft PEDOT-based hydrogels allows one to handle 3D conductive interfaces with mechanical properties very close to that of the neural tissue and therefore appears a particularly promising approach^[100–102]. In addition, while transparency of CPs enables the combination of recording of electrophysiological activity with high resolution imaging techniques such as two-photon imaging^[157], CP coatings can be engineered to deliver anti-inflammatory drugs to decrease FBR^[168] or anti-epileptic drugs to prevent the occurring of seizure events^[163]. Many efforts have been devoted to improving also the long-term electrical and biological performances of CPs. In particular, mixing or co-deposition of PEDOT in the

presence of pristine or functionalized CNTs has proven to be a preferred approach ^[169,180,212]. However, despite the encouraging results in terms of coating stability and low impedance reported in vitro, only minor improvements were apparently achieved in vivo through this approach. The declination of PEDOT's dopants into functional molecules or polymers and even more the introduction in the field of novel n- and p-type conjugated polymers, appear more as a promising route in order to make a quantum leap.

Besides recording, CPs coatings appear even more effective when used for stimulation purposes. Indeed, their large ionic pseudo-capacitance allows higher charge injection capacity and lower voltage transients compared to traditional metal electrodes, yielding lower risks of electrochemical side-reactions during chronic stimulation^[89]. In addition, CP coatings offer the possibility to work in typical ranges not covered by conventional electrodes, i.e. DC and low frequency regimes^[178]. Shifting the focus away from coatings, CP-based organic transistors are likely the most reliable approach to make the leap toward high SNR recording in vivo, due to their inherent in situ signal amplification signal. The possibility of recording LFPs, single- and multi-unit activity from the brain surface with the same accuracy of conventional intracortical probes was demonstrated in rats to date^[90]. Novel propositions addressing the safety “in operando” seem to effectively overcome the issues concerning the applications of additional voltages to the brain surface (Di Lauro et al., under review). These reasons, and a robust mature technology also mutated from other organic electronics applications, prompt the consideration that the use of organic transistors for neural recording in humans could be within reach.

Conflicts of interest

There are no conflicts to declare.

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References

- [1] R. M. Rothschild, *Front. Neuroeng.* **2010**, *3*, DOI 10.3389/fneng.2010.00112.
- [2] L. F. Nicolas-Alonso, J. Gomez-Gil, *Sensors* **2012**, *12*, 1211.
- [3] R. A. Andersen, S. Kellis, C. Klaes, T. Aflalo, *Curr. Biol.* **2014**, *24*, R885.
- [4] F. Boi, T. Moraitis, V. De Feo, F. Diotalevi, C. Bartolozzi, G. Indiveri, A. Vato, *Front. Neurosci.* **2016**, *10*, DOI 10.3389/fnins.2016.00563.
- [5] E. Greenwald, M. R. Masters, N. V. Thakor, *Med. Biol. Eng. Comput.* **2016**, *54*, DOI 10.1007/s11517-015-1429-x.
- [6] M. A. Lebedev, M. A. L. Nicolelis, *Physiol Rev* **2017**, *97*, 767.
- [7] S. M. Won, E. Song, J. T. Reeder, J. A. Rogers, *Cell* **2020**, *181*, 115.
- [8] J. P. Seymour, F. Wu, K. D. Wise, E. Yoon, *Microsystems Nanoeng.* **2017**, *3*, 1.
- [9] S. M. Won, E. Song, J. Zhao, J. Li, J. Rivnay, J. A. Rogers, *Adv. Mater.* **2018**, *30*, 1800534.
- [10] G. Ramantani, M. Dümpelmann, L. Koessler, A. Brandt, D. Cosandier-Rimélé, J. Zentner, A. Schulze-Bonhage, L. G. Maillard, *Epilepsia* **2014**, *55*, 278.
- [11] N. V. Thakor, *Sci. Transl. Med.* **2013**, *5*, DOI 10.1126/scitranslmed.3007303.
- [12] J. Hermiz, L. Hossain, E. M. Arneodo, M. Ganji, N. Rogers, N. Vahidi, E. Halgren, T. Q. Gentner, S. A. Dayeh, V. Gilja, *Front. Neurosci.* **2020**, *14*, DOI 10.3389/fnins.2020.00055.
- [13] A. I. Kashkoush, R. A. Gaunt, L. E. Fisher, T. M. Bruns, D. J. Weber, *Sci. Rep.* **2019**, *9*, DOI 10.1038/s41598-019-38924-w.
- [14] M. Vomero, M. F. Porto Cruz, E. Zucchini, F. Ciarpella, E. Delfino, S. Carli, C. Boehler, M. Asplund, D. Ricci, L. Fadiga, T. Stieglitz, *Biomaterials* **2020**, *255*, 120178.
- [15] M. Vomero, C. Gueli, E. Zucchini, L. Fadiga, J. B. Erhardt, S. Sharma, T. Stieglitz, *Adv. Mater. Technol.* **2020**, *5*, 2070009.
- [16] T. Ball, M. Kern, I. Mutschler, A. Aertsen, A. Schulze-Bonhage, *Neuroimage* **2009**, *46*, 708.
- [17] G. Buzsáki, C. A. Anastassiou, C. Koch, *Nat. Rev. Neurosci.* **2012**, *13*, 407.
- [18] J. F. Miller, M. Neufang, A. Solway, A. Brandt, M. Trippel, I. Mader, S. Hefft, M. Merkow, S. M. Polyn, J. Jacobs, M. J. Kahana, A. Schulze-Bonhage, *Science (80-.)*. **2013**, *342*, 1111 LP.
- [19] T. Bockhorst, F. Pieper, G. Engler, T. Stieglitz, E. Galindo-Leon, A. K. Engel, *Eur. J. Neurosci.* **2018**, *48*, 3583.
- [20] D. Khodagholy, J. N. Gelin, T. Thesen, W. Doyle, O. Devinsky, G. G. Malliaras, G. Buzsáki, *Nat. Neurosci.* **2015**, *18*, 310.
- [21] K. Dion, G. J. N., Z. Zifang, Y. Malcolm, L. Michael, G. J. D., D. Werner, D. Orrin, B. György, *Sci. Adv.* **2021**, *2*, e1601027.
- [22] L. Trebaul, P. Deman, V. Tuyisenge, M. Jedynak, E. Hugues, D. Rudrauf, M. Bhattacharjee, F. Tadel, B. Chanteloup-Foret, C. Saubert, G. C. Reyes Mejia, C. Adam, A. Nica, M. Pail, F. Dubeau, S. Rheims, A. Trébouchon, H. Wang, S. Liu, T. Blauwblomme, M. Garcés, L. De Palma, A. Valentin, E.-L. Metsähonkala, A. M. Petrescu, E. Landré, W. Szurhaj, E. Hirsch, L. Valton, R. Rocamora, A. Schulze-Bonhage, I. Mindruta, S. Francione, L. Maillard, D. Taussig, P. Kahane, O. David, *Neuroimage* **2018**, *181*, 414.
- [23] Y. Novitskaya, M. Dümpelmann, A. Vlachos, P. C. Reinacher, A. Schulze-Bonhage, *Neuroimage* **2020**, *214*, 116769.
- [24] S. Beniczky, D. L. Schomer, *Epileptic Disord.* **2020**, *22*, 697.
- [25] H. Chen, M. Z. Koubeissi, *Contin. Lifelong Learn. Neurol.* **2019**, *25*.
- [26] J. Parvizi, S. Kastner, *Nat. Neurosci.* **2018**, *21*, 474.
- [27] S. Hefft, A. Brandt, S. Zwick, D. von Elverfeldt, I. Mader, J. Cordeiro, M. Trippel, J. Blumberg, A. Schulze-Bonhage, *Neurosurgery* **2013**, *73*, 78.
- [28] S. F. Cogan, *Annu. Rev. Biomed. Eng.* **2008**, *10*, 275.
- [29] D. R. Merrill, M. Bikson, J. G. R. Jefferys, *J. Neurosci. Methods* **2005**, *141*, 171.
- [30] R. V Shannon, *IEEE Trans. Biomed. Eng.* **1992**, *39*, 424.
- [31] A. B. Weinberger, A. E. Green, E. G. Chrysikou, *Front. Hum. Neurosci.* **2017**, *11*, 246.

- [32] L. B. Razza, S. De Smet, A. Moffa, P. Sudbrack-Oliveira, M.-A. Vanderhasselt, A. R. Brunoni, *Psychiatry Res.* **2021**, 302, 114024.
- [33] F. Fregni, S. Thome-Souza, M. A. Nitsche, S. D. Freedman, K. D. Valente, A. Pascual-Leone, *Epilepsia* **2006**, 47, 335.
- [34] K. Kravalis, A. Schulze-Bonhage, *Neurol. Res. Pract.* **2020**, 2, 15.
- [35] F. Manzouri, C. Meisel, L. Kunz, M. Dümpelmann, T. Stieglitz, A. Schulze-Bonhage, *NeuroImage Clin.* **2021**, 31, 102778.
- [36] B. N. Lundstrom, J. Van Gompel, F. Khadjevand, G. Worrell, M. Stead, *Brain Commun.* **2019**, 1, fcz010.
- [37] R. Rokyta, J. Fricová, *Physiol. Res.* **2012**, 61, S23.
- [38] M. J. Morrell, *Neurology* **2011**, 77, 1295.
- [39] T. E. Schläpfer, A. Meyer-Lindenberg, M. Synofzik, V. Visser-Vandewalle, J. Voges, V. A. Coenen, *Dtsch Arztebl Int.* **2021**, 118, 31.
- [40] R. Fisher, V. Salanova, T. Witt, R. Worth, T. Henry, R. Gross, K. Oommen, I. Osorio, J. Nazzaro, D. Labar, M. Kaplitt, M. Sperling, E. Sandok, J. Neal, A. Handforth, J. Stern, A. DeSalles, S. Chung, A. Shetter, D. Bergen, R. Bakay, J. Henderson, J. French, G. Baltuch, W. Rosenfeld, A. Youkilis, W. Marks, P. Garcia, N. Barbaro, N. Fountain, C. Bazil, R. Goodman, G. McKhann, K. Babu Krishnamurthy, S. Papavassiliou, C. Epstein, J. Pollard, L. Tonder, J. Grebin, R. Coffey, N. Graves, *Epilepsia* **2010**, 51, 899.
- [41] R. Ngetich, J. Zhou, J. Zhang, Z. Jin, L. Li, *Front. Integr. Neurosci.* **2020**, 14, 35.
- [42] K. Kim, A. Schedlbauer, M. Rollo, S. Karunakaran, A. D. Ekstrom, N. Tandon, *Brain Stimul.* **2018**, 11, 213.
- [43] R. Biran, D. C. Martin, P. A. Tresco, *Exp. Neurol.* **2005**, 195, 115.
- [44] E. Fernández, B. Greger, P. A. House, I. Aranda, C. Botella, J. Albisua, C. Soto-Sánchez, A. Alfaro, R. A. Normann, *Front. Neuroeng.* **2014**, 7, 24.
- [45] C. S. Bjornsson, S. J. Oh, Y. A. Al-Kofahi, Y. J. Lim, K. L. Smith, J. N. Turner, S. De, B. Roysam, W. Shain, S. J. Kim, *J. Neural Eng.* **2006**, 3, 196.
- [46] M. Jorfi, J. L. Skousen, C. Weder, J. R. Capadona, *J. Neural Eng.* **2015**, 12, 11001.
- [47] T. Massey, L. Kuo, J. L. Fan, M. Maharbiz, *bioRxiv* **2018**, 380006.
- [48] V. S. Polikov, P. A. Tresco, W. M. Reichert, *J. Neurosci. Methods* **2005**, 148, 1.
- [49] S. M. Wellman, J. R. Eles, K. A. Ludwig, J. P. Seymour, N. J. Michelson, W. E. McFadden, A. L. Vazquez, T. D. Y. Kozai, *Adv. Funct. Mater.* **2018**, 28, 1.
- [50] T. D. Y. Kozai, N. B. Langhals, P. R. Patel, X. Deng, H. Zhang, K. L. Smith, J. Lahann, N. A. Kotov, D. R. Kipke, *Nat. Mater.* **2012**, 11, 1065.
- [51] T. M. Fu, G. Hong, T. Zhou, T. G. Schuhmann, R. D. Viveros, C. M. Lieber, *Nat. Methods* **2016**, 13, 875.
- [52] G. Hong, T. M. Fu, M. Qiao, R. D. Viveros, X. Yang, T. Zhou, J. M. Lee, H. G. Park, J. R. Sanes, C. M. Lieber, *Science (80-.)* **2018**, 360, 1447.
- [53] J. L. Skousen, S. M. E. Merriam, O. Srivannavit, G. Perlin, K. D. Wise, P. A. Tresco, in *Prog. Brain Res.*, Elsevier B.V., **2011**, pp. 167–180.
- [54] G. Hong, C. M. Lieber, *Nat. Rev. Neurosci.* **2019**, 20, 330.
- [55] R. A. Green, R. T. Hassarati, J. A. Goding, S. Baek, N. H. Lovell, P. J. Martens, L. A. Poole-Warren, *Macromol. Biosci.* **2012**, 12, 494.
- [56] T. D. Y. Kozai, A. S. Jaquins-Gerstl, A. L. Vazquez, A. C. Michael, X. T. Cui, *ACS Chem. Neurosci.* **2015**, 6, 48.
- [57] S. M. Wellman, T. D. Y. Kozai, *ACS Chem. Neurosci.* **2017**, 8, 2578.
- [58] T. D. Y. Kozai, J. R. Eles, A. L. Vazquez, X. T. Cui, *J. Neurosci. Methods* **2016**, 258, 46.
- [59] J. W. Salatino, K. A. Ludwig, T. D. Y. Kozai, E. K. Purcell, *Nat. Biomed. Eng.* **2017**, 1, 862.
- [60] U. Pannasch, N. Rouach, *Trends Neurosci.* **2013**, 36, 405.
- [61] M. Crisan, S. Yap, L. Casteilla, C.-W. Chen, M. Corselli, T. S. Park, G. Andriolo, B. Sun, B. Zheng, L. Zhang, C. Norotte, P.-N. Teng, J. Traas, R. Schugar, B. M. Deasy, S. Badylak, H.-J. Buhning, J.-

- P. Giacobino, L. Lazzari, J. Huard, B. Péault, *Cell Stem Cell* **2008**, *3*, 301.
- [62] L. Dimou, V. Gallo, *Glia* **2015**, *63*, 1429.
- [63] J. C. Williams, R. L. Rennaker, D. R. Kipke, *Brain Res. Brain Res. Protoc.* **1999**, *4*, 303.
- [64] I. Rembado, E. Castagnola, L. Turella, T. Ius, R. Budai, A. Ansaldo, G. N. Angotzi, F. Debertoldi, D. Ricci, M. Skrap, L. Fadiga, *Int. J. Neural Syst.* **2017**, *27*, DOI 10.1142/S0129065716500520.
- [65] M. Shokouejinejad, D. W. Park, Y. H. Jung, S. K. Brodnick, J. Novello, A. Dingle, K. I. Swanson, D. H. Baek, A. J. Suminski, W. B. Lake, Z. Ma, J. Williams, *Micromachines* **2019**, *10*, 1.
- [66] M. A. L. Nicoletis, M. A. Lebedev, *Nat. Rev. Neurosci.* **2009**, *10*, 530.
- [67] M. Schuettler, M. Ulloa, J. S. Ordonez, T. Stieglitz, in *2013 6th Int. IEEE/EMBS Conf. Neural Eng.*, **2013**, pp. 1171–1173.
- [68] Z. Zhao, R. Gong, L. Zheng, J. Wang, *Sensors (Basel)*. **2016**, *16*, 1851.
- [69] J. P. Neto, P. Baião, G. Lopes, J. Frazão, J. Nogueira, E. Fortunato, P. Barquinha, A. R. Kampff, *Front. Neurosci.* **2018**, *12*, 715.
- [70] M. F. P. Cruz, M. Vomero, E. Zucchini, E. Delfino, M. Asplund, T. Stieglitz, L. Fadiga, in *2019 9th Int. IEEE/EMBS Conf. Neural Eng.*, **2019**, pp. 924–927.
- [71] J. Koch, M. Schuettler, C. Pasluosta, T. Stieglitz, *J. Neural Eng.* **2019**, *16*, 61002.
- [72] M. F. Porto Cruz, Conformable Epicortical Implants for Brain Mapping: A Technological Perspective on Neural Recordings, University of Ferrara, **2021**.
- [73] S. Carli, L. Lambertini, E. Zucchini, F. Ciarpella, A. Scarpellini, M. Prato, E. Castagnola, L. Fadiga, D. Ricci, *Sensors Actuators, B Chem.* **2018**, *271*, 280.
- [74] C. Boehler, D. M. Vieira, U. Egert, M. Asplund, *ACS Appl. Mater. Interfaces* **2020**, *12*, 14855.
- [75] C. Boehler, T. Stieglitz, M. Asplund, *Biomaterials* **2015**, *67*, 346.
- [76] C. Boehler, F. Oberueber, S. Schlabach, T. Stieglitz, M. Asplund, *ACS Appl. Mater. Interfaces* **2017**, *9*, 189.
- [77] K. M. Szostak, L. Grand, T. G. Constandinou, *Front. Neurosci.* **2017**, *11*, 665.
- [78] Y. Shi, L. Peng, Y. Ding, Y. Zhao, G. Yu, *Chem. Soc. Rev.* **2015**, *44*, 6684.
- [79] T. Schoetz, B. Craig, C. Ponce de Leon, A. Bund, M. Ueda, C. T. J. Low, *J. Energy Storage* **2020**, DOI 10.1016/j.est.2019.101176.
- [80] D. T. Simon, E. O. Gabrielsson, K. Tybrandt, M. Berggren, *Chem. Rev.* **2016**, *116*, 13009.
- [81] D. N. Nguyen, H. Yoon, *Polymers (Basel)*. **2016**, *8*, 118.
- [82] R. Balint, N. J. Cassidy, S. H. Cartmell, *Acta Biomater.* **2014**, *10*, 2341.
- [83] A. Savva, R. Hallani, C. Cendra, J. Surgailis, T. C. Hidalgo, S. Wustoni, R. Sheelamanthula, X. Chen, M. Kirkus, A. Giovannitti, A. Salleo, I. McCulloch, S. Inal, *Adv. Funct. Mater.* **2020**, *30*, DOI 10.1002/adfm.201907657.
- [84] M. N. Gueye, A. Carella, J. Faure-Vincent, R. Demadrille, J.-P. Simonato, **2019**, DOI 10.1016/j.pmatsci.2019.100616.
- [85] M. Tomczykowa, M. E. Plonska-Brzezinska, *Polymers (Basel)*. **2019**, *11*, DOI 10.3390/polym11020350.
- [86] D. Mantione, I. del Agua, A. Sanchez-Sanchez, D. Mecerreyes, *Polymers (Basel)*. **2017**, *9*, DOI 10.3390/polym9080354.
- [87] J. Kim, J. Lee, J. You, M. S. Park, M. S. Al Hossain, Y. Yamauchi, J. H. Kim, *Mater. Horizons* **2016**, *3*, DOI 10.1039/C6MH00165C.
- [88] M. E. J. Obien, K. Deligkaris, T. Bullmann, D. J. Bakkum, U. Frey, *Front. Neurosci.* **2015**, *9*, 423.
- [89] S. Venkatraman, J. Hendricks, Z. A. King, A. J. Sereno, S. Richardson-Burns, D. Martin, J. M. Carmena, *IEEE Trans. Neural Syst. Rehabil. Eng.* **2011**, *19*, 307.
- [90] D. Khodagholy, T. Doublet, P. Quilichini, M. Gurfinkel, P. Leleux, A. Ghestem, E. Ismailova, T. Hervé, S. Sanaur, C. Bernard, G. G. Malliaras, *Nat. Commun.* **2013**, *4*, DOI 10.1038/ncomms2573.
- [91] S. Stříteský, A. Marková, J. Víteček, E. Šafaříková, M. Hrabal, L. Kubáč, L. Kubala, M. Weiter, M. Vala, *J. Biomed. Mater. Res. A* **2018**, *106*, 1121.
- [92] J. Rivnay, S. Inal, B. A. Collins, M. Sessolo, E. Stavrinidou, X. Strakosas, C. Tassone, D. M. Delongchamp, G. G. Malliaras, *Nat. Commun.* **2016**, *7*, 1.

- [93] G. G. Malliaras, *Biochim. Biophys. Acta* **2013**, 1830, 4286.
- [94] M. Di Lauro, A. De Salvo, G. C. Sebastianella, M. Bianchi, S. Carli, M. Murgia, L. Fadiga, F. Biscarini, *ACS Appl. Electron. Mater.* **2020**, 2, 1849.
- [95] S. Inal, G. G. Malliaras, J. Rivnay, *Nat. Commun.* **2017**, 8, 1.
- [96] A. Williamson, J. Rivnay, L. Kergoat, A. Jonsson, S. Inal, I. Uguz, M. Ferro, A. Ivanov, T. A. Sjöström, D. T. Simon, M. Berggren, G. G. Malliaras, C. Bernard, *Adv. Mater.* **2015**, 27, 3138.
- [97] C. Boehler, S. Carli, L. Fadiga, T. Stieglitz, M. Asplund, *Nat. Protoc.* **2020**, 15, 3557.
- [98] R. A. Green, R. T. Hassarati, L. Bouchinet, C. S. Lee, G. L. M. Cheong, J. F. Yu, C. W. Dodds, G. J. Suaning, L. A. Poole-Warren, N. H. Lovell, *Biomaterials* **2012**, 33, 5875.
- [99] X. Cui, D. C. Martin, *Sensors Actuators A Phys.* **2003**, 103, 384.
- [100] V. R. Feig, H. Tran, M. Lee, Z. Bao, *Nat. Commun.* **2018**, 9, 2740.
- [101] M. Shur, F. Fallegger, E. Pirondini, A. Roux, A. Bichat, Q. Barraud, G. Courtine, S. P. Lacour, *ACS Appl. Bio Mater.* **2020**, 3, 4388.
- [102] B. Lu, H. Yuk, S. Lin, N. Jian, K. Qu, J. Xu, X. Zhao, *Nat. Commun.* **2019**, 10, 1043.
- [103] H. Yano, K. Kudo, K. Marumo, H. Okuzaki, *Sci. Adv.* **2019**, 5, eaav9492.
- [104] H. Shi, C. Liu, Q. Jiang, J. Xu, *Adv. Electron. Mater.* **2015**, 1, 1500017.
- [105] N. Kim, S. Kee, S. H. Lee, B. H. Lee, Y. H. Kahng, Y.-R. Jo, B.-J. Kim, K. Lee, *Adv. Mater.* **2014**, 26, 2268.
- [106] Y. Xia, K. Sun, J. Ouyang, *Adv. Mater.* **2012**, 24, 2436.
- [107] Y. H. Kim, C. Sachse, M. L. MacHala, C. May, L. Müller-Meskamp, K. Leo, *Adv. Funct. Mater.* **2011**, 21, 1076.
- [108] M. Di Lauro, S. Benaglia, M. Berto, C. A. Bortolotti, M. Zoli, F. Biscarini, *Colloids Surfaces B Biointerfaces* **2018**, 168, 143.
- [109] F. Decataldo, T. Cramer, D. Martelli, I. Gualandi, W. S. Korim, S. T. Yao, M. Tessarolo, M. Murgia, E. Scavetta, R. Amici, B. Fraboni, *Sci. Rep.* **2019**, 9, 1.
- [110] J. Y. Gerasimov, R. Gabrielsson, R. Forchheimer, E. Stavriniidou, D. T. Simon, M. Berggren, S. Fabiano, *Adv. Sci.* **2019**, 6, 1.
- [111] M. Giordani, M. Sensi, M. Berto, M. Di Lauro, C. A. Bortolotti, H. L. Gomes, M. Zoli, F. Zerbetto, L. Fadiga, F. Biscarini, *Adv. Funct. Mater.* **2020**, n/a, 2002141.
- [112] A. V. Volkov, K. Wijeratne, E. Mitraka, U. Ail, D. Zhao, K. Tybrandt, J. W. Andreasen, M. Berggren, X. Crispin, I. V. Zozoulenko, *Adv. Funct. Mater.* **2017**, 27, 1.
- [113] D. A. Koutsouras, P. Gkoupidenis, C. Stolz, V. Subramanian, G. G. Malliaras, D. C. Martin, *ChemElectroChem* **2017**, 4, 2321.
- [114] J. Rivnay, P. Leleux, M. Ferro, M. Sessolo, A. Williamson, D. A. Koutsouras, D. Khodagholy, M. Ramuz, X. Strakosas, R. M. Owens, C. Benar, J. M. Badier, C. Bernard, G. G. Malliaras, *Sci. Adv.* **2015**, 1, 1.
- [115] J. Newman, W. Tiedemann, *AIChE J.* **1975**, 21, 25.
- [116] M. et al. Bianchi, *J. Mater. Chem. C* **n.d.**
- [117] M. Giordani, M. Berto, M. Di Lauro, C. A. Bortolotti, M. Zoli, F. Biscarini, *ACS Sensors* **2017**, 2, 1756.
- [118] A. Campana, T. Cramer, D. T. Simon, M. Berggren, F. Biscarini, *Adv. Mater.* **2014**, 26, 3874.
- [119] F. Greco, A. Zucca, S. Taccola, A. Menciasci, T. Fujie, H. Haniuda, S. Takeoka, P. Dario, V. Mattoli, *Soft Matter* **2011**, 7, 10642.
- [120] L. D. Garma, L. M. Ferrari, P. Scognamiglio, F. Greco, F. Santoro, *Lab Chip* **2019**, 19, 3776.
- [121] A. Sanchez-Sanchez, I. del Agua, G. G. Malliaras, D. Mecerreyes, in *Smart Polym. Their Appl.*, Elsevier, **2019**, pp. 191–218.
- [122] S. K. M. Jönsson, J. Birgersson, X. Crispin, G. Greczynski, W. Osikowicz, A. W. Denier van der Gon, W. R. Salaneck, M. Fahlman, *Synth. Met.* **2003**, 139, 1.
- [123] J. Rivnay, R. M. Owens, G. G. Malliaras, *Chem. Mater.* **2014**, 26, 679.
- [124] F. Jonas, J. T. Morrison, *Synth. Met.* **1997**, 85, 1397.
- [125] J. Rivnay, S. Inal, A. Salleo, R. M. Owens, M. Berggren, G. G. Malliaras, *Nat. Rev. Mater.* **2018**,

- 3, DOI 10.1038/natrevmats.2017.86.
- [126] C. Cea, G. D. Spyropoulos, P. Jastrzebska-Perfect, J. J. Ferrero, J. N. Gelinas, D. Khodagholy, *Nat. Mater.* **2020**, DOI 10.1038/s41563-020-0638-3.
 - [127] Y. Chen, G. Liu, C. Wang, W. Zhang, R. W. Li, L. Wang, *Mater. Horizons* **2014**, *1*, 489.
 - [128] D. Alemu, H. Y. Wei, K. C. Ho, C. W. Chu, *Energy Environ. Sci.* **2012**, *5*, 9662.
 - [129] G. D. Spyropoulos, C. O. Ramirez Quiroz, M. Salvador, Y. Hou, N. Gasparini, P. Schweizer, J. Adams, P. Kubis, N. Li, E. Spiecker, T. Ameri, H. J. Egelhaaf, C. J. Brabec, *Energy Environ. Sci.* **2016**, *9*, 2302.
 - [130] W. H. Kim, A. J. Mäkinen, N. Nikolov, R. Shashidhar, H. Kim, Z. H. Kafafi, *Appl. Phys. Lett.* **2002**, *80*, 3844.
 - [131] X. Li, J. Shao, S. K. Kim, C. Yao, J. Wang, Y. R. Miao, Q. Zheng, P. Sun, R. Zhang, P. V. Braun, *Nat. Commun.* **2018**, *9*, DOI 10.1038/s41467-018-04937-8.
 - [132] X. Cui, D. C. Martin, *Sensors Actuators B Chem.* **2003**, *89*, 92.
 - [133] K. A. Ludwig, N. B. Langhals, M. D. Joseph, S. M. Richardson-Burns, J. L. Hendricks, D. R. Kipke, *J. Neural Eng.* **2011**, *8*, 1.
 - [134] J. Bobacka, A. Lewenstam, A. Ivaska, *J. Electroanal. Chem.* **2000**, *489*, 17.
 - [135] E. Barsoukov, J. R. Macdonald, *Impedance Spectroscopy: Theory, Experiment, and Applications*, John Wiley & Sons, Inc., **2005**.
 - [136] N. A. Alba, Z. J. Du, K. A. Catt, T. D. Y. Kozai, X. T. Cui, *Biosensors* **2015**, *5*, 618.
 - [137] L. Ferlauto, P. Vagni, A. Fanelli, E. G. Zollinger, K. Monsorno, R. C. Paolicelli, D. Ghezzi, *Biomaterials* **2021**, *274*, 120889.
 - [138] G. Dijk, A. L. Rutz, G. G. Malliaras, *Adv. Mater. Technol.* **2020**, *5*, 1900662.
 - [139] M. Di Lauro, M. Berto, M. Giordani, S. Benaglia, G. Schweicher, D. Vuillaume, C. A. Bortolotti, Y. H. Geerts, F. Biscarini, *Adv. Electron. Mater.* **2017**, *3*, 1.
 - [140] L. Fang, S. Dai, Y. Zhao, D. Liu, J. Huang, *Adv. Electron. Mater.* **2020**, *6*, 1901217.
 - [141] S. Dai, X. Wu, D. Liu, Y. Chu, K. Wang, B. Yang, J. Huang, *ACS Appl. Mater. Interfaces* **2018**, *10*, 21472.
 - [142] M. Di Lauro, S. la Gatta, C. A. Bortolotti, V. Beni, V. Parkula, S. Drakopoulou, M. Giordani, M. Berto, F. Milano, T. Cramer, M. Murgia, A. Agostiano, G. M. Farinola, M. Trotta, F. Biscarini, *Adv. Electron. Mater.* **2020**, *6*, 1900888.
 - [143] Y. Lee, J. Y. Oh, W. Xu, O. Kim, T. R. Kim, J. Kang, Y. Kim, D. Son, J. B.-H. Tok, M. J. Park, Z. Bao, T.-W. Lee, *Sci. Adv.* **2018**, *4*, eaat7387.
 - [144] A. Giovannitti, D. T. Sbircea, S. Inal, C. B. Nielsen, E. Bandiello, D. A. Hanifi, M. Sessolo, G. G. Malliaras, I. McCulloch, J. Rivnay, *Proc. Natl. Acad. Sci. U. S. A.* **2016**, *113*, 12017.
 - [145] B. Meng, J. Liu, L. Wang, *Polym. Chem.* **2020**, *11*, 1261.
 - [146] D. Khodagholy, T. Doublet, M. Gurfinkel, P. Quilichini, E. Ismailova, P. Leleux, T. Herve, S. Sanaur, C. Bernard, G. G. Malliaras, *Adv. Mater.* **2011**, *23*, 268.
 - [147] S. Choi, H. Lee, R. Ghaffari, T. Hyeon, D.-H. Kim, *Adv. Mater.* **2016**, *28*, 4203.
 - [148] T. Cramer, A. Campana, F. Leonardi, S. Casalini, A. Kyndiah, M. Murgia, F. Biscarini, *J. Mater. Chem. B* **2013**, *1*, 3728.
 - [149] J. Rivnay, P. Leleux, M. Sessolo, D. Khodagholy, T. Hervé, M. Flocchi, G. G. Malliaras, *Adv. Mater.* **2013**, *25*, 7010.
 - [150] M. Berto, C. Diacci, R. D'Agata, M. Pinti, E. Bianchini, M. Di Lauro, S. Casalini, A. Cossarizza, M. Berggren, D. Simon, G. Spoto, F. Biscarini, C. A. Bortolotti, *Adv. Biosyst.* **2018**, *2*, 1700072.
 - [151] M. Berto, S. Casalini, M. Di Lauro, S. L. Marasso, M. Cocuzza, D. Perrone, M. Pinti, A. Cossarizza, C. F. Pirri, D. T. Simon, M. Berggren, F. Zerbetto, C. A. Bortolotti, F. Biscarini, *Anal. Chem.* **2016**, *88*, 12330.
 - [152] S. Casalini, F. Leonardi, T. Cramer, F. Biscarini, *Org. Electron.* **2013**, *14*, 156.
 - [153] F. Alibart, S. Pleutin, D. Guérin, C. Novembre, S. Lenfant, K. Lmimouni, C. Gamrat, D. Vuillaume, *Adv. Funct. Mater.* **2010**, *20*, 330.
 - [154] J. Y. Gerasimov, R. Gabrielsson, R. Forchheimer, E. Stavrinidou, D. T. Simon, M. Berggren, S. Fabiano, *Adv. Sci.* **2019**, *6*, 1801339.

- [155] K. J. Seo, P. Artoni, Y. Qiang, Y. Zhong, X. Han, Z. Shi, W. Yao, M. Fagiolini, H. Fang, *Adv. Biosyst.* **2019**, 3, DOI 10.1002/adbi.201800276.
- [156] H. Yuk, B. Lu, S. Lin, K. Qu, J. Xu, J. Luo, X. Zhao, *Nat. Commun.* **2020**, 11, DOI 10.1038/s41467-020-15316-7.
- [157] M. J. Donahue, A. Kaszas, G. F. Turi, B. Rózsa, A. Slézia, I. Vanzetta, G. Katona, C. Bernard, G. G. Malliaras, A. Williamson, *eneuro* **2018**, 5, ENEURO.0187.
- [158] Y. Qiang, P. Artoni, K. J. Seo, S. Culaclii, V. Hogan, X. Zhao, Y. Zhong, X. Han, P. M. Wang, Y. K. Lo, Y. Li, H. A. Patel, Y. Huang, A. Sambangi, J. S. V. Chu, W. Liu, M. Fagiolini, H. Fang, *Sci. Adv.* **2018**, 4, DOI 10.1126/sciadv.aat0626.
- [159] W. Lee, D. Kim, N. Matsuhisa, M. Nagase, M. Sekino, G. G. Malliaras, T. Yokota, T. Someya, *Proc. Natl. Acad. Sci. U. S. A.* **2017**, 114, 10554.
- [160] T. Yasuhara, T. Agari, M. Kameda, A. Kondo, S. Kuramoto, M. Jing, T. Sasaki, A. Toyoshima, S. Sasada, K. Sato, A. Shinko, T. Wakamori, Y. Okuma, Y. Miyoshi, N. Tajiri, C. V. Borlongan, I. Date, *Int. J. Mol. Sci.* **2013**, 14, 23390.
- [161] M. Gernert, M. Feja, *Pharmaceutics* **2020**, 12, DOI 10.3390/pharmaceutics12121134.
- [162] C. M. Proctor, I. Uguz, A. Slezia, V. Curto, S. Inal, A. Williamson, G. G. Malliaras, *Adv. Biosyst.* **2019**, 3, DOI 10.1002/adbi.201800270.
- [163] C. M. Proctor, A. Slézia, A. Kaszas, A. Ghestem, I. del Agua, A. M. Pappa, C. Bernard, A. Williamson, G. G. Malliaras, *Sci. Adv.* **2018**, 4, DOI 10.1126/sciadv.aau1291.
- [164] A. Puiggali-Jou, L. J. del Valle, C. Alemán, *J. Control. Release* **2019**, 309, 244.
- [165] R. Green, M. R. Abidian, *Adv. Mater.* **2015**, 27, 7620.
- [166] Z. Aqrawe, J. Montgomery, J. Travas-Sejdic, D. Svirskis, *Sensors Actuators B Chem.* **2018**, 257, 753.
- [167] M. R. Abidian, D.-H. Kim, D. C. Martin, *Adv. Mater.* **2006**, 18, 405.
- [168] C. Kleber, T. Stieglitz, M. Asplund, U. G. Hofmann, C. Bohler, N. Martini, I. Dryg, Y. Xie, *Biomaterials* **2017**, 129, 176.
- [169] E. Castagnola, S. Carli, M. Vomero, A. Scarpellini, M. Prato, N. Goshi, L. Fadiga, S. Kassegne, D. Ricci, *Biointerphases* **2017**, 12, 031002.
- [170] E. O. Gabrielsson, A. Armgarth, P. Hammarström, K. P. R. Nilsson, M. Berggren, *Macromol. Mater. Eng.* **2016**, 301, 359.
- [171] Y. Zhong, R. V Bellamkonda, *Brain Res.* **2007**, 1148, 15.
- [172] A. Noailles, L. Fernández-Sánchez, P. Lax, N. Cuenca, *J. Neuroinflammation* **2014**, 11, 186.
- [173] N. Yanguas-Casás, M. A. Barreda-Manso, S. Pérez-Rial, M. Nieto-Sampedro, L. Romero-Ramírez, *Mol. Neurobiol.* **2017**, 54, 6737.
- [174] S. Carli, G. Fioravanti, A. Armirotti, F. Ciarpella, M. Prato, G. Ottonello, M. Salerno, A. Scarpellini, D. Perrone, E. Marchesi, D. Ricci, L. Fadiga, *Chem. - A Eur. J.* **2019**, 25, 2322.
- [175] S. Carli, C. Trapella, A. Armirotti, A. Fantinati, G. Ottonello, A. Scarpellini, M. Prato, L. Fadiga, D. Ricci, *Chem. - A Eur. J.* **2018**, 24, 10300.
- [176] V. Castagnola, E. Descamps, A. Lecestre, L. Dahan, J. Remaud, L. G. Nowak, C. Bergaud, *Biosens. Bioelectron.* **2015**, 67, 450.
- [177] G. A. Snook, C. Peng, D. Fray, G. Chen, *Electrochem. commun.* **2007**, 9, 83.
- [178] J. Leal, N. Jedrusik, S. Shaner, C. Bohler, M. Asplund, *Biomaterials* **2021**, 275, 120949.
- [179] R. A. Green, P. B. Matteucci, R. T. Hassarati, B. Giraud, C. W. D. Dodds, S. Chen, P. J. Byrnes-Preston, G. J. Suaning, L. A. Poole-Warren, N. H. Lovell, *J. Neural Eng.* **2013**, 10, DOI 10.1088/1741-2560/10/1/016009.
- [180] C. L. Kolarcik, K. Catt, E. Rost, I. N. Albrecht, D. Bourbeau, Z. Du, T. D. Y. Kozai, X. Luo, D. J. Weber, X. T. Cui, *J. Neural Eng.* **2015**, 12, 16008.
- [181] A. Schander, H. Stemmann, A. K. Kreiter, W. Lang, *Proceedings* **2017**, 1, 511.
- [182] J. R. Eles, T. D. Y. Kozai, *Biomaterials* **2020**, 234, 119767.
- [183] H. Vara, J. E. Collazos-Castro, *Acta Biomater.* **2019**, 90, 71.
- [184] A. N. Dalrymple, U. A. Robles, M. Huynh, B. A. Nayagam, R. A. Green, L. A. Poole-Warren, J. B.

- Fallon, R. K. Shepherd, *J. Neural Eng.* **2020**, *17*, 26018.
- [185] E. M. Thaning, M. L. M. Asplund, T. A. Nyberg, O. W. Inganäs, H. von Holst, *J. Biomed. Mater. Res. Part B Appl. Biomater.* **2010**, *93B*, 407.
- [186] A. Inoue, H. Yuk, B. Lu, X. Zhao, *Sci. Adv.* **2020**, *6*, eaay5394.
- [187] L. Ouyang, B. Wei, C. Kuo, S. Pathak, B. Farrell, D. C. Martin, *Sci. Adv.* **2017**, *3*, e1600448.
- [188] G. Dijk, H. J. Ruigrok, R. P. O'Connor, *Adv. Mater. Interfaces* **2020**, *7*, 2000675.
- [189] C. Boehler, M. Asplund, in *2018 40th Annu. Int. Conf. IEEE Eng. Med. Biol. Soc.*, **2018**, pp. 2202–2205.
- [190] H. W.-C., L. Y.-C., C. C.-Y., L. H.-Y., C. Y.-Y., C. S.-Y., *Biomaterials* **2017**, *122*, 141.
- [191] C. Bodart, N. Rossetti, J. Hagler, P. Chevreau, D. Chhin, F. Soavi, S. B. Schougaard, F. Amzica, F. Cicoira, *ACS Appl. Mater. Interfaces* **2019**, *11*, 17226.
- [192] H. C. Tian, J. Q. Liu, X. Y. Kang, Q. He, B. Yang, X. Chen, C. S. Yang, *Sensors Actuators, A Phys.* **2015**, *228*, 28.
- [193] D. Qi, Z. Liu, Y. Liu, Y. Jiang, W. R. Leow, M. Pal, S. Pan, H. Yang, Y. Wang, X. Zhang, J. Yu, B. Li, Z. Yu, W. Wang, X. Chen, *Adv. Mater.* **2017**, *29*, DOI 10.1002/adma.201702800.
- [194] Y. Liu, J. Li, S. Song, J. Kang, Y. Tsao, S. Chen, V. Mottini, K. McConnell, W. Xu, Y. Q. Zheng, J. B. H. Tok, P. M. George, Z. Bao, *Nat. Biotechnol.* **2020**, *38*, 1031.
- [195] S. Chen, W. Pei, Q. Gui, R. Tang, Y. Chen, S. Zhao, H. Wang, H. Chen, *Sensors Actuators, A Phys.* **2013**, *193*, 141.
- [196] T. D. Y. Kozai, K. Catt, Z. Du, K. Na, O. Srivannavit, R. U. M. Haque, J. Seymour, K. D. Wise, E. Yoon, X. T. Cui, *IEEE Trans. Biomed. Eng.* **2016**, *63*, 111.
- [197] E. Castagnola, E. Maggiolini, L. Ceseracciu, F. Ciarpella, E. Zucchini, S. De Faveri, L. Fadiga, D. Ricci, *Front. Neurosci.* **2016**, *10*, 1.
- [198] E. Castagnola, L. Maiolo, E. Maggiolini, A. Minotti, M. Marrani, F. Maita, A. Pecora, G. N. Angotzi, A. Ansaldo, M. Boffini, L. Fadiga, G. Fortunato, D. Ricci, *IEEE Trans. Neural Syst. Rehabil. Eng.* **2015**, *23*, 342.
- [199] H. Charkhkar, G. L. Knaack, D. G. Mchail, H. S. Mandal, N. Peixoto, J. F. Robinson, T. C. Dumas, J. J. Pancrazio, *Acta Biomater.* **2016**, *32*, 57.
- [200] S. Carli, M. Bianchi, E. Zucchini, M. Di Lauro, M. Prato, M. Murgia, L. Fadiga, F. Biscarini, *Adv. Healthc. Mater.* **2019**, *1900765*, 1900765.
- [201] S. Carli, M. Di Lauro, M. Bianchi, mauro murgia, A. De Salvo, M. Prato, L. Fadiga, F. Biscarini, *ACS Appl. Mater. Interfaces* **2020**, DOI 10.1021/acsami.0c06538.
- [202] C. R. Martin, T. A. Rhoades, J. A. Ferguson, *Anal. Chem.* **1982**, *54*, 1639.
- [203] I. Rubinstein, A. J. Bard, *J. Am. Chem. Soc.* **1981**, *103*, 5007.
- [204] H. S. White, J. Leddy, A. J. Bard, *J. Am. Chem. Soc.* **1982**, *104*, 4811.
- [205] G. Kim, H. Kim, I. J. Kim, J. R. Kim, J. I. Lee, M. Ree, *J. Biomater. Sci. Polym. Ed.* **2009**, *20*, 1687.
- [206] S. Guzzo, S. Carli, B. Pavan, A. Lunghi, M. Murgia, M. Bianchi, *Nanomater.* **2021**, *11*, DOI 10.3390/nano11082022.
- [207] H. J. Kharat, K. P. Kakde, P. A. Savale, K. Datta, P. Ghosh, M. D. Shirsat, *Polym. Adv. Technol.* **2007**, *18*, 397.
- [208] J. C. Thiéblemont, M. F. Planche, C. Petrescu, J. M. Bouvieff, G. Bidan, *Stability of Chemically Synthesized Polypyrrole Films*, **1993**.
- [209] A. Wu, H. Kolla, S. K. Manohar, *Macromolecules* **2005**, *38*, 7873.
- [210] M. Nishizawa, H. Nozaki, H. Kaji, T. Kitazume, N. Kobayashi, T. Ishibashi, T. Abe, *Biomaterials* **2007**, *28*, 1480.
- [211] Z. D. Kojabad, S. A. Shojaosadati, S. M. Firoozabadi, S. Hamed, *J. Solid State Electrochem.* **2019**, 1533.
- [212] G. Baranauskas, E. Maggiolini, E. Castagnola, A. Ansaldo, A. Mazzoni, G. N. Angotzi, A. Vato, D. Ricci, S. Panzeri, L. Fadiga, *J. Neural Eng.* **2011**, *8*, DOI 10.1088/1741-2560/8/6/066013.
- [213] E. J. Welle, P. R. Patel, J. E. Woods, A. Petrossians, E. Della Valle, A. Vega-Medina, J. M. Richie, D. Cai, J. D. Weiland, C. A. Chestek, *J. Neural Eng.* **2020**, *17*, DOI 10.1088/1741-2552/ab8343.
- [214] M. Vomero, M. F. P. Cruz, E. Zucchini, A. Shabani, E. Delfino, S. Carli, L. Fadiga, D. Ricci, T.

Stieglitz, *Conf. Proc. ... Annu. Int. Conf. IEEE Eng. Med. Biol. Soc. IEEE Eng. Med. Biol. Soc. Annu. Conf.* **2018**, 2018, 4464.