# **Supporting Information**

## Strain-Modulated Interlayer Charge and Energy Transfers in MoS<sub>2</sub>/WS<sub>2</sub> Heterobilayer

Joon-Seok Kim,<sup>1,2,\*</sup> Nikhilesh Maity,<sup>3</sup> Myungsoo Kim,<sup>2</sup> Suyu Fu,<sup>4</sup> Rinkle Juneja,<sup>3</sup> Abhishek Singh,<sup>3,\*</sup> Deji Akinwande,<sup>2,\*</sup> and Jung-Fu Lin<sup>4,\*</sup>

<sup>1</sup> Department of Materials Science and Engineering, Northwestern University, Evanston, IL 60208, USA

<sup>2</sup> Microelectronics Research Center, The University of Texas at Austin, Austin, TX 78758, USA

<sup>3</sup> Materials Research Centre, Indian Institute of Science, Bangalore 560012, India

<sup>4</sup> Department of Geological Sciences, Jackson School of Geosciences, The University of

Texas at Austin, Austin TX 78712, USA

- \* joonseok.kim@northwestern.edu
- \* <u>abhishek@mrc.iisc.ernet.in</u>
- \* <u>deji@ece.utexas.edu</u>
- \* <u>afu@jsg.utexas.edu</u>



Figure S1. Raman intensity maps of the  $MoS_2/WS_2$  heterostructure on Si chip, showing (a) Si peak, (b)  $MoS_2$  A' peak, and (c)  $WS_2$  A' peak intensities. The outlines of the Si/Al<sub>2</sub>O<sub>3</sub> substrate, 1L-MoS<sub>2</sub>, and 1L-WS<sub>2</sub> are indicated as white, red, and blue dotted lines, respectively.



Figure S2. Evolution of Raman spectra of (a) 1L-MoS<sub>2</sub>, (b) MoS<sub>2</sub>/WS<sub>2</sub> heterostructure, and (c) 1L-WS<sub>2</sub> with increased pressure.



Figure S3. Calculated Raman active modes of the 1L-MoS<sub>2</sub>, 1L-WS<sub>2</sub>, and their heterostructure in ambient pressure conditions. Note that the calculated peak positions slightly deviate from experimental values, by a smaller margin compared to the literature.<sup>1–3</sup> The deviation may be attributed to the approximations used in the calculation, such as frozen phonon method, use of finite size supercell force constant, and vdW force approximation.

#### Note S1.

For the theoretical calculations of strain, the total energy costs due to the compressions have been calculated and shown in Figure S4. More specifically, the hydrostatic pressure has been modeled<sup>4</sup> by estimating the ease of compression along the x, y, and z direction:  $\varepsilon_x = \frac{x - x_0}{x_0}$ ,  $\varepsilon_y$  $= \frac{y - y_0}{y_0}$ , and  $\varepsilon_z = \frac{z - z_0}{z_0}$ , where  $\varepsilon_x$ ,  $\varepsilon_y$ , and  $\varepsilon_z$  are the uniaxial strain along the x, y, and z directions, respectively. The weak vdW interaction along the z-direction results in low energy cost for the out-of-plane direction compared to the in-plane direction, therefore  $\varepsilon_x = \varepsilon_y \neq \varepsilon_z$ . In order to discriminate pressure in each axis, we investigated the individual energy cost for in-plane-only ( $\varepsilon_x = \varepsilon_y \neq 0$ ;  $\varepsilon_z = 0$ ) and out-of-plane-only ( $\varepsilon_x = \varepsilon_y = 0$ ;  $\varepsilon_z \neq 0$ ) components. The hydrostatic pressure was mimicked by estimating the compression in all directions based on the total energy cost consumption (Figure S4). Consequently, applied effective pressure was determined using the energy cost per unit volume change of the system by using the given equation,

$$P = \frac{E_0 - E}{V_0 - V} \tag{1}$$

where, E and E<sub>0</sub> are the total energy for the strained and unstrained systems, respectively, and V and V<sub>0</sub> are the volume of the strained and unstrained systems, respectively. The volume of the cell is defined as  $V = a^2 c \sin (60)$ , where a is the lattice parameter, and c is the thickness of the layer.<sup>5</sup>



Figure S4. Modeling of hydrostatic pressure of the TMDCs monolayer. The variation of the applied strain with respect to the energy cost to compress the in-plane and out-of-plane bonds. (a) 1L-MoS<sub>2</sub> (b) 1L-WS<sub>2</sub>, and (c) MoS<sub>2</sub>/WS<sub>2</sub> heterostructure.



Figure S5. Calculated E' mode frequencies under hydrostatic, in-plane, and out-of-plane stress components (a) for MoS<sub>2</sub> and WS<sub>2</sub> monolayers, and (b) for MoS<sub>2</sub>/WS<sub>2</sub> heterobilayer.



Figure S6. Calculated A' mode frequencies under hydrostatic, in-plane, and out-of-plane stress components (a) for MoS<sub>2</sub> and WS<sub>2</sub> monolayers, and (b) for MoS<sub>2</sub>/WS<sub>2</sub> heterobilayer.

$d\omega/dp$	Hydrostatic			In-plane			Out-of-plane		
(cm <sup>-1</sup> /GPa)	MoS <sub>2</sub>	WS <sub>2</sub>	MoS <sub>2</sub> /	MoS <sub>2</sub>	WS <sub>2</sub>	MoS <sub>2</sub> /	MoS <sub>2</sub>	WS <sub>2</sub>	MoS <sub>2</sub> /
			WS <sub>2</sub>			WS <sub>2</sub>			WS <sub>2</sub>
E' mode	3.60	3.13	4.96/	2.64	2.23	3.74/	2.50	2.26	3.41/
			4.73			3.56			3.27
A' mode	3.66	3.45	4.99/	2.27	2.18	3.41/	3.04	2.90	4.18/
			5.11			3.59			4.21

Table S1. Summarized calculated blueshift rate of in-plane E' and out-of-plane A' modes with in-plane strain, out-of-plane strain, and hydrostatic pressure.



Figure S7. Calculated variation of intralayer S-S distance (d) with hydrostatic pressure for (a) monolayer  $MoS_2$  and (b) monolayer  $WS_2$ . (c) The variation of total inter S-S distance of heterostructure, where the interlayer distance (I) is the most effectively reduced.



Figure S8. Evolution of PL spectra of (a)  $1L-MoS_2$ , (b)  $MoS_2/WS_2$  heterostructure, and (c)  $1L-WS_2$  with increased pressure. Dashed lines are a guide of PL peak energy changes.



Figure S9. PL spectra of MoS<sub>2</sub>-only (bottom), WS<sub>2</sub>-only (top), and HS (middle) regions at representative pressures of (a) 2.1 GPa, (b) 4.1 GPa, (c) 5.8 GPa, and (d) 6.6 GPa. MoS<sub>2</sub>-like and WS<sub>2</sub>-like Gaussian lineshapes are shown in red and blue curves, respectively. At pressure 4.1 GPa and lower, WS<sub>2</sub>-like PL signal is strongly quenched in the HS spectra, whereas at higher pressures, WS<sub>2</sub>-like PL signal is less quenched in the HS spectra



Figure S10. Optical absorption spectra of  $MoS_2/WS_2$  heterostructure calculated by BSE (with e-h interaction; solid lines) and RPA (without e-h interaction; dotted lines) with  $A_{MoS2}$ ,  $B_{MoS2}$ ,  $A_{WS2}$  and  $B_{WS2}$  excitons. The vertical black lines represent the oscillator strength for the optical transition.



Figure S11. B Exciton energy levels as a function of hydrostatic pressure and corresponding in-plane and out-of-plane strain component (a) for  $MoS_2$  and  $WS_2$  monolayers, and (b) for  $MoS_2/WS_2$  heterobilayer.

dE <sub>g</sub> /dp	Hydrostatic			In-plane			Out-of-plane		
(meV/GPa)	MoS <sub>2</sub>	WS <sub>2</sub>	MoS <sub>2</sub> /	MoS <sub>2</sub>	WS <sub>2</sub>	MoS <sub>2</sub> /	MoS <sub>2</sub>	WS <sub>2</sub>	MoS <sub>2</sub> /
			WS <sub>2</sub>			WS <sub>2</sub>			WS <sub>2</sub>
A exciton	12.69	12.11	17.63/	32.63	34.19	49.85/	-9.06	9.14	-13.06
			13.47			55.22			/-21.58
B exciton	18.22	22.46	26.75/	30.14	31.56	39.50/	-1.44	16.55	-2.11
			22.23			23.19			/-8.41

Table S2. Summarized the rate of variation of A and B excitonic energy levels with in-plane, out-of-plane, and hydrostatic pressure.



Figure S12. (a) Variation of conduction band (CB) edges at K and  $\Lambda$  and valence band (VB) edge at K as a function of in-plane pressure in WS<sub>2</sub>. The inset indicates the variation of WS<sub>2</sub> band gap, where red and blue circles indicate direct and indirect band gap, respectively. (b) Closer view of the band gap variation, showing direct-to-indirect band gap transition of WS<sub>2</sub>.



Figure S13. Hybridized band structure of  $MoS_2/WS_2$  heterostructure in GGA-PBE approximation. The color gradient map indicates the projection of bands into isolated monolayers. Also, it is worth emphasizing that the  $\Lambda$ -K indirect bandgap is also clearly visible for WS<sub>2</sub>-dominant (blue) bands.

#### Note S2. Estimation of Interlayer CT/ET Rates

PL intensity is closely governed by the excitonic lifetimes, namely

$$\frac{1}{\tau_E} = \frac{1}{\tau_R} + \frac{1}{\tau_{NR}} + \frac{1}{\tau_{IL}}$$
(2)

where  $\tau_E$ ,  $\tau_R$ ,  $\tau_{NR}$ , and  $\tau_{IL}$  correspond to exciton's total lifetime, radiative lifetime, nonradiative lifetime, and lifetime of interlayer charge/energy transfer in the case of heterostructure, respectively.<sup>6</sup> (In case of monolayers,  $\tau_{IL} \rightarrow \infty$ .) PL intensities of monolayer and heterostructure can then be expressed as

$$I_{mono} \propto \frac{\tau_E}{\tau_R} = \frac{\tau_{NR}}{\tau_R + \tau_{NR}} \tag{3}$$

and

$$I_{hetero} \propto \frac{\tau_E}{\tau_R} = \frac{\tau_{NR} \tau_{IL}}{\tau_{NR} \tau_{IL} + \tau_R \tau_{IL} + \tau_R \tau_{NR}} = \eta \times \frac{\tau_{NR}}{\tau_R + \tau_{NR}}$$
(4)

, respectively, with the introduction of the intensity ratio  $\eta$ . These equations take into account various non-radiative recombination pathways such as defect state-assisted recombination and scattering.

The  $au_{IL}$  then can be expressed as

$$\tau_{IL} = \frac{\eta}{1-\eta} \times \frac{\tau_R \tau_{NR}}{\tau_R + \tau_{NR}}.$$
(5)

Assuming  $\tau_{NR} \approx 20$  ps and  $\tau_{NR} \approx 200$  ps for WS2,<sup>7,8</sup> the change in  $\eta$  from  $\approx 0.3$  at pressure below 1 GPa to  $\approx 0.03$  at 3-4 GPa corresponds to the  $\tau_{IL}$  changing from  $\approx 7.8$  ps to  $\approx 560$  fs, more than an order of magnitude change. The enhancement factor reaching unity at 7.7 GPa implies significant reduction of interlayer transfer (Figure S12). We note that this estimation is in the similar order of magnitude with the timescale to form charge transfer equilibrium at band extrema.<sup>9</sup>



Figure S14. Estimated lifetime of interlayer charge/energy transfers.



Figure S15. Band-decomposed charge densities of  $MoS_2/WS_2$  heterostructure for (a) CB1 and (b) CB2, as viewed along the z-direction (Top View) and x-y plane (Side view). The top view of the CB1 and CB2 shows that shape of the charge distributions ( $dz^2$  orbital) for the unstrained case is circular with smooth edges. With the application of strain, the shape of the charge distribution turns into a hexagonal shape with sharp edges. At the critical transition points, the charge distribution shape becomes completely hexagon, and the corner of each hexagon overlap with each other.

Note S3. Molecular Orbital Diagram

The valence bands and the frontier occupied bands (near the Fermi level) are antibonding in nature, whereas the remaining core valence bands have bonding nature. For unstrained monolayers, dxy and  $dx^2-y^2$  are the highest occupied (VB1) antibonding orbitals of transition metal along with small contribution from  $p_x$  and  $p_y$  states of chalcogen atoms of the respective monolayers. While in MoS<sub>2</sub>/WS<sub>2</sub> vdW heterostructure, they originate from W-dxy and  $dx^2-y^2$  and minute contributions from S (WS<sub>2</sub>)- $p_x$  and  $p_y$  (Figure S13a). The lowest unoccupied (CB1) antibonding states are contributed by  $d_z^2$  of transition metal and  $p_x$ ,  $p_y$  of chalcogen atoms of the respective monolayers, while in MoS<sub>2</sub>/WS<sub>2</sub>, it originates from Mo-S,  $d_z^2$  and S (MoS<sub>2</sub>)- $p_x$ ,  $p_y$  orbitals (Figure S13a).



Figure S16. Molecular orbital (MO) diagram of monolayer  $MoS_2$  (left; red),  $MoS_2/WS_2$  vdW heterostructure (middle) and  $WS_2$  (right; blue) at (a) unstrained condition, (b) at pressure between P1 and P2, (c) at pressure between P2 and P3, and (d) pressure above P3.

Bands	Unstrained (0%)	22.42 GPa (8%)	30.03 GPa (10%)	38.69 GPa (12%)
CB4	$[Mo - dxy, dx^2]$	[Mo - <i>dyz, dxz</i> ]	$[W-s, dz^2]$	$[W-s, dz^2]$
	$[\mathrm{S}(\mathrm{MoS}_2) - pz]$	$[S(MoS_2) - px, py]$	$[S(WS_2) - px, py]$	$[\mathrm{S}(\mathrm{WS}_2) - px, py]$
CB3	[Mo - dyz, dxz]	$[W-s, dz^2]$	• [Mo - <i>dyz, dxz</i> ]	$[Mo - s, dz^2]$
	$[\mathrm{S}(\mathrm{MoS}_2) - px, py]$	$[S(WS_2) - px, py]$	$[S(MoS_2) - px, py]$	$[S(MoS_2) - px, py]$
CB2	$[W-s, dz^2]$	<b>[</b> Mo - <i>dyz</i> , <i>dxz</i> ]	$[\text{Mo} - s, dz^2]$	$\mathbf{Mo} - dyz, dxz]$
	$[\mathbf{S}(\mathbf{WS}_2) - px, py]$	$[\mathrm{S}(\mathrm{MoS}_2) - px, py]$	$[S(MoS_2) - px, py]$	$[\mathrm{S}(\mathrm{MoS}_2) - px, py]$
CB1	$[Mo - s, dz^2]$	$[\text{Mo} - s, dz^2]  / $	• [Mo - <i>dyz, dxz</i> ]	[Mo - dyz, dxz]
	$[\mathrm{S}(\mathrm{MoS}_2) - px, py]$	$[\mathrm{S}(\mathrm{MoS}_2) - px, py]$	$[S(MoS_2) - px, py]$	$\rightarrow$ [S(MoS <sub>2</sub> ) - <i>px, py</i> ]
VB1	$[W - dxy, dx^2]$	$[W - dxy, dx^2]$	$[W - dxy, dx^2]$	$[W - dxy, dx^2]$
	$[\mathrm{S}(\mathrm{WS}_2) - px, py]$	$[S(WS_2) - px, py]$	$[S(WS_2) - px, py]$	$\bullet$ [S(WS <sub>2</sub> ) - $px$ , $py$ ]
VB2	$[Mo - dxy, dx^2]$	$[Mo - dxy, dx^2]$	$[Mo - dxy, dx^2]$	$[Mo - dxy, dx^2]$
	$[S(MoS_2) - px, py]$	$[S(MoS_2) - px, py]$	$[S(MoS_2) - px, py]$	$\bullet [S(MoS_2) - px, py]$
VB3	[Mo - dyz, dxz]	[Mo - dyz, dxz]	[W - dyz, dxz]	[W - dyz, dxz]
	$[\mathrm{S}(\mathrm{MoS}_2) - s, pz]$	$[\mathrm{S}(\mathrm{MoS}_2) - s, pz]$	$[S(WS_2) - s, pz]$	$[\mathrm{S}(\mathrm{WS}_2) - s,  pz]$
VB4	[W - dyz, dxz]	[W - dyz, dxz] /	► [Mo - <i>dyz, dxz</i> ]	[Mo - dyz, dxz]
	$[\mathrm{S}(\mathrm{WS}_2) - s,  pz]$	$[\mathrm{S}(\mathrm{WS}_2) - s,  pz]$	$[\mathrm{S}(\mathrm{MoS}_2) - s, pz]$	$[\mathrm{S}(\mathrm{MoS}_2) - s, pz]$

Table S3. The contribution of electronic states of frontier occupied and unoccupied bands at K point in stained and unstrained  $MoS_2/WS_2$  vdW heterostructure. Red and blue arrows indicate changes in states in  $MoS_2$  and  $WS_2$ , respectively, whereas black arrows indicate no changes.

Bands	Unstrained (0%)	22.42 GPa (8%)	30.03 GPa (10%)	38.69 GPa (12%)
CB4	[Mo - dyz, dxz]	[Mo - dyz, dxz]	[Mo - dyz, dxz]	[Mo - dyz, dxz]
	[S - pz, s]	→ [S - pz, s] —	→ [S - pz, s] —	► [S - pz, s]
CB3	$[Mo - dx^2, dxy]$	$[Mo - dx^2, dxy]$	$[Mo - dx^2, dxy]$	$[Mo - dx^2, dxy]$
	[S - pz] —	→ [S - pz] —	→ [S - pz]	[S - pz]
CB2	[Mo - dyz, dxz]	[Mo - dyz, dxz]	$[Mo - s, dz^2]$	$[Mo - s, dz^2]$
	[S - px, py]	→ [S - px, py]	[S - px, py]	[S - px, py]
CB1	$[Mo - s, dz^2]$	[Mo - s, dz <sup>2</sup> ]	[Mo - dyz, dxz]	[Mo - dyz, dxz]
	[S - px, py]	→ [S - px, py]	[S - px, py]	• [S - px, py]
VB1	$[Mo - dxy, dx^2]$	$[Mo - dxy, dx^2]$	$[Mo - dxy, dx^2]$	$[Mo - dxy, dx^2]$
	[S - px, py] —	→ [S - px, py]	[S - px, py]	[S - px, py]
VB2	[Mo - dyz, dxz]	[Mo - dyz, dxz]	[Mo - dyz, dxz]	[Mo - dyz, dxz]
	[S - s, pz]	→ [S - s, pz] —	→ [S - s, pz]	► [S - s, pz]
VB3	[Mo - pz]	[Mo - pz]	[Mo - pz]	$[Mo - dxy, dx^2]$
	[S - px, py]	→ [S - px, py]	► [S - px, py]	[S - s, pz]
VB4	[Mo - dxy, dx <sup>2</sup> ]	$[Mo - dxy, dx^2]$	[Mo - dxy, $dx^2$ ]	🔪 [Mo - pz]
	[S - s, pz]	[S - s, pz]	[S - s, pz]	[S - px, py]

Table S4. The contribution of electronic states in stained and unstrained  $MoS_2$  monolayer in frontier occupied and unoccupied bands at K point. Red arrows indicate changes in states, whereas black arrows indicate no changes.

Bands	Unstrained (0%)	22.42 GPa (8%)	30.03 GPa (10%)	38.69 GPa (12%)
CB4	[W - dyz, dxz]	[W - dyz, dxz]	[W - dyz, dxz]	[W - dyz, dxz]
	[S - pz, s] —	→ [S - pz, s] —	► [S - pz, s]	► [S - pz, s]
CB3	$[W - dx^2, dxy]$	$[W - dx^2, dxy]$	$[W - dx^2, dxy]$	$[W - dx^2, dxy]$
	[S - pz]	→ [S - pz]	[S - pz] —	→ [S - pz]
CB2	[W - dyz, dxz]	[W - dyz, dxz]	$[W - s, dz^2]$	[W - s, dz <sup>2</sup> ]
	[S - px, py]	→ [S - px, py]	[S - px, py]	→ [S - px, py]
CB1	$[W - s, dz^2]$	$[W - s, dz^2]$	🔌 [W - dyz, dxz]	[W - dyz, dxz]
	[S - px, py]	[S - px, py]	[S - px, py]	[S - px, py]
VB1	$[W - dxy, dx^2]$	$[W - dxy, dx^2]$	$\bullet  [W - dxy, dx^2]$	$[W - dxy, dx^2]$
	[S - px, py]	→ [S - px, py]	[S - px, py]	► [S - px, py]
VB2	[W - dyz, dxz]	[W - dyz, dxz]	[W - dyz, dxz]	[W - dyz, dxz]
	[S - s, pz]	→ [S - s, pz]	[S - s, pz]	[S - s, pz]
VB3	[W - pz]	[W - pz]	[W - pz]	$[W - dxy, dx^2]$
	[S - px, py]	[S - px, py]	► [S - px, py]	[S - s, pz]
VB4	$[W - dxy, dx^2]$	$[W - dxy, dx^2]$	$[W - dxy, dx^2] /$	🔪 [W - pz]
	[S - s, pz]	[S - s, pz]	[S - s, pz]	[S - px, py]

Table S5. The contribution of electronic states in stained and unstrained  $WS_2$  monolayer in frontier occupied and unoccupied bands at K point. Blue arrows indicate changes in states, whereas black arrows indicate no changes.

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